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WADC TECHNICAL REPORT 54-531
SUPPLEMENT 1

INVESTIGATION OF MATERIALS
FATIGUE PROBLEMS
APPLICABLE TO PROPELLER DESIGN

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FOREWORD

This report was prepared by the Technical Section, H.N. Cummings, Consulting Engineer, and F.B. Stulen, Chief Technical Engineer, and the Metallurgy Department, W.C. Schulte, Chief Metallurgist, of the Curtiss-Wright Corporation, Propeller Division, under Contract 33(616)-493, and covers work performed during the period of 1 April to 30 September 1955. The work was supported jointly by the Materials Laboratory and the Propeller Laboratory of Wright Air Development Center and the contract was initiated under Project No. 3346, "Propeller Blades", Task No. 73497, "Investigation of the Statistical Nature of Fatigue", formerly RDO No. 591-80. The contract was monitored by Lt. D.M. Forney, Jr., Materials Laboratory and J.S. Keeler, Propeller Laboratory, Project Engineers.

The interest and suggestions of Mr. G.W. Brady, Director of Engineering of Curtiss-Wright Corporation Propeller Division, are gratefully acknowledged. Other personnel of the Propeller Division who have assisted in the work are R.B. Christofferson, W.D. Lamson, F.G. Lehman, J.H. Redfern, C.A. Madden, and J.J. Scannelli.

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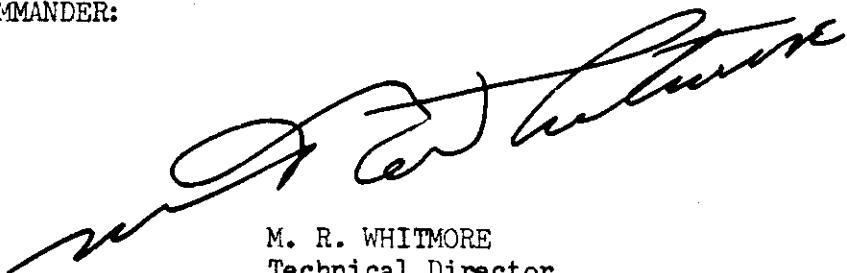
ABSTRACT

WADC Technical Report 54-531 reported the fatigue characteristics of SAE 4340 steel as a guide for propeller and rotor designers. Stress levels for probabilities of 10, 50 and 90% survival under repeated loading were determined for a constant life. The knowledge, however, that 90% of a material will survive a given stress for, say, ten million cycles is not adequate for the design engineer. What he would like to know is at what stress 100% of the material will survive. Therefore, the tests reported in WADC Technical Report 54-531 have been supplemented by additional fatigue tests in the long-life region in order to determine higher survival probabilities, in the order of 99%. Inclusion studies have revealed variability within the steel ingot that caused greater scatter in strength than was predicted by the tests reported in WADC TR 54-531.

PUBLICATION REVIEW

This report has been reviewed and is approved.

FOR THE COMMANDER:



M. R. WHITMORE
Technical Director
Materials Laboratory
Directorate of Research

Controls

TABLE OF CONTENTS

SECTION		PAGE
I	Introduction	1
II	Test Materials and Specimens	1
III	Testing Procedures	4
IV	Analysis Procedure	4
V	Results and Discussion	5
VI	Summary and Conclusions	5
	Bibliography	6
	Appendix I - Studies of Differences Between 1954 Tests and 1955 Tests	7
	I.1 - Comparison Tests at Finite Life	7
	I.2 - Comparison Tests at Long Life	7
	I.3 - Possible "Assignable Causes" for Difference Between 1954 and 1955 Tests	7

LIST OF TABLES

TABLE		PAGE
1	Comparison of Combined Tests with 1954 and 1955 Tests in the Long Life Region	11
2	Ninety-Nine Percent Probabilities of Survival	11
3	R.R. Moore Rotating Beam Fatigue Tests, at 11,000 RPM, of Smooth Specimens of SAE 4340 Steel, of Nominal 140 ksi UTS . . .	12
4	R.R. Moore Rotating Beam Fatigue Tests, at 11,000 RPM, of Smooth Specimens of SAE 4340 Steel, of Nominal 190 ksi UTS . . .	19
5	R.R. Moore Rotating Beam Fatigue Tests, at 11,000 RPM, of Smooth Specimens of SAE 4340 Steel, of Nominal 260 ksi UTS . . .	25
6	Comparison of 1955 Tests with 1954 Tests in the Finite Life Region	30
7	Effect of Oiling and Cleaning Surface of Specimens	31

LIST OF ILLUSTRATIONS

FIGURE		PAGE
1	Probabilities of Surviving 10^7 Cycles of Alternating Stress . .	2
2	SAE 4340 Steel. Lines of 50, 90 and 99% Survival VS. Ultimate Tensile Stress	3

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SECTION I. INTRODUCTION

The general purpose of these investigations is set forth in the contract as follows: ". . . to conduct a test program planned and analysed using statistical methods for the purpose of furnishing designers with the fatigue characteristics of SAE 4340 steel and which will serve as a guide for obtaining the fatigue characteristics of other propeller and rotor blade materials" (1)^{1/}. In reference (1) probabilities of 10, 50 and 90% survival of fatigue stress at constant life were determined. The probability that 90% of the specimens of a structural material will survive a given stress for, say, ten million cycles is not adequate information for the design engineer. What he would like to know is the stress that 100% of the specimens will probably survive. Even the powerful methods of modern statistical analysis will not give probabilities much beyond the ranges covered by a set of test data, unless the law of distribution of probabilities is known, and the only way to establish this law of distribution is to test very large numbers of specimens.

In the case of the SAE 4340 steel being investigated it has been established that in the long-life region, for the range from 10 to 90%, strength at constant life can be assumed to be either normally or log-normally distributed whichever is more convenient. This means that it is not known certainly whether either distribution is exact. And when probabilities of survival higher than 90% are sought small deviations from perfect normal distribution can change the survival stress very appreciably. In the terminology of the statistician, seeking for high-survival probabilities means looking for the fatigue strength of the weaker specimens in the uncertain thin width of the lower tail of the distribution of fatigue strengths.

The tests reported in this supplemental report were made for the purpose of obtaining probabilities of survival of stress at constant life, as much higher than 90% as time and the number of specimens available would permit. For the lowest stress at which tests were made a hundred specimens were used. This gave information regarding the probabilities of survival of the order of about 99%, based on the data reported herein combined with those reported in reference (1).

The specimens tested were from the same heat of steel as those used for reference (1), and the fabrication, testing, and analysing techniques were the same. However, wherever comparisons can be made, consistent and persistent small but significant differences appear between the results of the tests reported herein (which will be referred to as the "1955 tests"), and those reported in reference (1) (which will be referred to as the "1954 tests"). Possible explanations of these differences were investigated (see Appendix I) and it was finally concluded that differences in inclusion size and distribution in different parts of an ingot were responsible. In order to get more representative values of the mean and the high probability strength of the heat of steel, the 1954 tests were combined with the 1955 tests in the probit analyses from which the values shown in Tables 1 and 2 were obtained. For comparison, these tables show also the values as computed from the 1955 tests alone and from the 1954 tests alone. Test data for the 1955 tests are given in Tables 3, 4 and 5. Final results are given in Tables 1 and 2, and on Figs. 1 and 2.

SECTION II. TEST MATERIALS AND SPECIMENS

The SAE 4340 steel used for the tests reported herein was from the same heat (Republic Steel Heat No. H15701A) from which specimens were cut for the 1954 tests. The specimens were numbered as described in reference (1), and so distributed that each bar contributed speci-

1/ Numbers in parenthesis refer to references in the Bibliography.

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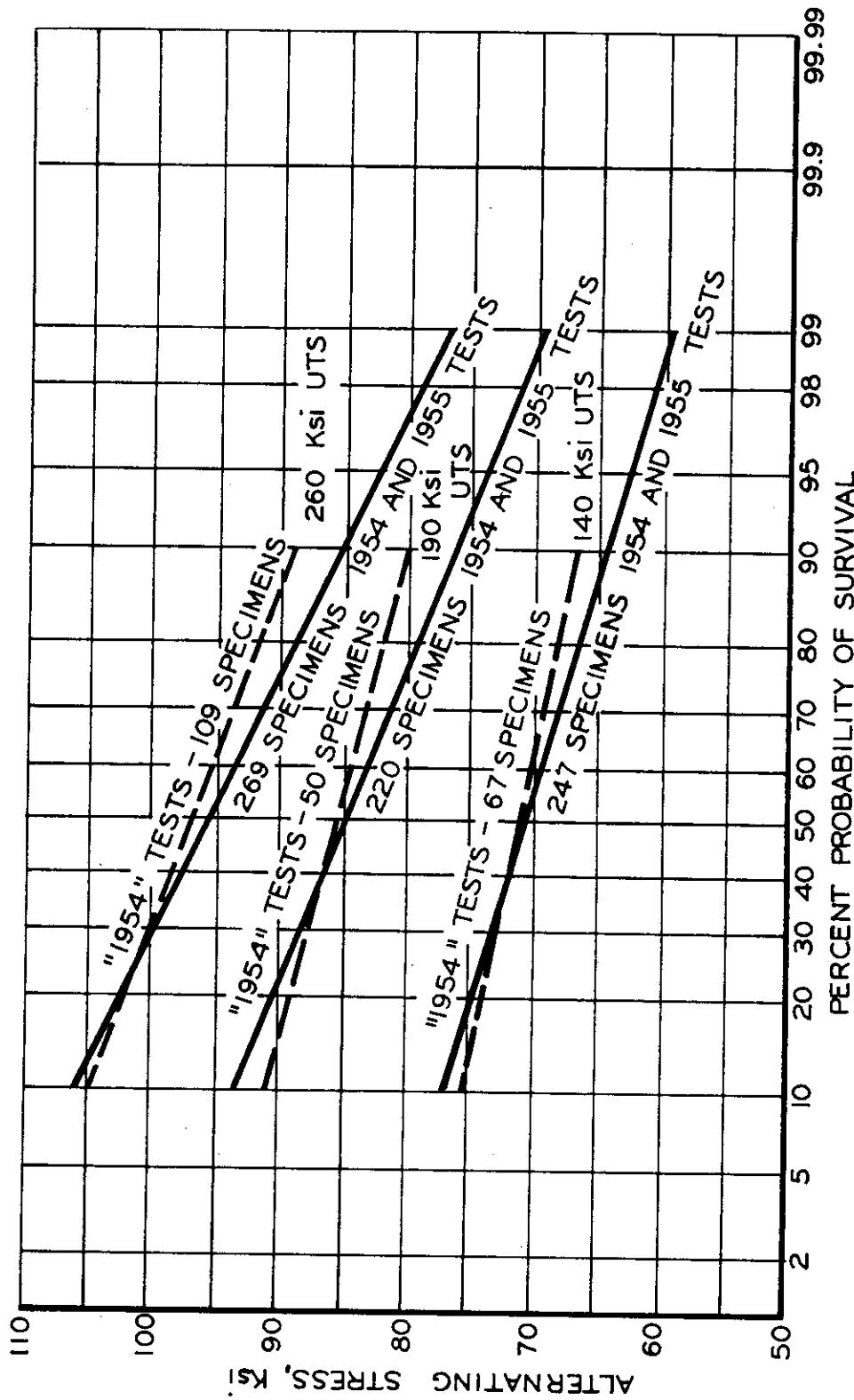


FIG. I

PROBABILITIES OF SURVIVING 107 CYCLES OF ALTERNATING STRESS
SMOOTH R.R. MOORE ROTATING BEAM
SPECIMENS OF SAE 4340 STEEL TESTED AT
10,000 - 11,000 RPM

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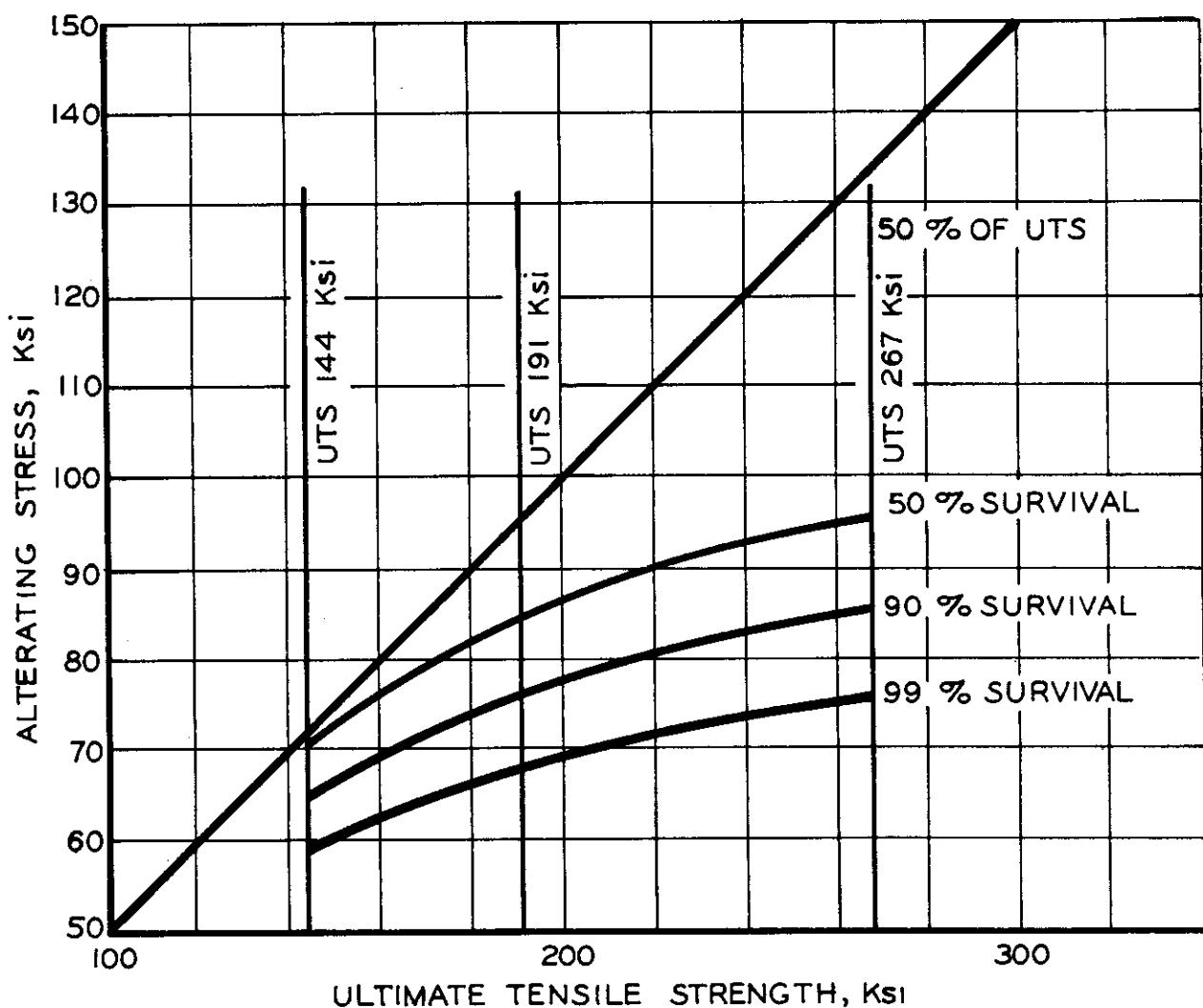


FIG. 2

(REVISION OF PART OF FIG. 30 REFERENCE I)

SAE 4340 STEEL, LINES OF 50, 90 AND 99 %

SURVIVAL OF STRESS AT 10^7 CYCLES

R.R. MOORE ROTATING BEAM TESTS OF SMOOTH SPECIMENS

AT 10,000 - 11,000 RPM VS. ULTIMATE TENSILE STRENGTH

Controls

mens to each of the three hardness levels, 140, 190, and 260 ksi, UTS, nominal. The fabrication, heat treatment, and stress relief procedures were also as described in reference (1).

SECTION III. TESTING PROCEDURES

The testing-laboratory routine was the same as was followed during the 1954 tests. The laboratory personnel was not the same, although supervision was by the same personnel as before. A possibility existed, of course, that this might account for the persistent differences between 1954 and 1955 results, but this was eliminated later (see Appendix I).

For each of the three hardness levels, the first tests were made at a stress level corresponding to a set of 1954 tests in the finite life region, to check on the validity of the assumption that the 1955 tests could be used for simply extending the 1954 test results to lower stress levels. Since all specimens were to be from the same heat of steel used previously, it had been supposed that this would be a mere matter of routine. However, after tests of 140 ksi UTS steel at 81 ksi stress level showed a significantly lower life at constant stress than the "adjusted" life of 1954 specimens, tests were run at 77 ksi, and at 75 ksi, with the same result - lower life at constant stress (see Table 6). This caused a change from the original plan of testing only at stress levels below those reported in reference (1). Test levels were set to over-lap the lower stress levels of the 1954 tests, and a set of ten specimens was tested at each step downward until a level was reached at which the ten all survived ten million cycles. At that same level testing continued in sets of ten until one or more failures occurred. Then testing was started at the next lower step, at which time about a hundred specimens were left. No breaks occurred in testing the 140 ksi steel until after 70 specimens had been tested. Then it was decided to continue testing on that level until the supply of specimens was exhausted. The same procedure was followed in testing the 190 and the 260 ksi UTS specimens.

It will be noted in Tables 3, 4, and 5, for test sets in which one or more specimens survived 10^7 cycles, the number of "kilocycles at end of test" is given. Failures to survive 10^7 cycles are indicated by asterisks. For the other specimens the tests were stopped after running for anywhere from just over 10 million to as many as 40 or 50 million cycles. Since no time could be gained by stopping tests as soon as 10 million cycles had been run, it was decided to let the tests run as long as convenient, and to let tests started Fridays run over the weekend, to see if breaks might occur after the traditional "endurance limit" test period of 10^7 cycles. A few such breaks did occur.

SECTION IV. ANALYSIS PROCEDURE

It was stated in Appendix XII of reference (1) that the "high survival probability" data that were to be the subject of this supplemental report would be analysed by a modification, recommended by Bartlett, of the usual methods of probit analysis. The data for the tests of 140 ksi UTS were analysed by the Bartlett procedure (see ref. 2), and also by the ordinary method that was used in reference (1). It was found, as suggested by Finney (2) that since there were so many specimens tested at the low stress levels, the results of the two methods were close to each other. Furthermore, for the 190 ksi UTS tests, since there were no failures in the 100 specimens available for testing at the lowest stress the Bartlett procedure could not be used. Therefore, this report is based on analyses by the same methods used in reference (1).

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SECTION V. RESULTS AND DISCUSSION

The final results of the analysis of combined 1954 and 1955 test data are shown in Fig. 1. Mean fatigue strengths and standard deviations, for 10 million cycles of fatigue stressing are given in Table 1, and 99% probabilities of strength for 10 million cycles of fatigue strength are given in Table 2. Fig. 2 is a revision of part of Fig. 30, reference (1). It shows slightly lower mean (50%) strengths at 10 million cycles than the 1954 tests showed, a somewhat greater lowering of 90% survival strengths, and an additional line for 99% survival strengths. The gradual flattening of the survival curves, on Fig. 2, as very high survival probabilities are approached suggests that although mean fatigue strengths of this SAE 4340 steel do not increase linearly with "static" ultimate tensile strengths, possibly very high probability strengths may be taken as increasing at a constant rate with tensile strength within the range of the tests and for a short distance outside the range, that is, from say 140 to 280 ksi UTS.

On Fig. 1, the relative reliability of the graphs is indicated thus:

a. More reliable results plotted as heavy solid lines.

b. Less reliable results shown by light dash lines. Reliability depends upon both the number of specimens tested and their distribution among the stress levels. The total number of specimens for each of the lines is shown on the chart. Obviously more reliable probabilities can be obtained from large numbers of tests than from a quarter to a half as many. This gives more presumptive reliability to the solid than to the dash lines. Furthermore, whereas the specimens used in the 1954 tests were distributed about evenly along the lines, relatively large numbers of specimens were used for the lowest stress levels to determine the low ends of the heavy solid lines. The method of probit analysis takes this into account and justifies stating that not only the mean (50%) points on the heavy lines but also the low ends are pretty close to correct. Attention is called to the fact that the heavy lines, representing results obtained from combinations of the stronger 1954 steel and the less strong 1955 steel, are lower than the 1954 lines, and that they have steeper slopes. The steeper slopes indicate larger standard deviations, that is, a widening of the scatter bands so as to include the lower strengths of the 1955 specimens that resulted from many of them having larger inclusions.

Table 2 brings out strongly the danger of extrapolating from inadequate data. Tentative extrapolation from the 1954 data alone gave results that were too optimistic. However, except for the accidental order in which bars were taken from the bundle for testing, the reverse could just as well have occurred, that is, the early results could have been too pessimistic.

SECTION VI. SUMMARY AND CONCLUSIONS

Data are presented for determining high probabilities of survival of stress at constant life, for SAE 4340 steel heat treated to 140, 190, and 260 ksi UTS. These data were combined with those reported in reference (1) in order to compute fatigue stresses that there are 50, 90 and 99% probabilities the specimens can survive for 10^7 cycles. The survival stresses are shown in Fig. 2.

The average size of inclusions in single nucleus fractures was found to be significantly larger in the 1955 specimens than in the 1954 specimens. This was taken to indicate

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variation within an ingot of steel, and to account for the lower values of life of the 1955 specimens as compared with the 1954 specimens. The 1955 tests were planned to overlap the 1954 tests so that when the two sets were combined the computed survival stresses would be more representative of the ingot than those computed from either set alone.

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APPENDIX I

STUDIES OF DIFFERENCES BETWEEN 1954 TESTS AND 1955 TESTS

I.1 Comparison Tests at Finite Life.

The first indication that the 1955 tests might not check with the 1954 tests appeared when tests of 140 ksi UTS specimens at 81 ksi stress-level were compared. The fatigue life values of the 1954 tests were reported (1) as observed (i.e., before adjustment), and also as adjusted because of stress-relief damage to the surfaces of stress relief groups 1 and 3. Statistical significance tests indicated (Table 6) no significant difference between the 1955 tests and the unadjusted 1954 tests. Similar results were found in the tests at 77 and at 75 ksi stress levels, and, later, in the tests of 190 and 260 ksi steel. This suggested the possibility that the 1955 test specimens were in general not as strong in fatigue as the 1954 specimens.

I.2 Comparison Tests at Long Life.

The fact that the values of mean strength at 10^7 cycles that are recorded in Table 1 are, in every case, a small amount lower for 1955 tests than for 1954 tests, and that this was predicted by the tests at finite life, indicates the existence of one or more "assignable causes". The differences are small, but statistically significant, and some supplementary testing and considerable supplementary analysis was required to track down what seems to be the main assignable cause. The next paragraphs describe in chronological order these investigations.

I.3 Possible "Assignable Causes" for Differences Between 1955 and 1954 Tests.

I.3.1 Oiled VS. Dry Surfaces of Specimens.

Early in the testing period a paper by Frankel and Bennett (3) became available. The authors found that "oil . . . raised the fatigue limit and increased the fatigue life . . ." of SAE 4340 steel. At about the time reference (3) became available, oiling of the 190 ksi UTS and the 260 ksi UTS specimens were being considered since hot humid weather had arrived and rusting of the surfaces was a possibility. It was immediately decided to test four sets of ten specimens each, of 190 ksi UTS steel, at 96 ksi stress level, for comparison with the corresponding set of ten in the 1954 tests. Surface treatment was as follows:

- a. Not oiled. Tested as received after stress relief.
- b. Oiled with Cosmoline 265 (Houghton Oil Co.) and tested without cleaning.
- c. Oiled with Cosmoline 265, then degreased with trichlorethylene vapor degreaser before testing.
- d. Oiled with Cosmoline 265, then wiped dry, with soft tissue paper, before testing.

Results of the tests are given in Table 4. Statistical studies of the data, as shown in Table 7, fully confirmed the findings reported in reference (3), and also indicated that there was no significant difference among treatments a, c, and d. Based on these findings, it was decided to immediately oil specimens when received after stress relief, and to wipe them dry with soft tissue paper just before testing. (The tests just described do not give any indication as to how thin the oil coating can be and still appreciably increase the fatigue life of a specimen of this steel. This subject requires further investigation.)

Upon questioning the laboratory technician who did the 1954 testing it was learned that although the specimens were not oiled, they were wiped with soft tissue paper before testing, to remove any grease that might have gotten onto their surfaces. This put the 1954 speci-

Controls

mens in the same category as the 1955 specimens that had treatments a, c, or d as listed above, and appeared to eliminate oiled or greasy surfaces as the reason for higher values in the 1954 tests.

I.3.2 Variations in Heat Treatments.

It was thought that possibly the temperature controls on the heat treatment furnaces might have varied so that small differences in hardness and in corresponding tensile strength might have occurred. Tensile strength and fatigue strength depend upon hardness, therefore tests of hardness were made on samples chosen at random from specimens tested in 1954 and in 1955. The following values of Rockwell C-scale hardness were found - each number being the average of three tests on a specimen. Hardness readings were taken to one decimal place and averages computed to two places.

Rockwell C Hardness Tests on Specimens of 260 ksi UTS

<u>1954 Specimens</u>	<u>1955 Specimens</u>
52.10	52.03
52.07	52.00
51.93	51.97
51.93	51.83
51.87	51.77
51.63	51.77
51.63	51.70
51.57	51.70
51.50	51.60
51.43	51.03
Mean, 51.77	Mean, 51.74

There is no indication in this brief study that heat treatments were out of control.

I.3.3 Eccentricity of Specimens in R. R. Moore Machines.

The laboratory reports on the 1955 tests indicated that there was upwards of twice as much eccentricity in the mounting of the specimens as there was in the 1954 tests. This might be caused by (a) less perfect machining of the specimens, (b) more wear in the collets and bearings of the machines, or (c) less care by the laboratory technician in placing specimens in the machines.

a. Since the general testing of both 1954 and 1955 specimens had been completed there were no specimens in "as received" condition for comparison as to machining imperfections. There was no reason to believe the machining vendor had lowered his standards, but the possibility remained until an opinion could be reached from other tests.

b. Although it had been concluded, as stated in reference (1), that eccentricities of the magnitude measured in the early tests did not correlate with fatigue life, records have been kept of the static eccentricity of practically all specimens tested. An analysis of these records gave the following results, which have a definite significance as to the cause of increased 1955 eccentricity, since the data below are listed in chronological order of testing.

Contrails

Hardness Level ksi	Mid Date of Testing	No. of Specimens	Mean Eccentricity 10 ⁻³ inch	Increase in Eccentricity 10 ⁻³ inch	Stand. Dev. 10 ⁻³ in.	Increase in St. Dev. 10 ⁻³ inch
140	3-31-54	324	0.775	0.140	0.233	0.002
190	4- 8-54	315	0.915	0.350	0.235	0.057
260	9- 7-54	312	1.265	0.320	0.292	0.086
140	5-24-55	196	1.585	0.270	0.378	0.052
190	6-16-55	209	1.855	0.300	0.430	0.030
260	7-18-55	169	2.155		0.460	

From these data it is concluded that the consistency of the increase in eccentricity, as testing continued, indicates a continuous wear in equipment, which made low eccentricity more and more impossible to obtain. This seemed to indicate that the change in laboratory personnel was not responsible for the increased eccentricity. It further indicated that if the 1955 technician were given new spindle assemblies for the R. R. Moore machines he could match the close settings of the earliest tests, or perhaps better them, and produce a set of tests that could be used to compare with the 1954 tests and/or with his 1955 tests to study the effect of eccentricity on fatigue strength.

c. Although lack of careful work by the laboratory technician seemed to have been eliminated as in any way responsible for the lower values obtained in the 1955 tests, there was no longer any reason to doubt his workmanship when tests were made using new spindle assemblies. At no time did his first assembling of specimen and housings give an eccentricity of over 0.00075 inches, and two or three or even four re-assembling operations failed to noticeably improve the reading of the dial (see Fig. 54, p. 80, ref. 1). Two sets of tests were run, with the following results, shown in comparison with earlier 1955 tests.

UTS ksi	Stress ksi	No. of Specimens	Surface Treatment	Mean Eccentricity	Mean Life/ 100 Kc
190	96	10 5	Oiled & Degreased " "	0.00205 in. 0.00035	105 100
260	116	10 6	Oiled & Wiped " & Degreased	0.00205 in. 0.00040	66 57

Statistical studies based on F-ratio and on t-tests indicate no significant difference between the results shown above obtained with large and with low eccentricities. It was concluded that wear in the testing equipment had not progressed far enough to account for the lower results obtained in the 1955 tests, and that there had been no lowering of machining standards.

I.3.4 Variation Within An Ingot.

Along with the lower fatigue strength values reported in the 1955 tests, larger inclusions were frequently reported. When the 1955 testing was finished, a general study of inclusion sizes was made. It was felt that a comparison of the sizes of all the single-nucleus inclusions found in all the fractured specimens tested in 1955 with those tested in 1954 might indicate a difference in inclusion size and/or distribution. The results of this study were as follows:

	No. of Specimens	Mean of Inclusion Sizes	Standard Deviation
1954 tests	208	0.00132 in.	0.00046 in.
1955 tests	125	0.00173 in.	0.00054 in.

1/ Life corresponding to mean log-life.

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Statistical analysis, using F-ratio and t-tests indicate that the scatter is somewhat significant and that the difference in mean sizes is very significant. This leads to the conclusion that the specimens tested in 1955 were more vulnerable to fatigue stressing, at least in the long-life (generally single nucleus) region than were the specimens tested in 1954. The steel bars were received from the steel company in bundles and the bundles were stored in racks. So far only one bundle has been opened. Bars for 1954 tests were taken from the top of that bundle and bars for 1955 tests were taken from about the middle of the same bundle. On the assumption that each bundle came from an ingot, the difference between the inclusion contents of the two sets of bars brings out a variation of fatigue strength within an ingot.

TABLE 1

Comparison of Combined Tests with 1954 and with 1955 Tests
 in the Long Life Region, Based on Probit Analysis
 of Strength at N = 10⁷ Cycles
 R. R. Moore Rotating Beam Tests of Smooth Specimens of
 SAE 4340 Steel, all from the Same Heat

	140 ksi UTS	190 ksi UTS	260 ksi UTS
	Mean (50%) Strength	Standard Deviation	Mean (50%) Strength
Combined Tests	69.0 ksi	4.4 ksi	84.5 ksi
1954 Tests	71.0 ksi	3.5 ksi	85.5 ksi
1955 Tests	67.4	4.0	84.0

TABLE 2

Ninety-nine Percent Probabilities of Survival
 of Stress (Strength) for 10⁷ Cycles, as Determined from
 Combined 1954 and 1955 Tests, and from 1954 Tests Alone, of SAE 4340 Steel.
 Smooth Specimens. R. R. Moore Rotating Beam Tests.
 Speed of Testing, 10,000 - 11,000 RPM.

	140 ksi UTS	190 ksi UTS	260 ksi UTS
Combined Tests	59 ksi	69 ksi	76 ksi
1954 Tests, ksi	63 *	75 *	82 *

* These 1954 values were extrapolated from inadequate data.
 This matter is discussed in TR 54-531.

TABLE 3

R. R. Moore Rotating Beam Fatigue Tests, at 11,000 RPM, of Smooth Specimens
of SAE 4340 Steel, of Nominal 110 ksi UTS

NOTE: Specimens not oiled previous to testing.

81 ksi Stress Level, Nominal

Specimen No.	FHB103	FLBK18	FPBL05	DUBD13	DYBF34	FGBH18	FDBH20
Diam., Inches	0.2310	0.2305	0.2305	0.2308	0.2308	0.2303	0.2298
Machine No.	10	13	10	13	10	13	10
Kilocycles to Fracture	113	150	111	102	84	120	116
Dist., Ctr. to Fracture **	-0.02	-0.09	-0.12	0	+0.02	-0.03 $\frac{1}{2}$	-0.07 $\frac{1}{2}$

(Reproduced from WADC TR 54-531) 81 ksi Stress Level, Nominal

Specimen No.	ABAAL4	AKAC11	ANAD01	BMAN31	BPAA028	BUAP05	BXAR35	CBA506	CFAT02
Diam., Inches	0.2298	0.2300	0.2301	0.2306	0.2310	0.2305	0.2310	0.2300	0.2305
Machine No.	19	20	21	22	23	24	19	20	21
Kilocycles to Fracture *	163	120	124	125	228	293	112	123	98
Dist., Ctr. to Fracture **	+0.02 $\frac{1}{2}$	+0.02 $\frac{1}{2}$	-0.05	-0.14 $\frac{1}{2}$	+0.01 $\frac{1}{2}$	+0.02 $\frac{1}{2}$	+0.02 $\frac{1}{2}$	-0.01 $\frac{1}{2}$	+0.05 $\frac{1}{2}$
Stress Relief Group	1	1	1	1	2	2	2	3	3
Adjusted Kilocycles	260	188	195	198	228	293	112	195	186

77 ksi Stress Level, Nominal

Specimen No.	FBBH01	FFBI22	FLBK06	FOBL30	DUBD04	FGBH09	FHB130	FMBL14	FNBH01
Diam., Inches	0.2300	0.2310	0.2305	0.2305	0.2310	0.2305	0.2310	0.2310	0.2305
Machine No.	10	13	19	20	21	22	23	24	25
Kilocycles to Fracture	152	281	301	191	216	132	180	186	140
Dist., Ctr. to Fracture **	+0.04	-0.14	+0.08 $\frac{1}{2}$	+0.09 $\frac{1}{2}$	+0.02	+0.07 $\frac{1}{2}$	-0.03	+0.03 $\frac{1}{2}$	+0.04

* Before adjustment.

** Minus sign indicates distance toward motor end of specimen. Distance in inches.

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TABLE 3
R. R. Moore Rotating Beam Fatigue Tests, at 11,000 RPM, of Smooth Specimens
of SAE U340 Steel, of Nominal 110 ksi UTS

NOTE: Specimens not oiled previous to testing.

(Reproduced from WADC TR 54-531) 77 ksi Stress Level, Nominal

Specimen No. Diam., Inches	AACAA34 0.2295 23 198 -0.14 1 373	AHAB36 0.2300 21 136 -0.03½ 1 251	AMAD01 0.2300 25 168 -0.06 1 314	ATAC05 0.2296 26 732 -0.23½ 1 1480	APAF09 0.2295 19 104 +0.08½ 2 104	AWAH33 0.2300 20 234 -0.11½ 2 234	AXAT11 0.2298 21 315 -0.02 2 315	BAAK10 0.2300 22 449 -0.11½ 2 449	BDAL20 0.2305 23 549 +0.13½ 2 549	BIAM16 0.2300 19 419 -0.07½ 2 419	
Machine No.											
Kilocycles to Fracture*											
Dist., Ctr. to Fracture **											
Stress Relief Group											
Adjusted Kilocycles											

(Reproduced from WADC TR 54-531) 77 ksi Stress Level, Nominal (Continued)

Specimen No. Diam., Inches	BMAN25 0.2295 20 317 +0.03½ 2 317	BPA022 0.2302 21 330 +0.03½ 2 330	AXAT17 0.2290 22 313 +0.03 2 313	CHAT36 0.2295 23 313 -0.04½ 3 321	ATAH03 0.2296 19 390 +0.13 2 390	BTAP33 0.2310 20 172 +0.13 3 321	BXAR29 0.2302 21 155 +0.04 2 390	CAAS37 0.2305 22 229 -0.07 3 288	CDAT34 0.2305 23 134 +0.02 3 425		
Machine No.											
Kilocycles to Fracture*											
Dist., Ctr. to Fracture **											
Stress Relief Group											
Adjusted Kilocycles											

75 ksi Stress Level, Nominal

Specimen No. Diam., Inches	DUBD22 0.2308 19 292 -0.06	DIBF07 0.2307 23 255 +0.11	FBBH19 0.2297 19 203 +0.05	FBBL13 0.2305 23 171 +0.04	FBBK28 0.2302 19 298 +0.01½	FNBL25 0.2298 23 151 +0.00½	DWB30 0.2302 19 221 +0.01½	FAEF05 0.2302 23 176 +0.09	FCBH03 0.2307 19 185 -0.03½	FHBL18 0.2306 23 158 +0.02	
Machine No.											
Kilocycles to Fracture*											
Dist., Ctr. to Fracture **											

* Before adjustment.

** Minus sign indicates distance toward motor end of specimen. Distances in inches.

Controls

R. R. Moore Rotating Beam Fatigue Tests, at 11,000 RPM, of Smooth Specimens
of SAE 4340 Steel, of Nominal 140 ksi UTS

TABLE 3

NOTE: Specimens not oiled previous to testing.

(Reproduced from WADC TR 54-531) 75 ksi Stress Level, Nominal

Specimen No.	ABAA34	AFAB36	ANAD31	BRA021	BUAP35	BYAR25	OBAS36	CPAT32
Diam., Inches	0.2295	0.2295	0.2305	0.2275	0.2300	0.2300	0.2300	0.2305
Machine No.	23	24	25	26	19	20	21	22
Kilcycles to Fracture *	275	299	228	392	383	208	319	188
Dist., Ctr. to Fracture **	+0.08 $\frac{1}{2}$	-0.06 $\frac{1}{2}$	+0.06 $\frac{1}{2}$	+0.12	+0.00 $\frac{1}{2}$	+0.06 $\frac{1}{2}$	-0.11	-0.08 $\frac{1}{2}$
Stress Relief Group	1	1	1	1	2	2	3	3
Adjusted Kilcycles	302	330	251	432	383	208	247	352

* Before Adjustment

** Minus sign indicates distance toward motor end of specimen. Distances in inches.

Specimen No.	FABF35	FDBH23	FHB106	FBBLQ0	FMBL34	DWBD36	DZBF03	FCBH24	FIB108	FKBK19
Diam., Inches	0.2299	0.2305	0.2304	0.2306	0.2307	0.2308	0.2307	0.2307	0.2307	0.2305
Machine No.	10	16	18	19	20	21	22	23	25	26
Kilcycles to End of Test	195*	241*	571*	14,232	184*	261*	474*	251*	270*	275*
Dist., Ctr. to Fracture **	-0.09	+0.07 $\frac{1}{2}$	-0.13	-0.07	-0.07	-0.16	+0.08	-0.04	-0.11	+0.05

Specimen No.	FPBL32	DDDB02	DIBFL9	FBBH28	FFBFI07	FLBK21	FOBL03	DUBD16	DZBZF06	FDBH35
Diam., Inches	0.2305	0.2306	0.2302	0.2305	0.2299	0.2305	0.2306	0.2304	0.2309	0.2310
Machine No.	24	16	18	19	20	21	22	23	25	26
Kilcycles to End of Test	243*	359*	744*	197*	11,426	11,989	511*	232*	232*	238*
Dist., Ctr. to Fracture **	+0.04 $\frac{1}{2}$	+0.10 $\frac{1}{2}$	-0.03	-0.03 $\frac{1}{2}$	-0.03	-0.06	+0.06	-0.05	-0.05	+0.06

* FAILED

** Minus sign indicates distance toward motor end of specimen. Distances in inches.

TABLE 3

Page 4 of 7

R. R. Moore Rotating Beam Fatigue Tests, at 11,000 RPM, of Smooth Specimens
of SAE 1340 Steel, of Nominal 110 ksi UTS

NOTE: Specimens not oiled previous to testing.

69 ksi Stress Level, Nominal

Specimen No. Diam., Inches	FFBK20 0.2306	FFBK22 0.2307	FNBL31 0.2305	DXBDO5 0.2305	FABF20 0.2308	FBH32 0.2305	FHB124 0.2303	FKBK34 0.2298	FOBL06 0.2302	FABF23 0.2298
Machine No.	16	18	19	20	21	22	23	24	25	26
Kilcycles to End of Test	12,035	230*	326*	397*	182*	429*	513*	11,710	286*	-0.04
Dist., Ctr. to Fracture **	-0.07	-0.13	+0.06	1/2	0	-0.05	-0.06			

67 ksi Stress Level, Nominal

Specimen No. Diam., Inches	DUBD19 0.2310	DYBF04 0.2305	FIBI27 0.2311	FLBK31 0.2303	FLBK36 0.2308	FOBL15 0.2302	FDBL26 0.2306	FIBI23 0.2308	FABF32 0.2308	DXBDD29 0.2308
Machine No.	16	18	19	20	21	22	23	24	25	26
Kilcycles to End of Test	12,436	12,243	12,753	13,429	757*	1,179*	12,250	12,889	513*	265*
Dist., Ctr. to Fracture **				-0.09	+0.24			+0.09	+0.09	+0.07

65 ksi Stress Level, Nominal

Specimen No. Diam., Inches	DWBD33 0.2302	FABF02 0.2301	FIBI26 0.2298	FLBK03 0.2298	FOBL18 0.2305	FPBL29 0.2297	FABF26 0.2300	DXBDO8 0.2302	FHB121 0.2300	FBBH31 0.2308
Machine No.	16	18	19	20	21	22	23	24	25	26
Kilcycles to End of Test	62,220	62,480	68,440	810*	62,673	56,301	63,434	104*	57,324	63,495
Dist., Ctr. to Fracture **			+0.11				+0.02			

63 ksi Stress Level, Nominal

Specimen No. Diam., Inches	DUBD01 0.2305	DYBF22 0.2305	FGBH06 0.2310	FBH127 0.2310	FBK11 0.2305	FPBL35 0.2305	DWBDO3 0.2310	DZBFD24 0.2300	FDBH08 0.2305	FIBI29 0.2305
Machine No.	10	13	19	20	21	22	23	24	25	26
Kilcycles to End of Test	12,856	12,822	12,514	12,681	11,604	11,959	11,818	12,035	700*	12,325
Dist., Ctr. to Fracture **								+0.002		

* FAILED

** Minus sign indicates distance toward motor end of specimen. Distances in inches.

TABLE 3

Page 5 of 7

R. R. Moore Rotating Beam Fatigue Tests, at 11,000 RPM, of Smooth Specimens
of SAE 1340 Steel, of Nominal 140 ksi UTS

NOTE: Specimens not oiled previous to testing.

63 ksi Stress Level, Nominal (Continued)

Specimen No.	DZBF27	FDBH11	FIBI32	FMBK29	FPBL08	DYBD20	DYBF01	FBBH16	FFBI04	DUBD3L
Diam., Inches	0.2310	0.2305	0.2310	0.2305	0.2306	0.2305	0.2301	0.2301	0.2307	0.2302
Machine No.	10	13	19	16	18	19	20	21	22	23
Kilocycles to End of Test	17,216	2,612*	16,527	11,554	11,246	12,875	12,630	5,953*	11,688	11,644
Dist., Ctr. to Fracture **	-0.02							+0.09		

61 ksi Stress Level, Nominal

Specimen No.	DZBF28	FCCBH12	FHB133	FMBK17	FNBL04	DWBD09	DZBF30	FDBH14	FIBI35	FBKB04
Diam., Inches	0.2305	0.2310	0.2300	0.2300	0.2305	0.2300	0.2305	0.2307	0.2308	0.2308
Machine No.	20	21	22	23	24	25	26	10	13	19
Kilocycles to End of Test	20,181	20,311	15,493	15,349	16,017	14,463	15,877	17,392	17,567	17,405
Dist., Ctr. to Fracture										

61 ksi Stress Level, Nominal (Continued)

Specimen No.	FBBK01	FBBH04	FFBI25	FLBK09	FOBL09	DUBD07	DYBF25	FMBK20	FNBL07	FBBH07
Diam., Inches	0.2300	0.2300	0.2315	0.2308	0.2310	0.2310	0.2311	0.2302	0.2307	0.2309
Machine No.	22	23	24	25	26	16	18	10	13	16
Kilocycles to End of Test	15,496	15,760	15,969	14,807	16,236	17,564	16,743	52,297	50,291	52,977
Dist., Ctr. to Fracture										

61 ksi Stress Level, Nominal (Continued)

Specimen No.	FLBK15	FOBL15	DUBD10	DYBF31	FCBBL5	DYBD35	FHB136	FDDBL17	FLBK12
Diam., Inches	0.2307	0.2305	0.2305	0.2310	0.2300	0.2307	0.2306	0.2306	0.2302
Machine No.	18	19	20	21	22	23	24	25	10
Kilocycles to End of Test	49,449	49,947	51,493	47,730	46,645	48,903	48,782	46,044	49,469
Dist., Ctr. to Fracture									

* FAILED

** Minus sign indicates distance toward motor end of specimen. Distances in inches.

TABLE 2

R. R. Moore Rotating Beam Fatigue Tests, at 11,000 RPM, of Smooth Specimens
of SAE 4340 Steel, of Nominal 140 ksi UTS

NOTE: Specimens not oiled previous to testing.

61 ksi Stress Level, Nominal (Continued)

Specimen No. Diam., Inches	FMBK23 0.2302	FNBL10 0.2307	DWBD12 0.2310	DZBF33 0.2302	FTB131 0.2303	FBBH10 0.2300	FBBL02 0.2302	FBBI34 0.2302	DXBD23 0.2310	FABFL4 0.2300
Machine No.	13	16	18	19	20	21	22	23	24	25
Kilocycles to End of Test	16,001	16,643	16,379	16,084	16,334	14,521	15,128	15,374	15,533	14,563
Dist., Ctr. to Fracture **										

61 ksi Stress Level, Nominal (Continued)

Specimen No. Diam., Inches	FBBH13 0.2307	FMBK26 0.2305	FNBL13 0.2308	DWBD15 0.2309	DZBF36 0.2310	FTB101 0.2305	FBBK07 0.2297	FNBL07 0.2300	DXBD32 0.2299
Machine No.	26	10	13	16	18	19	20	22	23
Kilocycles to End of Test	15,472	46,699	47,024	47,482	44,753	46,585	45,153	45,854	42,883
Dist., Ctr. to Fracture **									

61 ksi Stress Level, Nominal (Continued)

Specimen No. Diam., Inches	FABF17 0.2300	FCBH21 0.2298	FIBI05 0.2305	FBBL3 0.2306	FNBL28 0.2295	DWBD27 0.2305	DZBF09 0.2307	FBBH34 0.2302	FBBI15 0.2299	FLBK24 0.2295
Machine No.	24	Not Recorded	45,002	46,729	10,620	18	19	20	22	23
Kilocycles to End of Test					10,130	12,236	12,010	10,887	10,722	11,154
Dist., Ctr. to Fracture **										

61 ksi Stress Level, Nominal (Continued)

Specimen No. Diam., Inches	FCBL27 0.2309	DWBD26 0.2308	DZBF12 0.2305	FTBBL2 0.2302	FIBIL4 0.2310	FBBL09 0.2307	DUBD31 0.2303	DYBF16 0.2305	FCBH36 0.2310
Machine No.	24	25	26	16	18	19	21	22	23
Kilocycles to End of Test	11,278	11,057	11,256	12,175	11,448	13,160	12,895	12,113	11,709
Dist., Ctr. to Fracture **									

** Minus sign indicates distance toward motor end of specimen. Distances in inches.

TABLE 3

R. R. Moore Rotating Beam Fatigue Tests, at 11,000 RPM, of Smooth Specimens
of SAE 4340 Steel, of Nominal 110 ksi UTS

NOTE: Specimens not oiled previous to testing.

61 ksi Stress Level, Nominal (Continued)

Specimen No. Diam., Inches Machine No. Kilocycles to End of Test Dist., Ctr. to Fracture **	FFBI16 0.2297 24	FKBK31 0.2302 25	FOBL12 0.2305 26	FMBK35 0.2300 16	FPBL11 0.2308 18	DUBD25 0.2303 19	DZBFL8 0.2293 20	FCBH30 0.2308 21	FMBK05 0.2304 22	FPBL17 0.2299 23
	12,463	11,280	296*	12,225	10,945	13,354	1,120**	11,376	11,840	12,085
			-0.10				-0.13			

61 ksi Stress Level, Nominal (Continued)

Specimen No. Diam., Inches Machine No. Kilocycles to End of Test Dist., Ctr. to Fracture **	DWBD21 0.2303 24	DDBH26 0.2295 25	DXBD30 0.2305 26	DXBD11 0.2309 16	FABF29 0.2307 18	FHBI09 0.2302 19	FLBK33 0.2301 29	FPBL23 0.2299 21	DWBBD18 0.2304 22	DZBFL15 0.2304 23
	13,191	928*	12,845	14,998	13,798	15,527	15,573	14,065	14,562	14,447
		0.00								

61 ksi Stress Level, Nominal (Concluded)

Specimen No. Diam., Inches Machine No. Kilocycles to End of Test Dist., Ctr. to Fracture **	FBBH22 0.2302 24	FIBI17 0.2310 25	DWBD06 0.2307 26	FNBL19 0.2302 24	DXBD14 0.2307 25	SYBF10 0.2302 26	FBBH25 0.2309 16	FBBI19 0.2309 18	FMBK02 0.2308 19	FOBL21 0.2308 20
	15,463	13,443	15,243	12,454	2,334*	12,025	11,796	11,538	12,855	12,343
					-0.03					

* FAILED

** Minus sign indicates distance toward motor end of specimen. Distances in inches.

TABLE 4

R. R. Moore Rotating Beam Fatigue Tests, at 11,000 RPM, of Smooth Specimens
of SAE 1340 Steel, of Nominal 190 ksi UTS

NOTE: All specimens oiled, and dried with Kleenex before testing, unless otherwise noted.

96 ksi Stress Level, Nominal (Not Oiled)

Specimen No. Diam., Inches	DULC02 0.2309	FCLF23 0.2305	FHLH28 0.2299	FMLJ12 0.2306	FPLX36 0.2301	DMLQ14 0.2300	DZLDD25 0.2305	FDLF09 0.2299	FILH30 0.2303
Machine No.	19	23	0.2304	0.2303	0.2301	0.2300	0.2305	0.2299	0.2303
Kilocycles to Fracture	96	90	19	19	19	19	23	19	23
Dist., Ctr. to Fracture **	-0.05 $\frac{1}{2}$	-0.04 $\frac{1}{2}$	0	0	+0.12	-0.06	+0.12	-0.15	-0.12

96 ksi Stress Level, Nominal

Specimen No. Diam., Inches	DXLG12 0.2305	FDLF30 0.2300	FHLH10 0.2302	FHLI34 0.2303	FPLK24 0.2302	DMLQ19 0.2308	DZLDD16 0.2302	FBLF23 0.2307	FILH18 0.2307
Machine No.	19	23	19	23	23	19	23	19	23
Kilocycles to Fracture	114	110	91	117	135	101	240	19	122
Dist., Ctr. to Fracture **	-0.05	-0.02	+0.04	+0.14	0	-0.13	-0.01 $\frac{1}{2}$	-0.03 $\frac{1}{2}$	+0.07 $\frac{1}{2}$

96 ksi Stress Level, Nominal - oiled, then degreased.

Specimen No. Diam., Inches	DYLK20 0.2308	DYLC15 0.2307	DYLD11 0.2303	FFLF26 0.2300	FMLI20 0.2302	FMLI03 0.2309	FOLK22 0.2305	DZLDD22 0.2305	FILF30 0.2305
Machine No.	19	23	19	23	19	23	19	23	23
Kilocycles to Fracture	114	86	101	94	134	103	129	115	101
Dist., Ctr. to Fracture **	+0.04 $\frac{1}{2}$	-0.06	+0.01	+0.01 $\frac{1}{2}$	-0.12	-0.01 $\frac{1}{2}$	-0.04	-0.17	-0.04

96 ksi Stress Level, Nominal - oiled, then tested while oiled.

Specimen No. Diam., Inches	FMLI36 0.2306	FPLK12 0.2306	DULC26 0.2305	DZLDD19 0.2307	FOLK31 0.2305	FMLI06 0.2301	FBLK18 0.2303	DWLC22 0.2308	FILF27 0.2303
Machine No.	19	23	19	23	19	23	19	23	23
Kilocycles to Fracture	231	220	475	1031	346	166	231	388	23
Dist., Ctr. to Fracture **	-0.13 $\frac{1}{2}$	+0.01	-0.06	+0.05	+0.01	+0.12	+0.07	-0.06	+0.05

** Minus sign indicates distance toward motor end of specimen. Distance in inches.

TABLE 4

R. R. Moore Rotating Beam Fatigue Tests, at 11,000 RPM, of Smooth Specimens
of SAE 4340 Steel, of Nominal 190 ksi UTS

NOTE: All specimens oiled, and dried with Kleenex before testing, unless otherwise noted.

(Reproduced from WADC TN 54-531)

Specimen No.	A SKF27	AWKH25	AZK136	CCKS15	CHKT22	ACKA11	AHKB10	AIKC21	AOKD20	BKMK05
Diam., Inches	0.2310	0.2310	0.2305	0.2295	0.2301	0.2306	0.2298	0.2305	0.2300	0.2301
Machine No.	22	23	24	20	24	10	13	10	20	24
Kilocycles to Fracture	322	304	226	136	92	209	286	316	207	135
Dist., Ctr. to Fracture **	-0.17	+0.14	-0.12	-0.14	-0.02	-0.07	-0.24	+0.10	-0.03	-0.02

86 ksi Stress Level, Nominal

Specimen No.	FPLK33	DXLC03	DYLD20	FBLF29	FFLH08	FFLI22	FOLK04	DULC17	DZLD07	FDLF36
Diam., Inches	0.2309	0.2305	.0.2300	0.2309	0.2300	0.2308	0.2312	0.2304	0.2305	0.2307
Machine No.	11	13	17	20	21	22	23	24	25	26
Kilocycles to End of Test	437*	307*	494†	337*	43,439	342*	43,832	45,935	43,132	123*
Dist., Ctr. to Fracture **	+0.08	+0.02	-0.02	-0.22	+0.05					-0.08‡

92 ksi Stress Level, Nominal

Specimen No.	FALD36	FDLF24	FILH07	FKL111	FNLK35	DWLCO1	DZLD04	FCLF25	FILH09	FKL120
Diam., Inches	0.2305	0.2303	0.2306	0.2302	0.2307	0.2302	0.2307	0.2303	0.2315	0.2306
Machine No.	16	18	19	20	21	22	23	24	25	26
Kilocycles to End of Test	15,317	223*	17,315	16,943	135*	218	16,021	231	15,528	16,684
Dist., Ctr. to Fracture **	+0.03			-0.07	-0.05			-0.05	-0.08‡	

78 ksi Stress Level, Nominal - Not Oiled

Specimen No.	FBLF02	FFLH23	FLLI07	FOLK31	DULC05	FCLF10	FILH31	FMLI15	FNLK02	DWLCO7
Diam., Inches	0.2297	0.2302	0.2301	0.2297	0.2300	0.2301	0.2301	0.2300	0.2301	0.2297
Machine No.	16	18	19	20	21	22	23	24	25	26
Kilocycles to End of Test	25,608	24,710	27,009	24,283	25,546	24,334	26,894	340*‡	22,968	25,545
Dist., Ctr. to Fracture **								+0.05‡		

* FAILED

** Minus sign indicates distance toward motor end of specimen. Distance in inches.

TABLE 4

Page 3 of 6

R. R. Moore Rotating Beam Fatigue Tests, at 11,000 RPM, of Smooth Specimens
of SAE 1340 Steel, of Nominal 190 ksi UTS

NOTE: All specimens oiled, and dried with Kleenex before testing, unless otherwise noted.

74 ksi Stress Level, Nominal - Not Oiled.

Specimen No.	Dia.m., Inches	FDLF12	FILH33	FKL102	FBIF05	FPLH26	DULC14	DULC08	DYLD26
0.2303	0.2303	0.2306	0.2303	0.2303	0.2303	0.2305	0.2308	0.2308	0.2305
16	18	19	20	21	22	23	24	25	26
Kilocycles to End of Test		12,643	12,500	1,544*	11,867	12,123	12,588	12,285	11,302
Dist., Ctr. to Fracture **		371*	-0.06	-0.12					

70 ksi Stress Level, Nominal - Not Oiled

Specimen No.	Dia.m., Inches	FCLF13	FILH34	FMLK05	DWLC10	DZLD31	FILH36	FCLF105
0.2305	0.2304	0.2302	0.2306	0.2307	0.2302	0.2300	0.2305	0.2305
16	18	19	20	21	22	23	24	26
Kilocycles to End of Test		10,650	10,087	8,963*	10,493	10,806	10,137	11,559
Dist., Ctr. to Fracture **			-0.06				11,149	10,530

70 ksi Stress Level, Nominal (Continued)

Specimen No.	Dia.m., Inches	FILH35	DULC24	FALD15	FBLF14	FILH04	FPLI19	FPLK06	DULC14	DYLD35	FCLF19
0.2304	0.2308	0.2302	0.2300	0.2308	0.2308	0.2309	0.2309	0.2310	0.2302	0.2308	0.2305
16	18	19	20	21	22	23	24	25	24	25	26
Kilocycles to End of Test		44,328	45,577	45,416	43,458	43,440	43,092	43,440	44,902	22,721*	43,131
Dist., Ctr. to Fracture **									-0.07		

70 ksi Stress Level, Nominal (Concluded)

Specimen No.	Dia.m., Inches	FPLH11	FILH23	FNLK32	DXLC06	FALD21	FDLF33	FHLH25	FOLK07	FALD24
0.2307	0.2302	0.2304	0.2304	0.2307	0.2305	0.2305	0.2308	0.2310	0.2308	0.2305
16	18	19	20	21	22	23	24	25	25	26
Kilocycles to End of Test		12,006	13,749	12,885	12,185	11,437	12,352	12,902	11,968	12,103
Dist., Ctr. to Fracture **										

* FAILED

** Minus sign indicates distance toward motor end of specimen. Distance in inches.

TABLE 4

R. R. Moore Rotating Beam Fatigue Tests, at 11,000 RPM, of Smooth Specimens
of SAE 10340 Steel, of Nominal 190 ksi UTS

NOTE: All specimens oiled, and dried with Kleenex before testing, unless otherwise noted.

66 ksi Stress Level, Nominal

Specimen No. Diam., Inches Machine No. Kilocycles to End of Test Dist., Ctr. to Fracture	FNLH03 0.2303 16	FMLI27 0.2302 18	FNLK11 0.2303 19	DWLIC16 0.2307 20	DZLD01 0.2303 21	FDLF21 0.2310 22	FFLH02 0.2303 23	FNLK17 0.2309 25	DXLC33 0.2302 26
10,708	10,944	11,505	11,291	10,692	10,652	10,504	10,032	10,561	10,989

66 ksi Stress Level, Nominal (Continued)

Specimen No. Diam., Inches Machine No. Kilocycles to End of Test Dist., Ctr. to Fracture	FHLH01 0.2303 16	FMLI24 0.2306 18	FNLK11 0.2307 19	DWLIC13 0.2302 20	DZLD34 0.2308 21	FDLF18 0.2306 22	FFLH32 0.2309 23	FBLF11 0.2310 24	FLLJ16 0.2308 25
11,057	11,686	11,812	11,264	11,604	11,629	11,487	11,947	11,148	11,389

66 ksi Stress Level, Nominal (Continued)

Specimen No. Diam., Inches Machine No. Kilocycles to End of Test Dist., Ctr. to Fracture	FMLI21 0.2298 16	FNLK08 0.2300 18	FBLF08 0.2305 19	FFLH29 0.2295 20	FLLJ3 0.2303 21	FOLK01 0.2298 22	DULC11 0.2310 23	DYLD32 0.2300 24	FCLF16 0.2298 25
12,506	11,230	10,854	12,191	11,203	11,977	10,056	12,444	11,979	10,990

66 ksi Stress Level, Nominal (Continued)

Specimen No. Diam., Inches Machine No. Kilocycles to End of Test Dist., Ctr. to Fracture	FALD18 0.2312 16	FCLF22 0.2300 18	FILH06 0.2308 19	FMLI30 0.2305 20	FPLK09 0.2299 21	DULC21 0.2312 22	DYLD02 0.2298 23	FBLF17 0.2305 24	FLLH05 0.2303 25
18,972	19,403	21,513	20,931	19,317	17,459	19,681	19,277	18,856	20,561

1/ Because of power failure these 10 specimens were under static load for several hours, after about 3,000 kilocycles of stressing.

TABLE 4

R. R. Moore Rotating Beam Fatigue Tests, at 11,000 RPM, of Smooth Specimens
of SAE 1340 Steel, of Nominal 190 ksi UTS

NOTE: All specimens oiled, and dried with Kleenex before testing, unless otherwise noted.

66 ksi Stress Level, Nominal (Continued)

Specimen No.	DWLCL4	FALD03	FILH27	FLLI04	FOLK19	FPLK30	FALD27	DWLCL09	FHLH22	FBLF32
Diam., Inches	0.2302	0.2304	0.2304	0.2295	0.2306	0.2305	0.2312	0.2302	0.2303	0.2306
Machine No.	16	18	19	20	21	22	23	24	25	26
Kilocycles to End of Test	12,004	12,307	13,459	12,854	12,661	12,138	12,413	12,577	11,946	12,392
Dist., Ctr. to Fracture										

66 ksi Stress Level, Nominal (Continued)

Specimen No.	DULC20	DYLD05	FCLF28	FILH12	FLLI01	FOLK16	FPLK27	FILH24	FALD33	DULC30
Diam., Inches	0.2304	0.2310	0.2307	0.2292	0.2305	0.2305	0.2303	0.2308	0.2295	0.2299
Machine No.	16	18	19	20	21	22	23	24	25	26
Kilocycles to End of Test	14,370	14,105	15,657	15,175	14,126	14,156	14,002	14,789	13,987	15,117
Dist., Ctr. to Fracture										

66 ksi Stress Level, Nominal (Continued)

Specimen No.	DULC23	DYLD08	FBLF20	FPLH14	FKL129	FMLK26	DWLCL31	FALD06	FCLF04	FHLH19
Diam., Inches	0.2315	0.2310	0.2306	0.2300	0.2300	0.2315	0.2302	0.2308	0.2305	0.2307
Machine No.	16	18	19	20	21	22	23	24	25	26
Kilocycles to End of Test	14,150	13,984	15,014	14,936	13,359	13,635	13,625	14,369	13,518	14,436
Dist., Ctr. to Fracture										

66 ksi Stress Level, Nominal (Continued)

Specimen No.	FKL114	FNLK29	DWLCL08	DZLD10	FBLF35	FHLH16	FLLR25	FOLK28	DXLC27	DZLD13
Diam., Inches	0.2306	0.2305	0.2307	0.2295	0.2304	0.2301	0.2304	0.2305	0.2304	0.2304
Machine No.	16	18	19	20	21	22	23	24	25	26
Kilocycles to End of Test	58,642	61,651	63,662	64,196	60,982	56,276	60,514	60,496	58,582	64,672
Dist., Ctr. to Fracture										

TABLE 4

Continues

R. R. Moore Rotating Beam Fatigue Tests, at 11,000 RPM, of Smooth Specimens
of SAE 4340 Steel, of Nominal 190 ksi UTS

NOTE: All specimens oiled, and dried with Kleenex before testing, unless otherwise noted.

66 ksi Stress Level, Nominal (Continued)

Specimen No.	FDLF03	FHLH15	FMLI09	DULC10	DULC32	FCLF01	FHLH17	FHLI32	FOLK13
Diam., Inches	0.2307 16	0.2312 18	0.2295 19	0.2306 20	0.2310 21	0.2308 22	0.2309 23	0.2306 24	0.2305 25
Machine No.									0.2305 26
Kilocycles to End of Test	10,880	11,229	11,933	12,203	11,394	10,803	11,227	11,359	11,907
Dist., Ctr. to Fracture									

66 ksi Stress Level, Nominal (Concluded)

Specimen No.	FHLH13	FHLI17	FPLK21	DULC25	DULD14	FHLI26	FPLI15	FUDL12	FHLI28
Diam., Inches	0.2308 16	0.2310 18	0.2310 19	0.2303 20	0.2305 21	0.2306 22	0.2306 23	0.2302 24	0.2303 25
Machine No.									0.2305 26
Kilocycles to End of Test	10,718	11,216	11,828	11,795	11,290	10,374	11,085	11,536	10,919
Dist., Ctr. to Fracture									11,662

TABLE 5

R. R. Moore Rotating Beam Fatigue Tests, at 11,000 RPM, of Smooth Specimens
of SAE 4340 Steel, of Nominal 260 ksi UTS

NOTE: All specimens, oiled, then dried with Kleenex, before testing.

116 ksi Stress Level, Nominal

Specimen No. Diam., Inches	FRTCO3 0.2305	FRTD24 0.2306	FRTFO8 0.2305	DUTA06 0.2303	FOTC11 0.2300	FRTD32 0.2308	FRTF16 0.2307	FATH03 0.2305	DWTAO8 0.2301
Machine No.	19	23	19	19	23	19	23	19	23
Kilcycles to Fracture	54	66	83	76	55	110	57	50	23
Dist., Ctr. to Fracture **	-0.07	0	+0.01 $\frac{1}{2}$	+0.11	-0.03	+0.06	-0.18	+0.02	90 $\frac{1}{2}$
									+0.09 $\frac{1}{2}$

Reproduced from WADC TR 54-531 - 116 ksi Stress Level, Nominal

Specimen No. Diam., Inches	AASA12 0.2310	ADSBB11 0.2315	AISCB16 0.2317	AMSD15 0.2315	APSPF20 0.2320	ASTH14 0.2315	AXST128 0.2310	BASK30 0.2311	ATSC04 0.2311
Machine No.	19	21	20	19	21	19	20	19	21
Kilcycles to Fracture	128	168	205	108	331	89	132	258	21
Dist., Ctr. to Fracture **	-0.04	+0.04	+0.04	+0.11	+0.11	+0.07	0	+0.15	180
									+0.11

96 ksi Stress Level, Nominal

Specimen No. Diam., Inches	DZTB29 0.2302	FOTC13 0.2308	FRTD34 0.2310	FRTFO3 0.2310	FRTCO6 0.2297	FRTD27 0.2304	FRTF11 0.2303	FOTR35 0.2309	DUTA09 0.2305
Machine No.	16	18	19	20	21	22	23	24	25
Kilcycles to End of Test	122*	341*	4008*	664*	249**	10,013	10,859	10,972	10,541
Dist., Ctr. to Fracture **	-0.04	-0.06	-0.23	+0.01	-0.11 $\frac{1}{2}$				

91 ksi Stress Level, Nominal

Specimen No. Diam., Inches	DZTB30 0.2308	FOTC14 0.2312	FRTD35 0.2302	FRTF19 0.2308	FATH06 0.2308	DUTA11 0.2303	DZTB32 0.2305	FRTF22 0.2309	FRTD01 0.2306
Machine No.	16	18	19	20	21	22	23	24	25
Kilcycles to End of Test	145*	4280*	51,422	40,000+	200*	40,627	47,376	47,265	94*
Dist., Ctr. to Fracture **	0	+0.06			+0.06				+0.01

* FAILED

** Minus sign indicates distance toward motor end of specimens. Distance in inches.

TABLE 5

R. R. Moore Rotating Beam Fatigue Tests, at 11,000 RPM, of Smooth Specimens
of SAE 4340 Steel, of Nominal 260 ksi UTS

NOTE: All specimens oiled, then dried with Kleenex, before testing.

86 ksi Stress Level, Nominal

Specimen No.	FRTC16	FNTH09	FBTC09	FFT D30	FLTF14	FOTH02	DUTA12	DYTB33	FCTC17	DUTA01
Specimen No.	0.2305	0.2301	0.2302	0.2306	0.2303	0.2307	0.2307	0.2307	0.2307	0.2306
Diam., Inches	16	18	19	20	21	22	23	24	25	26
Machine No.	11,530	11,559	13,174	12,445	11,453	10,509	12,001	11,771	11,602	138*
Kilcycles to End of Test										-0.07
Dist., Ctr. to Fracture **										

81 ksi Stress Level, Nominal

Specimen No.	FRTD01	FNTH15	DWTA17	DZTB02	FRTC22	FRTD03	FKTF09	FNTH18	DUTA34
Specimen No.	0.2303	0.2306	0.2307	0.2308	0.2304	0.2305	0.2308	0.2305	0.2305
Diam., Inches	16	18	19	20	21	22	23	24	26
Machine No.	11,512	11,301	12,801	12,126	11,374	10,270	12,113	11,778	11,210
Kilcycles to End of Test									
Dist., Ctr. to Fracture									

81 ksi Stress Level, Nominal (Continued)

Specimen No.	FRTD02	FNTH12	DWTA14	DZTB35	FRTC19	FRTD33	FRTF17	FPTF04
Specimen No.	0.2304	0.2305	0.2307	0.2306	0.2302	0.2305	0.2308	0.2300
Diam., Inches	16	18	19	20	21	22	23	26
Machine No.	11,226	11,661	13,621	12,479	11,512	10,969	11,868	11,587
Kilcycles to End of Test								
Dist., Ctr. to Fracture								

81 ksi Stress Level, Nominal (Concluded)

Specimen No.	FATB19	FRTC23	FITD07	FMFTF31	FPTH10	DUTA22	DYTB03	FRTF18	FPTD06	DUTA36
Specimen No.	0.2305	0.2310	0.2305	0.2305	0.2303	0.2304	0.2309	0.2309	0.2310	0.2307
Diam., Inches	16	18	19	20	21	22	23	24	25	26
Machine No.	11,564	11,018	12,615	3,118*	11,560	10,606	12,463	3,424*	11,502	12,720
Kilcycles to End of Test										
Dist., Ctr. to Fracture **										

* FAILED

** Minus sign indicates distance toward motor end of specimens. Distance in inches.

Contrails
TABLE 5

R. R. Moore Rotating Beam Fatigue Tests at 11,000 RPM, of Smooth Specimens
of SAE 4340 Steel, of Nominal 260 ksi UTS

NOTE: All specimens oiled, then dried with Kleenex, before testing.

76 ksi Stress Level, Nominal (Continued)							
Specimen No. Diam., Inches Machine No. Kilocycles to End of Test Dist., Ctr. to Fracture	FATB01 0.2307 16 45,056	FATC25 0.2309 18 43,487	FHTD08 0.2305 19 48,661	FHTF12 0.2300 20 46,672	FNTH36 0.2308 21 43,231	DWTA02 0.2308 22 41,310	DZTB05 0.2298 23 44,635
Specimen No. Diam., Inches Machine No. Kilocycles to End of Test Dist., Ctr. to Fracture	FPTH34 0.2306 16 11,603	DHTA04 0.2305 18 11,389	DHTB21 0.2298 19 13,015	FHTC30 0.2306 20 12,461	FHTD09 0.2305 21 11,567	FHTF23 0.2305 22 10,776	FOTH05 0.2307 23 12,084
Specimen No. Diam., Inches Machine No. Kilocycles to End of Test Dist., Ctr. to Fracture	FHTD12 0.2305 16 10,924	FHTF24 0.2303 18 11,312	FNTH33 0.2303 19 12,426	DHTA07 0.2305 20 12,173	DHTB22 0.2306 21 11,329	FHTC34 0.2301 22 10,390	FHTD26 0.2305 23 11,006
Specimen No. Diam., Inches Machine No. Kilocycles to End of Test Dist., Ctr. to Fracture	DHTA21 0.2310 16 11,683	DHTB06 0.2310 18 11,652	FHTC29 0.2308 19 12,442	FHTD13 0.2300 20 11,997	FHTF02 0.2304 21 11,302	FOTH17 0.2309 22 10,647	FPTH28 0.2300 23 11,809
Specimen No. Diam., Inches Machine No. Kilocycles to End of Test Dist., Ctr. to Fracture	DHTA31 0.2310 16 11,683	DHTB34 0.2305 18 11,652	FHTC29 0.2308 19 12,442	FHTD25 0.2297 24 11,813	FHTF02 0.2300 23 11,270	FOTH08 0.2295 25 10,913	FATH25 0.2306 26 11,056

Controls

R. R. Moore Rotating Beam Fatigue Tests, at 11,000 RPM, of Smooth Specimens
of SAE 4340 Steel, of Nominal 260 ksi UTS

TABLE 5

NOTE: All specimens oiled, then dried with Kleenex, before testing.

76 ksi Stress Level, Nominal (Continued)

Specimen No. Diam., Inches	DUTA24 0.2310 16	DYTB09 0.2310 18	FRTD21 0.2308 19	FRTF30 0.2303 20	FNTD27 0.2305 21	DWTA32 0.2297 23	FATB07 0.2310 24	FCTC05 0.2309 25	FHTD20 0.2305 26
Machine No.									
Kilcycles to End of Test	10,246	10,414	11,118	11,076	10,368	10,085	10,240	10,025	10,254
Dist., Ctr. to Fracture									10,662

76 ksi Stress Level, Nominal (Continued)

Specimen No. Diam., Inches	FHTD14 0.2311 16	FRTF18 0.2309 18	DWTAA26 0.2310 19	DYTBD15 0.2305 20	DYTBD15 0.2305 21	FRTF27 0.2298 22	FOTH26 0.2308 23	DWTAA19 0.2305 24	FATB13 0.2307 25
Machine No.									
Kilcycles to End of Test	45,203	42,456	50,147	47,758	46,118	42,241	43,478	48,495	45,489
Dist., Ctr. to Fracture									50,132

76 ksi Stress Level, Nominal (Continued)

Specimen No. Diam., Inches	FNTD21 0.2308 16	DXTA16 0.2305 18	DYTBL2 0.2309 19	FRTD21 0.2305 20	FRTD21 0.2303 21	FRTF04 0.2301 22	FOTH23 0.2310 23	DUTA30 0.2302 24	DZTB23 0.2296 25
Machine No.									
Kilcycles to End of Test	11,070	1,444*	11,985	12,281	11,628	12,002	10,427	11,557	10,779
Dist., Ctr. to Fracture **		-0.22							11,936

76 ksi Stress Level, Nominal (Continued)

Specimen No. Diam., Inches	DXTA13 0.2310 16	FATB31 0.2310 18	FDTD07 0.2306 19	FHTD11 0.2306 20	FHTD11 0.2306 21	FHTF35 0.2303 22	DWTA20 0.2306 23	DZTB17 0.2309 24	FBTG24 0.2310 25
Machine No.									
Kilcycles to End of Test	14,575	14,043	12,042	15,755	15,377	14,582	13,385	15,890	14,574
Dist., Ctr. to Fracture									16,056

* FAILED

** Minus sign indicates distance toward motor end of specimens. Distance in inches.

TABLE 5

R. R. Moore Rotating Beam Fatigue Tests, at 11,000 RPM, of Smooth Specimens
of SAE 4340 Steel, of Nominal 260 ksi UTS

NOTE: All specimens oiled, then dried with Kleenex, before testing.

76 ksi Stress Level, Nominal (Continued)

Specimen No.	FRTD16	FRTF10	FTH11	DUTA33	DYTB18	FCTC02	FRTD18	FRTF33	FOTH14
Diam., Inches	0.2302	0.2310	0.2306	0.2312	0.2308	0.2308	0.2308	0.2311	0.2306
Machine No.	16	18	19	20	21	22	21	25	26
Kilocycles to End of Test	10,936	11,034	12,285	11,866	11,844	10,439	10,711	11,737	10,808
Dist., Ctr. to Fracture									11,956

76 ksi Stress Level, Nominal (Concluded)

Specimen No.	FRTF15	FTH30	DWTA29	DZTB11	FBTG36	FRTD17	FRTF26	FTH29	DXTA28	DZTB11
Diam., Inches	0.2310	0.2306	0.2300	0.2310	0.2308	0.2308	0.2306	0.2309	0.2310	0.2307
Machine No.	16	18	19	20	21	22	23	24	25	26
Kilocycles to End of Test	42,131	45,732	45,862	43,763	42,629	44,351	44,616	41,419	44,553	44,553
Dist., Ctr. to Fracture										

TABLE 6

Comparison of "1955" Tests with "1954" Tests in the Finite Life Region.
 R. R. Moore Rotating Beam Tests of Smooth Specimens of SAE 4340 Steel.

(Values of Fatigue Life at the Specified Stress Level are Anti-logs
 of Mean Log-Life.)

Ult. Strength Nominal	110 ksi			75 ksi		
Alternating Stress Level	81 ksi		77 ksi		75 ksi	
Life, 1955 Tests	112 kc	112 kc	183 kc	183 kc	205 kc	205 kc
Life, 1954 Tests	123 kc, not adjusted	193 kc, adjusted	193 kc, not adjusted	362 kc, adjusted	257 kc, not adjusted	282 kc, adjusted
Indications of F- and t- Tests on difference in log-life	not significant	significant	not significant	significant	may be significant	significant

Ult. Strength, Nominal	190 ksi	260 ksi
Alternating Stress Level	96 ksi	116 ksi
Life, 1955 Tests, Specimens oiled, then cleaned	119 kc	66 kc
Life, 1954 Tests	207 kc	157 kc
Indications of F- and t- Tests on Difference in log-life	significant	significant

Contrails

TABLE 7

Effect of Oiling and Cleaning Surface of Specimens
 R. R. Moore Rotating Beam Tests of Smooth Specimens of SAE 4340 Steel
 (Values of Fatigue Life at the Specified Stress Level are Anti-logs
 of Mean Log-Life)

190 ksi UTS		96 ksi Stress Level		"1955" Tests	
Oiled, then wiped dry, 119 kc	Oiled, then wiped dry, 119 kc			Oiled, then wiped dry, 119 kc	Oiled, then wiped dry, 119 kc
Oiled, 321 kc	Not Oiled, 100 kc			Oiled, then degreased, 105 kc	Oiled, then degreased, 105 kc
F- and t- tests: Significant	F- and t- Tests: Not significant			F- and t- Tests: Not significant	F- and t- Tests: Not significant
190 ksi UTS		96 ksi Stress Level		"1955" Tests	
1954 Tests		1955 Tests:	Oiled then wiped dry	207 kc	321 kc
		F- and t- tests:		Significant	Significant