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FOREWORD

This report was prepared by The Marquardt Corporation, Van Nuys, California, under Contract No. AF 33(616)-8209. This contract was initiated under Project No. 7381, "Materials Application," Task No. 738101, "Exploratory Design and Prototype Development." The work was administered under the direction of the Directorate of Materials and Processes, Deputy Commander/Technology, Aeronautical Systems Division, Wright-Patterson Air Force Base, Ohio with Lt. L. N. Hjelm as the project engineer.

This report covers work performed during the period from April 1961 through April 1962.

The following staff members contributed to the development of this program and their assistance is gratefully acknowledged: Mr. C. A. Hauck, Mr. R. M. Davids, Mr. G. D. Jones, and Mr. W. R. McIntosh.

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ABSTRACT

Metal reinforced refractory ceramic coating systems have been developed to successfully withstand temperatures in excess of 4000°F while providing insulative values for the supporting substructure. Eighty coating compositions were formulated and thermally evaluated in the plus 4000°F range.

Refractory coating of model nose cones and leading edges has been accomplished by utilizing a vibratory casting technique and the feasibility of this approach has been demonstrated. Twelve small scale leading edge models and six small scale nose cone models were coated in addition to one 500 square inch leading edge and one 500 square inch nose cone model. Thicknesses ranged from 0.3 to 0.75 inch.

Mechanical data and thermophysical properties of an advanced coating system were obtained.

An up-to-date bibliography of reinforced refractory ceramic systems and allied subjects is included.

This technical documentary report has been reviewed and is approved.

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I. INTRODUCTION

The tremendous advances which have been made in the field of aerospace vehicle and propulsion systems have created an urgent need for reliable hardware, with the inherent capability of existing in temperature regimes of plus 4000°F. Previous work (reported in Reference 1 under Contract AF 33(616)-5441) has illustrated the competence of a system composed of metal reinforced refractory ceramic coatings to fill such requirements.

The current work has been directed toward the optimization of these coating systems and the fabrication of model leading edges and nose cones for simulated environmental testing. The work reported under the subject contract has been divided into the following phases:

- Welding of refractory metal reinforcement media to a substrate and subsequent attempts to provide sufficient oxidation protection to the refractory metal
- 2. The optimization of coating compositions
- 3. The selection and optimization of the application technique of vibratory casting
- 4. Fabrication and coating of nose cone and leading edge models
- 5. Compilation of thermophysical data, including differential thermal analysis and thermal expansion characteristics
- 6. A bibliography of the state of the art of ceramic coating systems for use in the temperature regions above 3000°F

Manuscript released by The Marquardt Corporation on 14 May, 1962 for publication as a RTD Technical Documentary Report.

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II. SUMMARY

A. Purpose

The purpose of this program was to develop and evaluate reinforced refractory insulating type ceramic coatings of the type potentially useful on re-entry vehicles. The objectives of this program were;

- 1. To conduct a literature search and prepare a bibliography of related references
- 2. To select and optimize suitable coatings for plus 4000°F service temperature
- 3. To optimize application techniques
- 4. To prepare model nose cones and leading edges for simulated environmental testing
- 5. To document the development in the form of 35 mm color slides

B. Conclusions

1. Reinforcement

The present state of the art precludes the use of tantalum or molybdenum alloy reinforcement on N-155 structural substrate. Although they can be welded to the substrate satisfactorily, all attempts to protect these refractory metal reinforcement media resulted in unsuccessful systems.

Type 321 stainless steel reinforcement media can operate without requiring antioxidation coatings and without thermal failure when suitably submerged in the refractory ceramic coating.

2. Coatings

Practical reinforced refractory coating systems for long time use at plus 4000°F have been demonstrated.

3. Compositions

Refractory grains such as calcia-stabilized zirconia have been bonded into monolithic coatings utilizing fluorine-containing acids, amongst which hydrofluosilicic, monofluorophosphoric, and hydrofluosulfonic acids gave outstanding results.



4. Coating Techniques

Vibratory casting of thixotropic compositions has been demonstrated as a very successful technique of emplacing the refractory ceramic coating compositions.

For minor patching and touch-up requirements, troweling application of the coating composition is permissible.

5. Emittance Modifiers

The addition of metallic oxides to the basic coating compositions for the purpose of increasing the coating emittance at elevated temperatures has resulted in moderate gains in emittance values and, unfortunately, in drastic losses of structural integrity for the coatings.

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III. WELDING OF REFRACTORY METAL REINFORCING TO N-155 ALLOY SUBSTRATE

Tantalum and 0.5% titanium-molybdenum alloy were selected as reinforcing media for N-155 substrate structure. These refractory metals were obtained in the form of ribbon, 0.125-inch wide by 0.010-inch thick. The ribbon had been rolled from wire so that rounded edges were provided. These ribbons were then passed through mated gears which formed a continuous corrugated strip as shown in Figure 1. Different sets of mated gears provided corrugated strip which measured a nominal 5/32, 5/16, and 1/2 inch, peak-to-peak.

The corrugated tantalum reinforcement was spot welded at each node to 0.040-inch thick N-155 alloy substrate with no apparent embrittlement of the reinforcing media. A Stryco Welder, Type 4559 XUD, was used for this phase of the work. The following welder settings were typical:

Heat percent - - - - 52
Weld time cycles - - - 2

The 0.5% titanium-molybdenum alloy ribbon was quite brittle as received. It was quite difficult to form a continuous corrugated ribbon because of the inherent brittleness, and the brittleness increased markedly during the spot welding operation.

Because of these difficulties, 0.5% titanium-molybdenum alloy was discarded as a reinforcing medium. No attempt was made to find the reason for the brittleness.



IV. OXIDATION-PROTECTIVE COATINGS FOR TANTALUM AND 0.5% TITANIUM-MOLYBDENUM REINFORCING MEDIA

As previously noted, tantalum and 0.5% titanium-molybdenum alloys were selected as the reinforcing media for N-155 substrate structure. In an effort to evaluate oxidation-resistant coatings from outside vendors, three flat tensile specimens of 0.5% titanium-molybdenum alloy were sent to American Machine & Foundry to receive their siliconized coating for refractory metals. Samples of tantalum and 0.5% titanium-molybdenum reinforced N-155 were also sent to Boeing in Seattle for siliconizing.

The three flat tensile specimens of 0.5% titanium-molybdenum which were sent to American Machine & Foundry to receive their siliconized antioxidation coatingwere returned to Marquardt for evaluation. The test samples were resistance heated to 3000°F and held there for 30 minutes. Up to 2300°F, some small amounts of smoke were noted. This ceased when the temperature rose above 2300°F. No further smoking appeared. Some edge cracks were noted after cooling. The coating appeared glassy with a checkerwork of fine crazing. There was no apparent deterioration of the specimens.

Even though the results of coating of 0.5% titanium-molybdenum appeared promising, the material was discarded because of its brittleness, as discussed more fully in the preceding section.

The Boeing Company reported attack on nickel bearing alloys in their fluid-ized-bed siliconizing process. They therefore declined to siliconize the tantalum and 0.5% titanium-molybdenum reinforced N-155 test specimens which Marquardt sent them for processing.

The problems at this point were as follows: If the refractory metals were protected against oxidation by a diffusion coating such as American Machine & Foundry's disilicide, it was not possible to spot weld these coated metals to the N-155 substrate. According to Boeing's report if the reinforcing media were first spot welded to the N-155 substrate, subsequent disilicide coating would cause intergranular corrosion and low melting eutectic compounds in the substructure. Because of these problems, emphasis had to be shifted to the development of a dip type coating for tantalum that could be applied subsequent to the spot weld operation. The dip type coating systems which were evaluated are listed in Table I. A discussion of the evaluation of these compositions follows.

The tantalum reinforcing media was spot welded to the substrate as noted above. The coating compositions were mixed to a dipping consistency and the entire assembly was then dip coated. The coated assembly was air dried. Following the drying cycle, excess coating was removed from the back side of the substrate. Where required, a low temperature oven cure was employed.

The coated assembly was then encapsulated in a zirconia insulative body (TherMarq ZPF-100) and cured at 300°F for 2 hours to set up the chemical bond of the body.



Coating Number	Composition*	Cure Temperature
D-1	Sodium silicate (S-35 brand) + water	200°F
D-2	Potassium silicate (Kasil) + water	200°F
D-3	Pyro-Chrome	300°F
D-4	CP-44	300°F
D - 5	Alkaphos C	200°F
D-6	Aluminum powder (-325 mesh) + Brolite	ambient
D - 7	Chromium powder (-325 mesh) + Nicrobraz liquid	ambient
D - 8	Columbium powder (-325 mesh) + Nicrobraz liquid	ambient
D - 9	Vanadium powder (-325 mesh) + Nicrobraz liquid	ambient
D-10	Nickel powder (-325 mesh) + Nicrobraz liquid	ambient
D-11	Nichrome powder (-325 mesh) + Nicrobraz liquid	ambient
D-12	Electroless nickel plate	
D - 13	Silver paint	ambient
D-14	Aluminum paint	ambient
D-15	Alumina paint	600°F
D-16	Phosphatherm	200°F
D-17	Aluminum powder (-325 mesh) + Nicrobraz liquid	ambient
D-18	Zirconia paint	300°F

^{*}NOTE: All compositions were mixed to a slurry of dipping consistency.



The cured samples were then tested using the thermal drop test apparatus shown in Figure 2. Table II contains the data gathered from this evaluation.

The best protection was afforded by Composition D-7 which consisted of -325 mesh chromium powder suspended in Nicrobraz liquid.

More complex formulations for the oxidation protection of tantalum are shown in Table III.

Attempts to coat the refractory metal reinforcement, as spot welded to the substructure, using dip or paint type coatings were unsuccessful. Such coatings only partially protected the reinforcement from oxidation or they caused severe embrittlement after exposure to elevated temperatures.

In view of the poor results obtained in trying to provide oxidation protection for tantalum, another look was taken at the feasibility of utilizing type 321 stainless steel as the reinforcing media. This material requires no special oxidation protection when submerged in the refractory ceramic coating. When the apexes of the nodes were submerged about 0.150 inch below the ceramic hot face at 4000°F, the reinforcement remained intact.

At this point, the decision to use Type 321 stainless steel reinforcement rather than tantalum was made.



	COATINGS FOR TANTALIM ENCAPSULATED IN Thermarg ZPF-100	Visual Evaluation of Tantalum after Removal of TherMarq 2PF-100	Top of reinforcement gone.	Reinforcement completely gone.	Reinforcement intact but oxidized.	Reinforcement intact but oxidized.	Severe oxidation.	Reinforcement oxidized.	Reinforcement intact.	Reinforcement not oxidized.	Reinforcement badly oxidized.	Badly oxidized.	Severely oxidized.
	PALUM	F) Rear T/C	1550	1700 1505	111	: :	; ;	l l l l	1 1	; ;	1 1	! !	11
H	S FOR TAN	Temperature (°F) ont Front F ical Corrected T	1 1	; ;	1 1 1	11	11	11	1 1	11	1 1	1 1	1 ; 1 1
TABLE II		Tempe Front Optical	000† 000†	000†	000† ₁ 000† ₁	0001	0001	0001	000† 000†	000†	000t	0001	000†
	ING OF ANTIOXIDANT	Description of Coating	NazSiO4	Ag Paint	Pyro-Chrome	CP-44	Kasil	Alkophos C	Al powder	Cr powder	Cb powder	V powder	Ni powder
	THERMAL DROP TESTING	Coating Number	D-1	D-13	D-3	η - Ω	D-2	D-5	D-6	D-7	D-8	D-9	D-10
	HERMAI	Run Time (min)	10	10	7 00 W	9 4	919	9 #	10	10	10 10	. . † 00	Ω 4
	O.F.	Run No.	чa	H 0	Нам	rl (V	H (V)	чa	Цα	-l си	H (V)	нα	нα
	RESULTS	Sample* Number	Т	QI	8	4	<u>ι</u>	9	_	ω	6	10	11

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	Visual Evaluation of Tantalum after Removal of TherMarq ZPF-100	Reinforcement slightly oxidized.	Oxidized.	Reinforcement badly oxidized.	No oxidation evident.	Some oxidation.	Some oxidation.	No oxidation.	Slight oxidation.	Slight oxidation.	Oxidation evident.	Excessive oxidation.	a nominal 1/16 inch below the surface inch.
	°F) Rear d T/C	1 1	1 L 1 L	1 1	1640	1 1	1 1		1 1	ł	! ! ! !	1480 1470 1485	was a nomi s 1/8 inch.
II (Continued)	Temperature (°) ont Front ical Corrected	: :	11	1 1	ή210 	11	4190 4190	06५ <i>५</i> 06५५	4120 4120	7,260	4120 4120	111	reinforcing waistance was
	Tempe Front Optical	000† ₁	000+1	000+	3760 3800	1,020 1,000	3740 3740	000+	3680 3680	3800	3680 3680	0001	talum reir This dist
TABLE	Description of Coating	Nichrome powder	Electroless Ni	Al paint	Cr powder	Al ₂ O ₃ paint	${ m Na}_2{ m SiO}_{4}$	Cr powder	Phosphatherm	Al powder	Al203 paint	ZrO ₂ paint	The top of tan 15 through 22:
	Coating Number	D-11	D-12	D-14	D-7	D-15	D-1	D-7	D-16	D-17	D-15	D-18	through 14: Samples Nos.
	Run Time (min)	10	10	10	10	6 -8	5	10	10	15	10	ろうう	\-"°
	Run No.	1 0	цα	нα	Hα	Hα	чα	Hα	нα	н	Hα	наи	Samples Nos. of the body.
	Sample* Number	72	13	41	15	16	17	18	19	8	21	22	* Sampoof the

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TABLE III FORMULATION OF OXIDATION PROTECTIVE COATINGS FOR TANTALUM REINFORCEMENT MEDIA

	Parts by Weight									
Composition	1	2	3	4	5	6	7	8	9	10
Cr ₂ 0 ₃	50	10	45	45						
Colloidal Al ₂ 0 ₃		40	5	5						
Colloidal ZrO2								76	90	27
ZrO ₂ (-325 m)								18	6.5	20
NH4H2PO4	4.5	4.5	4.5							
H ₂ SiF6	25	25	10	4.5						
H ₂ PO ₃ F										1.0
Chrome metal				5						
Tin flake					50(a)	50(ъ)	50(c)			
Aluminum flake					50	50	50			
Poly vinyl chloride								6		
Kasil No. 1									3.5	
Gum Tragacanth										0.2
Nicrobraz							qs			
H ₂ O			qs	qs						

Notes: (a) - In air-oxidized

Samples 1, 3, and 4 afforded no oxidation protection during firing. Sample 2 was too lumpy and it therefore was not evaluated. No adherence was obtained with Samples 8 to 10.

⁽b) - Covered with graphite-oxidized (c) - Cured in vacuum at 400°F - no adherence after firing



V. BODY STUDIES

A. Composition Formulae

All composition formulae are given in Appendix III of this report.

B. Binders

Previous work (Reported in Reference 4) had clearly indicated the improvement of binders when fluorophosphoric acid was substituted for phosphoric acid. The incorporation of a fluorine radical was deemed significant. Work done at the Boeing Airplane Company (Reference 2) suggests the use of sulfuric acid as a binding agent for zirconia.

Therefore, to check the validity of the assumption regarding fluorine radicals, the compositions designated as TherMarq ZFS-101 to ZFS-106, inclusive were compounded. Compositions TherMarq ZFS-101 and ZFS-102, compounded directly from fluosulphonic acid, were quite difficult to handle. The acid fumed and was noxious. The compositions were of a mortar-like consistency which did not flow, resulting in difficult application. The curing cycle follows:

Time	Condition	Result
3 hours	Air dry	Soft
1 hour	250 °F	Reasonably hard
1 hour	300°F	Hard, but steel will scratch
1 hour	450°F	Hard, not scratched by steel

The noted curing cycle resulted in an extremely hard body for TherMarq ZFS-101 and ZFS-102, but bloating was evident.

TherMarq ZFS-103 was prepared as follows: $\rm H_2SO_4$ was added to water, then the solution was added to HF, blended $\rm ZrO_2$ grains were then added to the resultant solution. This method of compounding the formulation eliminated the violent reaction and most of the fumes noted above. The formulation molded easily and exhibited evidence of thixotropy.

The TherMarq ZFS-104 composition was applied to a reinforced panel and exposed to the Thermal Drop Test Apparatus. TherMarq ZFS-104 was essentially the same as TherMarq ZFS-101, but it contained less water. The sample withstood a corrected front face temperature of 4340°F for five minutes. There was no glazing or cracking of the surface. Aside from the difficulty in preparation, this composition was satisfactory.

The TherMarq ZFS-105 composition utilized both fluosulphonic acid and monofluorophosphoric acid. The composition was more of a slurry than mortar consistency. Four hours of air drying did not result in much "set". Twenty-four hours of air drying were required for this composition to set sufficiently for



handling. Further work on this composition was discontinued because of its poor setting characteristic.

TherMarq ZFS-106 was an effort to compound a zirconia body without the direct use of acids. The body did not hold up under thermal testing and this approach was discontinued.

The TherMarq ZPF-100 body, utilizing monofluorophosphoric acid, was further investigated using lithium zirconate and chromium metal powder. There was also an investigation of grain size of the fused zirconia. Results of thermal drop tests of these bodies are tabulated in Table VIII.

As an evaluation of the curing temperature, two TherMarq ZPF-100 bodies were prepared and cured at 300°F and 450°F, respectively. These bodies were exposed to ambient atmosphere conditions and their weights were periodically recorded. The weight changes shown in Table IV indicate that the higher curing temperature definitely slows down water absorption by the TherMarq ZPF-100 body.

TABLE IV

WATER ABSORPTION CHARACTERISTICS OF TherMarq ZPF-100 AT TWO CURE TEMPERATURES

Elapsed	TherMarq ZP	F-100 Cured at	300°F	TherMarq ZPF-100 Cured at 450°F				
Time (hrs)	Wt. (grams)	Wt. Gain Cum.	% Gain	Wt. (grams)	Wt. Gain Cum.	% Gain		
As cured	61.91			56.97				
18	62.15	0.24				- -		
24	62.17	0.26		- -		- -		
42	62.30	0.39	+0.63	57.04	+0.07	+0.12		
48	62.31	0.40	- -	57.04				
66	62.32	0.41	+0.66	57.06	0.09	+0.16		
90	62.32			57.07	0.10			
162	62.32			57.07				

All subsequent formulations were cured at 450°F, based on the data shown in Table IV.

Hydrofluosilicic acid was next investigated as a binding agent for zirconia bodies. Compositions formulated with this binder were highly thixotropic and exhibited longer working times than either the monofluorophosphoric or hydrofluosulfonic acids. Formulations compounded with H₂SiF₆ also stood up well under thermal testing. For these reasons, hydrofluosilicic acid was chosen as the best bonding agent.

C. Formulations

Various approaches were taken to improve the characteristics of the basic hydrofluosilicic-zirconia body, TherMarq ZSF-101. The modifying action of dif-



ferent ammonium salts on the composition was noted, as well as the effect of changes in water concentration. The composition was formulated using hydrofluosilicic acids of varying concentration obtained from several different commercial sources. The concentration of acid in the formulation itself was varied. Refractory oxides and coloring agents were added to or substituted in the formulation in attempts to increase emittance. A discussion of the technique for evaluating emittance is contained in Reference 3. Test coupons were cast as shown in Figure 3.

1. Salts and/or Metal Additions

The effects of salts and/or metal additions are demonstrated in TherMarq ZSF-101 through TherMarq ZSF-109.

These compositions showed the effect of various anions on the binding action of the acid. TherMarq ZSF-101 was a molar substitution of H_2SiF_6 for H_2PO_3F in the basic TherMarq ZPF-100 body. Compositions TherMarq ZSF-102 and -103 omitted the $NH_4H_2PO_4$ and had less working time and fluidity than ZSF-101. Ther-Marq ZSF-104 was formulated with zirconium metal to increase the emittance of the body. This composition mixed readily. However, approximately one minute of mixing time resulted in the evolution of an odorless, colorless gas as evidenced by foaming of the mix. After trowelling into a 1/4 by 2 by 3-inch cavity, further evidence of gas evolution was shown by lifting of the panel out of the cavity. The coating was cured at $450^{\circ}F$ and resulted in a hard, dense body of a light mottled gray color. Later exposure to the Thermal Drop Test-Apparatus resulted in violent breaking up of the body.

Compositions ZSF-105 through -109 also released volatiles in sufficient amounts to cause bloating.

Compositions TherMarq ZSF-105 through -109 contained additives designed to improve the workable life of the coating. The ammonium carbonate (Ther-Marq ZSF-105) reacted with the hydrofluosilicic acid and liberated carbon dioxide. There was a slight bond formation but the resultant body was very crumbly.

Compositions TherMarq ZSF-105 and ZSF-108 contained ammonium fluoride which is highly hygroscopic. TherMarq ZSF-106 was a "wet" mix that worked fairly well. Doubling the amount of ammonium fluoride resulted in a very friable piece. This approach was discontinued.

Compositions TherMarq ZSF-107 and ZSF-109 contained ammonium chloride. The ammonium chloride appeared to be beneficial in slowing the acid action and resulted in workable bodies.

2. Hydrofluosilicic Acid from Various Vendors

Work on the optimization of the ZSF series was complicated by receipt of hydrofluosilicic acids in varying concentrations from several sources. Each of the acids was titrated with standard NaOH to the phenolphthalein end point. The percent acidity of each vendor's product was calculated. These results are tabulated in Table V.



ANALYSIS OF HYDROFLUOSILICIC ACID
FROM VARIOUS VENDORS

Vendor	% Acid
Baker & Adamson	31.0
Baker Chemical Company	32.5
Davison Chemical (1 lb sample)	35.1
Davison (bulk)	34.3
Harshaw (bulk)	30.9

Based on the varying concentration of the acid, TherMarq ZSF-101 was formulated as shown in Table VI.

TABLE VI
FORMULATION OF TherMarq ZSF-101 USING DIFFERENT HYDROFIUOSILICIC ACIDS

		Parts by Weight									
Constituent	B & A	Baker	Davison	Davison (bulk)	Harshaw (bulk)						
ZrO ₂ -30F (Norton H)	40	40	40	40	40						
ZrO ₂ -325F (Norton I)	30	30	30	30	30						
NH ₄ H ₂ PO ₄	1	1	1	1	ı						
H ₂ SiF ₆	7.0	6.7	6.2	6.2	7.0						
н ₂ о			ąв	аp							

No difference was noted when TherMarq ZSF-101 was made up with the above acids in the quantities noted with H₂O as required. The Davison acid was used to determine the maximum and minimum useful acid concentration.

Acid Concentration

The effects of acid concentration, substitution of refractories, modifying effect of $NH_4H_2PO_4$, and weight reduction were evaluated in TherMarq ZSF-124 through -130 and -132 through -139.

TherMarq ZSF-125 was overly fluid, and set slowly. This composition, therefore, was designated as having the upper acid concentration. Approximately 2 hours were required for TherMarq ZSF-125 to set at room temperature in comparison with TherMarq ZSF-101 which set in 10 minutes under the same condition. The TherMarq ZSF-125 coupon blistered badly after thermal testing.

The acid concentration was then reduced in an effort to determine the minimum quantity of acid which still resulted in a bond. TherMarq ZSF-126 resulted in a very weak coupon and was therefore not thermally evaluated. Compo-



sitions TherMarq ZSF-127 through -130 were intermediate to TherMarq ZSF-125 and ZSF-126.

Water was added dropwise as the acid was further decreased. A visual attempt was made in each case to maintain a consistency similar to previous mixes. TherMarqs ZSF-127, -128, and -129, on air drying, set in a similar short time interval. TherMarq ZSF-130 set more slowly with little green strength, and 200°F curing was required to develop a better bond equivalent to the preceding mixes at room temperature. However, the coupon did not endure well during testing. It was therefore inferred that TherMarq ZSF-130 represented the lower limit of acid concentration. Thermal evaluation of these mixes is shown in Table XI and depicted in Figure 3.

Samples of TherMarq ZSF-132 through -139 developed blisters and small cracks during exposure to $4000^{\circ}F$ (uncorrected)*. Increasing the percentage of NH₄H₂PO₄, as in TherMarq ZSF-136 through -139, resulted in visible volatiles being given off during curing and firing. A one gram addition (1.43%) of NH₄H₂PO₄ to 70 grams of ZrO₂ was sufficient to prolong the working time of the mixture without creating visible volatiles.

Increasing the percentage of acid, as in TherMarq ZSF-132 through -135, indicated the effective acid range to be 6 to 7 grams per 70 grams of zirconia. TherMarq ZSF-132 was friable after firing. TherMarq ZSF-135 appeared slightly "soupy" during casting.

These compositions are shown after 4000°F exposure in Figure 4.

TherMarq ZSF-131 is basically the same composition as ZSF-124 but the differences turned out to be quite important. At the time this mix was formulated, proper raw materials were not on hand and the following substitutions were made:

TABLE VII
COMPARISON OF ZIRCONIA GRAINS

Constituent	ZSF-101	ZSF-124	ZSF-131
ZrO2 -30F (Norton H)	40	¹ 40	
ZrO ₂ -36F (Norton H)			40
ZrO ₂ -325F (Norton I)	30	30	~
ZrO ₂ -325F (Zircoa B)			30
NH [†] H ⁵ bo [†]	1	1	1
H ₂ SiF ₆ (Davison)	6.2	7	7

To pinpoint further the effective range of acid concentration, compositions ZSF-124 through -130 were subjected to thermal evaluation by torch testing. These results are shown in Table XI.

^{*} Ray-O-Tube out of service.



During thermal drop testing, it was noted that the temperature of the back face of test coupons never exceeded 2230°F. In an attempt to reduce the weight of the insulative coating, TherMarq ZSF-114 was formulated. This composition was a standard ZSF-101 mix with a 1/16-inch layer of Fiberfrax placed between the $\rm ZrO_2$ and the substrate. The thermal drop was not noticeably improved, and the effect on adherence of the coating to the substrate was detrimental.

4. Synergism

During previous experimental work, the phenomenon of "synergism" had been anticipated and noted. According to Webster, synergism is defined as "the cooperative action of discrete agencies such that the total effect is greater than the sum of the effects when the agency is taken independently." Compositions TherMarq ZSF-140 through -147 were formulated to evaluate synergistic effects.

TherMarq ZSF-144 and -145 were the wettest compositions of the series. TherMarq ZSF-146 and -147 required water addition. The addition of $(NH_4)_2$ SiF₆ to TherMarq ZSF-140 to -142, -145, and -146 seemed to dry out these compositions. A very pungent odor was apparent when H_2PO_3F was used in conjunction with H_2SiF_6 as in TherMarq ZSF-142 through -147.

The fired properties of TherMarq ZSF-144 and -145 were comparable to those of TherMarq ZSF-141. Since 14.3 percent less acid was used to bond Ther-Marq ZSF-144 and -145, it can be definitely concluded that this was the result of synergism. Visual evidence of this phenomenon is shown in Figure 5.

5. Emittance Additives

Coloring agents and refractory oxides were added to the basic formulation to raise the emittance of the body. TherMarq ZSF-110 through -113, -115 through -123, and -148 through -165 depict this phase of the work. Results are tabulated in Tables IX, X, XII, XIII, and XIV and illustrated in Figures 6 and 7.

The consistencies of these compositions were essentially the same with exception of TherMarq ZSF-113 and -162. These compositions required considerable additional water because of the presence of SrZrO₃. Thermal evaluation data appear in Tables IX and XIII.

TherMarq formulations ZSF-140 through -147, which were shown to be synergistic, were repeated with the partial substitution of chromia for zirconia fines in an effort to raise the emittance. Two wetting agents were also evaluated in this series. These formulations were designated TherMarq ZSF-148 through -155.

All of these formulations were too dry, and therefore required the addition of water to achieve coating consistency. The addition of the wetting agents did not appear significant. All of the compositions were light green in color. Emittance values and thermal evaluations are tabulated in Table XII.

TherMarq ZSF-164 and -165 were formulated to evaluate the substitution of cobalt oxide for zirconia fines.



These composites were later subjected to the thermal drop test torch along with a control sample of TherMarq ZSF-101. Thermal results are shown in Table XIV.

D. Thermal Evaluation

The thermal drop test was used to evaluate all coupons that exhibited any cohesiveness after curing.

Since the Ray-O-Tube was not always in operation, some data are included for which no emittance values are shown.

TherMarq ZSF-104 withstood a corrected front face temperature of 4340°F for 5 minutes when exposed to the thermal drop test.

Thermal histories of formulations based on monofluorophosphoric acid are tabulated in Table VIII.

TABLE VIII
RESULTS OF THERMAL DROP TESTING OF MODIFIED Thermary ZPF-100 BODIES

TherMarq Number	Coating Thickness		Ray-0-T	ube (°F) Back	Thermal Drop (°F/0.001 in.)	Remarks
ZPF-101	0.284	5	3650°	1300°	8.3°	No cracks - slight glaze
ZPF-102	0.303	5	3695°	1325°	7.8°	No cracks - slight glaze
ZPF-103	0.295	5	3650°	1250°	8.1°	No cracks - slight glaze

TherMarq ZPF-104 through -108 were exposed to the thermal drop test torch for a 15-minute cycle at temperatures of approximately 3700°F, uncorrected on the optical pyrometer. The bodies looked quite good in all cases. Small cracks formed, but these did not appear to penetrate the body thickness.

The thermal tolerances for some of the early high emittance coatings are shown in Table IX.

TABLE IX
THERMAL DROP TESTING OF Thermarq ZSF-110 THROUGH -113

TherMarq Number	Color Before Test	Run Time (min)	Optical Front Face Temperature* (°F)	Remarks
ZSF-110	Brown	5	3880°	Eroded and turned gray
ZSF-111	Green	5	4000° to 4100°	Glazed, no color change, looked good
ZSF-112	Pale blue-green	5	3930°	Cracked and blistered
ZSF-113	Ivory	5	4000° to 4250°	Glazed, looked good

^{*} Ray-0-Tube unavailable.



The comparison shown in Table IX was visual. Later composites of similar formulations of the TherMarq ZSF series (ZSF-119 and ZSF-155) showed the emittance to be in the 0.46 to 0.66 range. Measured emittance values of TherMarq ZSF-115 to -123 are compared with that of TherMarq ZSF-101 in Table X.

TABLE X

MEASURED EMITTANCE VALUES OF TherMarq ZSF-115 THROUGH -123

AS COMPARED TO TherMarq ZSF-101

TherMarq Number	Temperature (Corrected) (°F)	Emittance	Remarks							
ZSF-101	4257°	0.45	Slight cracks							
ZSF-115	3960°	0.75	Slight surface peeling - looked good							
ZSF-116	3950°	0.65	Slight glaze, very small cracks, looked good							
ZSF-117	4060°	0.52	Blistered							
ZSF-118		- -	Specimen friable - not tested							
ZSF-119	3803°	0.66	Peeling, small holes, and cracks							
ZS F- 120	4120°	0.77	Blistered and cracked							
ZSF-121	4200°	0.68	Blistered and cracked							
ZSF-122	4118°	0.78	Blistered and cracked							
ZSF-123	4155°	0.77	Blistered and cracked							

TherMarq formulations ZSF-115 and ZSF-116 emerged as the most promising compositions, having both high emittance and relatively good surface characteristics. Since the remaining coupons all blistered, cracked, and peeled, the resultant rough surfaces may have been responsible for the high emittances exhibited by these coupons in addition to the effect of the coloring agents.

The thermal evaluation of TherMarq ZSF-124 through -130 is shown in Table XI. This series of mixes was used to evaluate mechanical strength (tabulated in Tables XVI through XVIII, Section VII).

TherMarq ZSF-132 through -139 were a series of coatings which incorporated increasing amounts of binder or $NH_{\downarrow}H_{2}PO_{\downarrow}$ in their formulation. They are shown after testing in Figure 3. Visible volatiles were given off during exposure to $4000^{\circ}F$ (uncorrected) and all fired coupons developed blisters and small cracks.

TherMarq ZSF-140 through -147 were used to evaluate the phenomenon of synergism. Their appearances after exposure to a temperature of 4000°F is shown in Figure 5.

After exposure to the thermal drop test torch for fifteen minutes at 3900°F*, TherMarq ZSF-141, -144, and -145 had few blisters and still looked good. All other composites were blistered and cracked. Since TherMarq ZSF-144 and -145

^{*} Estimated



contained less combined acids than the binder used in ZSF-141, evidence of syner-gism was again concluded after thermal evaluation. Visual evidence of this result is shown in Figure 5.

TABLE XI

THERMAL EVALUATION OF Thermarq ZSF-124 THROUGH -130

TherMarq Number	Run Time (min)	Optical Front Face Temperature* (°F)	Remarks
ZSF-124	5	4000°	Very slight glazeappeared dense
ZSF-125	5	4000°	Badly blistered in 20 seconds; then rerun for 5 minutescracked and glazed
ZSF-126			Not run because of poor green cohesiveness
ZSF-127	5	4000°	Slight glazeappeared porous
ZSF-128	5	4000°	Glazedappeared porous
ZSF-129	5	4000°	Very glazedappeared porous
ZSF-130	5	4000°	Very, very glazedappeared porous and decrepitated

^{*} Uncorrected because Ray-O-Tubes were not available for this test.

TherMarq ZSF-148 through -155 contained partial substitutions of chrome oxide for zirconia fines. Thermal test data for this series are shown in Table XII.

TABLE XII
THERMAL EVALUATION OF Thermarq ZSF-148 THROUGH -155

		TherMarq ZSF-Number									
	148	149	150	151	152	153	154	155			
Optical pyrometer (°F)	3580	3510	3560	3000	3530	3460	3490	3850			
Ray-O-Tube (°F)	3350	3325	3410	2550	3140	3130	3150	3500			
True temperature (°F)	3880	3700	3730	3360	3930	3830	3850	4250			
Emittance	0.56	0.67	0.71	0.35	0.42	0.45	0.48	0.46			
Time (minutes)	5	5	5	5	5	5	_5	5			

The use of chrome oxide in these compositions raised the emittance. In all cases, the fired color was green to light gray with attendant blisters and cracks. A white powder was also deposited on the periphery of specimens ZSF-148 to -154. TherMarq ZSF-155 did not have this white precipitate. Since $(NH_{\downarrow})_2SiF_6$ was omitted from this composition, it may be concluded that its presence in Ther-Marq ZSF-148 to -154 is the cause of the white precipitate which was noted. It is believed that the difference in expansion rates of Cr203 and Zr02 was the



cause of the cracks and blisters noted. These composites are shown in Figure 6 as they appeared after exposure to a temperature of 4000°F.

Thermal results of a series containing additives other than chrome oxide are tabulated in Table XIII.

TABLE XIII

THERMAL EVALUATION OF Thermarq ZSF-155 THROUGH -163

				TherMa	arq ZSF	Number			
	155	156	157	158	159	160	161	162	163
Thickness (inches)	0.291	0.276	0.262	0.291	0.292	0.287	0.303	0.284	0.285
Optical pyrometer (°F)	3150	3440	3025	3410	3410	3200	3220	3190	3280
Ray-O-Tube (°F)	2450	3000	2450	2825	2850	2 420	2500	2500	2500
Thermocouple (°F)	1870	1740	1720	1860	1965	1810	1930	1770	1830
Time (minutes)	10	10	10	10	10	10	10	10	10
Emittance	0.35	0.36	0.55	0.26	0.27	0.38	0.42	0.45	0.37
True temperature (°F)	4180	3900	3530	4010	3970	3680	3980	3890	4170
Thermal drop (°F/mil)	7.9	7.8	6.9	7.4	6.9	6.5	6.8	7.5	8.2
Color (Fired)	Gray- green	Speck- led gray	Gray	Yellow -white	Yellow -white		Yellow -white		

TherMarq ZSF-162 and -163 had large cracks but no evidence of glaze or blisters. TherMarq ZSF-155, -157, -160, and -161 exhibited smaller cracks with no blisters or glazed spots. The remaining mixes, TherMarq ZSF-156, -158, and -159, were all blistered and cracked. TherMarq ZSF-157, which contained Co₂O₃ exhibited the highest emittance. Other mixes, based on cobalt oxide, are tabulated in Table XIV. The appearances of TherMarq ZSF-155 through -163 after exposure to a temperature of 4000°F are shown in Figure 7.

All specimens were cracked to some degree with the severity of the cracking directly related to the increasing percentage of cobalt oxide. The small gain in emittance hardly appeared worth the sacrifice in structural characteristics of the coating. The appearance of the specimens after exposure to a temperature of 4000°F is shown in Figure 8.



TABLE XIV

THERMAL EVALUATION OF Thermarq ZSF-164 AND ZSF-165 CONTAINING COBALT OXIDE AND COMPARISON WITH CONTROL SAMPLE Thermarq ZSF-101

	ZSF-101	ZSF-164	ZSF-165	
Thickness (inches)	0.285	0.288	0.279	
Optical pyrometer (°F)	3350	3220	3250	
Ray-O-Tube (°F)	2610	2650	2675	
Thermocouple (°F)	1800	1840	1900	
Time (minutes)	10	10	10	
True temperature (°F)	4200	3760	3820	
Emittance	0.41	0.53	0.53	
Thermal drop (°F/mil)	8.4	6.7	6.9	
Color (Fired)	Yellow- white	Gray	Gray	



VI. SIMULATED APPLICATION THERMAL STUDIES

Logical applications for reinforced refractory coatings include nose caps and leading edges. In both applications, the substrate back faces are submerged so that rejection of heat by radiation and convection is severely limited. The following experiment was conducted to simulate this condition and to obtain some comparative data.

A rectangular cavity with an annular ledge was carved into an alumina insulating brick in such a manner that a 2 by 3-inch reinforced refractory coupon could be placed into the ledge to close the cavity. The refractory face of the coupon was flush with the face of the brick, while the N-155 substrate of the coupon faced the rear of the cavity in the brick. Thermocouple leads were introduced through small diameter holes in the sides of the brick at an angle with the back of the coupon. The thermocouple was welded to the N-155 material in the usual fashion at a site opposing the area on the coating which received the thermal drop test flame.

Thermal drop tests were conducted to steady state conditions. A series of specimens of TherMarqs ZSF-101 and ZPF-100 were made each with nominal thickness of 1/4, 1/2, and 3/4 inch. These samples were then thermal drop tested both in air and encased in brick. The results are tabulated in Table XV.

It is quite apparent from Table XV that in all cases the thermal drop efficiency deteriorates quite rapidly in thicknesses over 1/4 inch. TherMarq ZSF-101 appears to have a slight edge over ZPF-100 in thicker sections. There is also less tendency to glaze and crack.



TABLE XV

COMPARATIVE THERMAL DROP DATA FOR TherMarq ZSF-101 AND ZPF-100

		TherMa	rq Number			
	ZSF-101	ZSF-101	ZPF-100	ZPF-100		
Run condition	In brick	In air	In brick	In air		
Time (minutes)	9•5	12	10	10		
Thickness (inches)	0.217	0.257	0.246	0.237		
Front face temperature (°F)	3130°	2820°	3750°	2880°		
Front face (corrected) (°F)	3490°	3140°	4200°	3205°		
Back face temperature (°F)	2020°	1200°	2230°	1205°		
Thermal drop (°F/mil)	6.8°	7.5°	8.0°	8.4°		
Run Condition	In brick	In air	In brick	In air		
Time (minutes)	16	11.5	21.5	16.5		
Thickness (inches)	0.542	0.490	0.493	0.488		
Front face temperature (°F)	2770°	2940°	3210°	2920°		
Front face (corrected) (°F)	3080°	3270°	3580°	3250°		
Back face temperature (°F)	1410°	1120°	1720°	1210°		
Thermal drop (°F/mil)	3.1°	4.4°	3.8°	4.2°		
Run condition	In brick	In air	In brick	In air		
Time (minutes)	25	30	25	21.5		
Thickness (inches)	0.758	0.788	0.722	0.723		
Front face temperature (°F)	2680°	2950°	2550°	2900°		
Front face (corrected) (°F)	2980°	3285°	2830°	3230°		
Back face temperature (°F)	1270°	760°	1270°	940°		
Thermal drop (°F/mil)	2.3°	3.2°	2.2°	3.1°		



VII. MECHANICAL PROPERTIES

A. Impact Strength

TherMarq compositions ZSF-124 through -130 were checked for resistance to impact using a falling ball technique. A ball was held by an electromagnet at a prescribed height above the specimen so that only potential energy was transmitted on impact. Sections of these coupons were first qualitatively checked by repeatedly dropping a steel ball bearing from a 2-foot distance. The parts were tested to destruction. Impact strength data are tabulated in Tables XVI through XVIII.

QUALITATIVE COMPARISON OF IMPACT STRENGTHS OF TherMarq ZSF-124 THROUGH -130

TherMarq Number	Weight of Ball (gms)	Diameter of Ball (inches)	Number of Times Dropped	Remarks					
ZSF-124	21.5	0.6875	5	Cracked					
ZSF-125	21.5	0.6875	3	Cracked					
ZSF-126				Not testedno green strength					
ZSF-127	21.5	0.6875	1 4	Cracked					
ZSF-128	21.5	0.6875	3	Cracked					
ZSF-129	21.5	0.6875	3	Cracked					
ZSF-130	21.5	0.6875	1	OK					
ZSF-130	95.0	0.905	1	Cracked					

Other sections of coupons of TherMarq ZSF-124 through -130 were evaluated quantitatively by gradually raising the electromagnet and thereby increasing the potential energy of the ball to the breaking point of the specimen. The results are shown in Table XVII.



			ZSF-130	No change	Cracked	Chipped	Broken										_							
	UGH -130		ZSF-129	No change	No change	Cracked	Cracked	Cracked	Cracked	Cracked	Cracked	Broken				-								
	QUANTITATIVE COMPARISON OF IMPACT STRENGTHS OF TherMargs ZSF-124 THROUGH -150	Numbers	ZSF-128	No change	Cracked	Cracked	Cracked	Broken																
	TherMarqs Z	TherMarq	LZT-ASZ	No change	No change	No change	Cracked	Cracked	Cracked	Cracked	Cracked	Cracked	Broken											
TABLE XVII	RENCTHS OF		Szr-isz	No change	Cracked	Cracked	Cracked	Cracked	Cracked	Cracked	Broken													
TAB	FIMPACT ST		ηΖΙ-ISZ	No change	Cracked	Cracked	Cracked	Cracked	Broken															
	OMPARISON O	Potential Energy	(inch- pounds)	0.028	0.071	0.113	0.156	0.199	0,240	0.284	0.327	0.369	0.412	0.188	0.470	0.752	1.034	1.316	1.612	1.880	2,162	ተ ተተ -	2.726	
	TITATIVE C	Distance	(inches)	. ᡮ	01	16	22	28	34	04	947	52	58		10	16	22	58	34	9	917	52	58	
	QUAN	Diameter of	Ball (inches)	0.375	0.375	0.375	0.375	0.375	0.375	0.375	0.375	0.375	0.375	0.6875	0.6875	0.6875	0.6875	0.6875	0.6875	0.6875	0.6875	0.6875	0.6875	
		Weight of	Ball (gms)	3.2	3.2	3.2	3.2	3.2	3.2	3.2	3.2	3.2	3.5	21.5	21.5	21.5	21.5	21.5	21.5	21.5	21.5	21.5	21.5	



Impact strengths calculated from these data are shown in Table XVIII.

TABLE XVIII

ULTIMATE IMPACT STRENGTHS OF TherMarq ZSF-124 THROUGH -130

TherMarq Number	Impact Strength to Break (inch-pounds)	Impact Strength to Crack (inch-pounds)
ZSF-124	2.726	1.612
ZSF-125	1.316	0.369
ZSF-126	Not tested	Not tested
ZSF-127	0.412	0 .1 56
ZSF-128	0.752	0.412
ZSF-129	0.369	0.113
ZSF-130	0.327	0.240

B. Modulus of Rupture

Test bars approximately 5 by 1 inch were molded for modulus of rupture tests. The bars were cured to 450°F and then broken in a Baldwin Physical Testing Machine using three point loading on a 4-inch span. The loading rate in all cases was 500 lbs/min. Modulus of rupture data for these compositions are shown in Table XIX.

TABLE XIX

MODULUS OF RUPTURE COMPARISON OF TherMarq ZSF-101, -124, & -131

TherMarq Number	Sample Number	Modulus of Rupture (psi)
ZSF-101	1 2 3 Average	942 949 <u>930</u> 940
ZSF-124	1 2 Average	515* 1050 783
ZSF-131	1 2 3 4 Average	708 595 770 <u>718</u> 698

^{*} Presumed anomolous reading, may have been due to incipient crack or nonuniform porosity.

Table XIX shows that TherMarq ZSF-101 and -124 are apparently stronger than the TherMarq ZSF-131.



VIII. THERMOPHYSICAL PROPERTIES

A. Thermal Expansion

The thermal expansion characteristics of the TherMarq ZSF-124 coating composition were determined using a Brinkman-Netzsch automatic recording dilatometer equipped with a sintered alumina sample holder and transmission rod. The apparatus is shown in Figure 9. The equipment was calibrated using a sample of single crystal alumina whose orientation and thermal expansion were precisely known.

Expansion runs were made with the TherMarq ZSF-124 composition in the air dried condition, after curing at 450°F, and after firing at 2700°F for 12 hours. A program controlled heating rate of 9°F/min was used for all determinations. The results of these determinations are shown graphically in Figure 10. The length changes indicated for the air dried and 450°F cured specimens were both reversible and irreversible. The length changes shown for the prefired specimen were totally reversible. The magnitude of the permanent shrinkage of the air dried and 450°F cured specimens when fired to 2700°F were 2.43 and 1.98 percent, respectively.

B. Differential Thermal Analysis (DTA)

The DTA equipment shown in Figure 11 was used to detect the presence of chemical or phase transformations in the TherMarq ZSF-124 coating composition. The DTA apparatus consisted of a vertical Pt-40% Rh wound tube furnace that could be raised or lowered for access to the sample holder. The sample holder was a sintered alumina block containing three centrally located cavities for the sample, the reference material, and the reference thermocouple. Reaction between the sample material and sample holder was prevented by encapsulating the sample in platinum foil before inserting into the sample holder cavity. Pt vs. Pt/13 Rh was used for both the reference and the differential thermocouples. The output of the couples was fed into a Brown X-Y recorder. An output of 40 millivolts produced a one-inch deflection on the DTA trace.

Results from numerous differential thermal analyses of TherMarq ZSF-124 at temperatures up to 2700°F and at various heating rates showed this formulation to be completely free of any chemical or phase transformation.



IX. VIBRATORY CASTING TECHNIQUES

Reinforced refractory ceramic coatings have been applied in the past as trowelled coatings. Vibratory casting is an excellent method of extending the application range of this type coating.

The $\rm H_2SiF_6$ bonded ceramic coatings were highly thixotropic: a stiff, mortar-like consistency when at rest and quite fluid under vibratory loading. A vibratory table was constructed which consisted of a suitably suspended plate, activated by means of an "Airoviber" vibrator. Metal or modified plaster molds were used to form the parts. The coating material was placed inside the cavity and the induced vibration caused the material to flow and fill the cavity. An added advantage of this method is that the size of the part to be coated is limited only by the capacity of the vibrator.



X. MODEL NOSE CONES AND LEADING EDGES

Designs for model leading edges and nose cones of approximately 20 square inch and 500 square inch sizes were submitted to the ASD Project Engineer for approval before fabrication of the units was started.

Once approval had been obtained and the prototype substrates had been fabricated, molds were designed and fabricated of steel and Hydrocal plaster. The plaster was impregnated with 20% Elvax-fortified, paraffin wax as a parting agent. The molds were used in the vibratory casting of the small scale model nose cones and leading edges. The table was vibrated at a speed of 5400 vibrations per minute during the casting of the model nose cones and leading edges. These molds were designed with interchangeable inserts as shown in Figure 12. Figures 13 and 14 show the nose cone and leading edge molds before casting of the plaster support.

Type 321 stainless steel reinforcement 0.125 inch wide by 0.010 inch thick was spot welded to the substrates in preparation for refractory coating. The welding operation is shown in Figure 15 and Figures 16 and 17 are views of the small scale models with reinforcement completely attached.

The parting surface of each of the leading edge molds was stainless steel. When these surfaces were waxed, they afforded readily releasable areas. Because of the complexity of fabricating a metal mold for the nose cone, the parting surface of the mold was waxed plaster.

The refractory coating material (TherMarq ZSF-124) was mixed in the sigma bladed vacuum dough mixer shown in Figure 18. Ammonium hydroxide was used to neutralize and prevent acid vapors from entering the vacuum pump.

The small scale components were aligned in their proper molds, bolted to a vibratory table, and the casting operation was completed. Figure 19 shows the casting operation. No problems were encountered in releasing the leading edges from the molds. In the case of the nose cones, however, extremely poor parting characteristics resulted in a 50% yield when the parts were removed from the mold.

Figure 20 shows the completed small scale components after curing. Thermocouples were attached at the request of the ASD Project Engineer. Three basic curvatures of leading edge models were coated; two samples of each thickness were made. The cured weights of the specimens are shown in Table XX.

The general procedure in vibratory casting of the 500 square inch models was similar to that followed in the fabrication of the small scale components. The 500 square inch leading edge model substrate is shown in Figure 21. Stainless steel reinforcing ribbon 0.125 inch wide by 0.10 inch thick was spotwelded to the substrate in preparation for refractory coating. Approximately 3,000 spot welds were needed to attach the reinforcements to each model. The 500 square inch leading edge model with reinforcement attached is shown in Figure 22. A similar view of the 500 square inch nose cone model is shown in Figure 23.

TABLE XX
WEIGHTS OF REFRACTORY CAST ON SMALL SCALE COMPONENTS

Component	Part Number	Thickness of Refractory (inches)	Refractory Weight (grams)
Nose cone	703228-401	0.3	899
	703228-401	0.3	879
	703228-401	0.3	877
Nose cone	703228-401	0.5	1650
	703228-401	0.5	1615
	703228-401	0.5	1600
Leading edge	703228-403	0.3	1010
	703228-403	0.3	1080
Leading edge	703228-403	0.5	1810
	703228-403	0.5	1 7 50
Leading edge	703228-403	0.75	2582
	703228-403	0.75	2670
Leading edge	703228-404	0.3	1073
	703228-404	0.3	1066
Leading edge	703228-404	0.5	1770
	703228-404	0.5	1741
Leading edge	703228-405	0.3	1150
	703228-405	0.3	1128

To avoid the necessity of fabricating a special male mandrel, the substrate for the nose cone was first temporarily coated with plaster and used as a mandrel upon which the female mold could be cast. Figure 24 shows the machining of this mandrel to the desired dimension. The finished mandrel, coated with shellac, is shown in Figure 25.

Because of the difficulties experienced in separating the small scale nose cones from their molds, a five-piece mold was designed for the 500 square inch nose cone model. This mold is shown in Figure 26. A fluorine-base parting agent-KEL-F--soft wax, Grade 40 was also obtained from E. I. DuPont, Inc. as a mold release coating.

The mold for the 500 square inch leading edge model was fabricated by bending a sheet of cold rolled steel to the desired dimension. This surface was then waxed in a similar fashion to the method used for the small scale leading edge molds.

Both large scale components were then aligned in their respective molds, they were bolted to a vibratory table, and the refractory casting operation was



accomplished. These components are shown prior to casting in Figures 27 and 28 and after casting in Figures 29 and 30.

No problems were experienced in releasing the 500 square inch leading edge model from its mold. The parting agent--KEL-F soft wax, Grade 40--used in the nose cone mold tended to adhere to the mold and the part rather than permit separation. Extensive patching of the part was therefore required. The weight of refractory used to coat the 500 square inch nose cone model was 10,201 grams, and that used to coat the 500 square inch leading edge mold was 10,276 grams. The refractory coating thickness was a nominal 0.3 inch in both cases. The finished components are shown in Figures 31 and 32.



XI. FUTURE WORK

Lack of time necessitated curtailment of development work in several areas which gave promise of beneficial results if pursued further. Other areas were not touched upon because they were outside the scope of the contract, although related thereto. In making recommendations for future work, both areas are included in the following items:

- 1. Reliable antioxidant coatings must be developed for refractory metal reinforcement media. Such coating must be applied without detrimentally affecting the substrate structure, or interfering with attachment welds.
- 2. Because the trend is to higher temperatures and more dynamic environments, the need for stronger and more refractory reinforced ceramic coatings to withstand the environments becomes apparent. Thoria and modified thoria coatings should be investigated for the plus 5000°F regime.
- 3. The applicability of vibratory casting has been demonstrated. However, the mold preparation and better parting agents require further investigation.
- 4. Synergism has been lightly touched upon. Further work on this phenomenon is indicated.
- 5. No satisfactory additive was found to raise the emittance of the coating without tending to destroy the coating. Alteration of the surface by staining, roughening, or combinations of these operations may yield the desired result. This should be investigated.



XII. ADDENDUM

The following work was not done under the subject contract but is a pertinent item of interest, since it concerns a large scale application of reinforced refractory coatings to a 2800°F Clean Air Heater under Contract AF 33(600)-37639.

A large heat exchanger approximately 14 feet high by 18 inches ID in four sections with eight pipe sections and elbows was coated with 1 5/8 inch thick reinforced TherMarq ZSF-124.

The reinforcement was corrugated Type 321 stainless steel 1/8-inch wide by 0.010 inch thick. Over 25,000 spot welds were used to fasten the reinforcement to the ID of the heat exchanger sections. Approximately one ton of coating material was installed by vibratory casting. Three sections of the heat exchanger with reinforcement attached are shown in Figures 33 through 35. Figure 36 shows the top dome section during the coating operation. Four views of sections of the heat exchanger after coating are shown in Figures 37 through 40. The secondary piping for the unit is shown in Figures 41 and 42, and Figure 43 shows the completely installed unit.

The completed unit will operate using vitiated air at 3100°F as the heating medium. Clean air will enter at 1500°F and will be heated to 2800°F. The normal operating pressure will be 220 psig.



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- 4. WADC TR 59-102, "Development and Evaluation of Insulating Type Ceramic Coatings", The Marquardt Corporation, E. W. Blocker, et al, November, 1959.



FIGURE 1. Forming Corrugated Reinforcement Strip

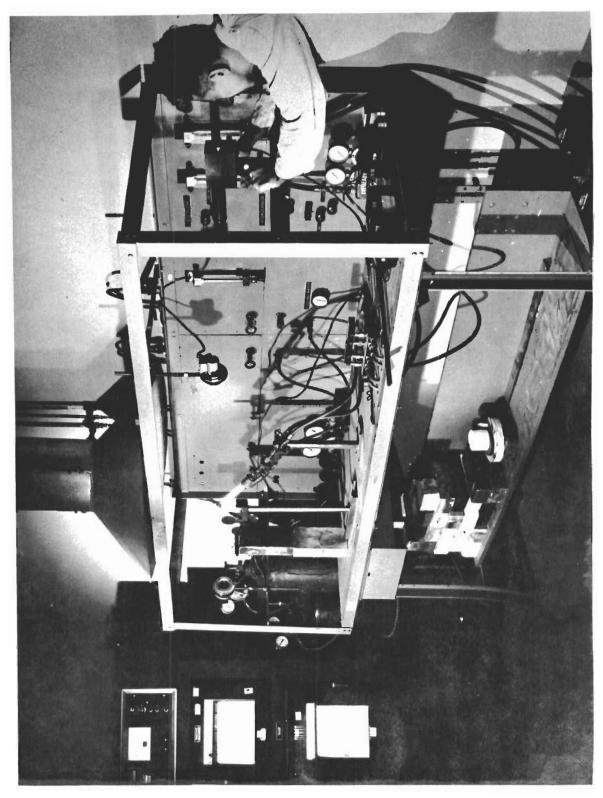


FIGURE 2. Setup for Thermal Drop Testing

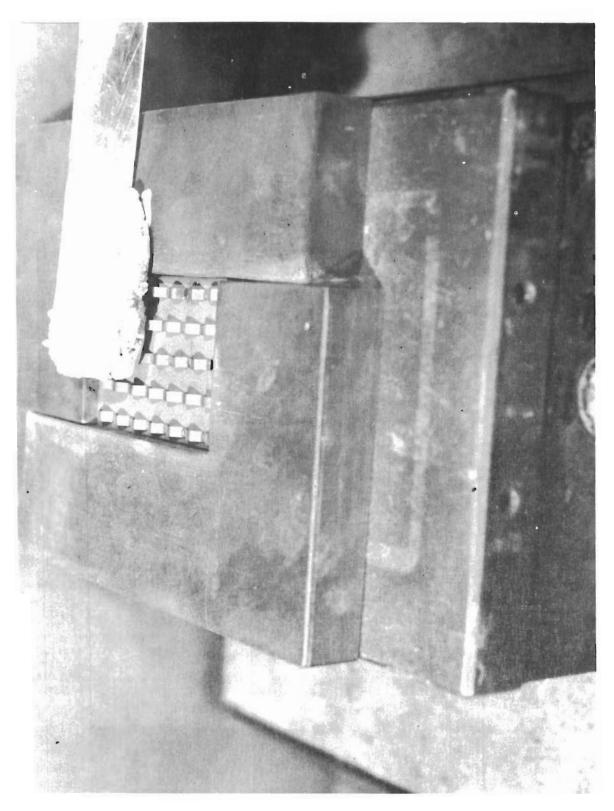


FIGURE 3. Casting of Coupons

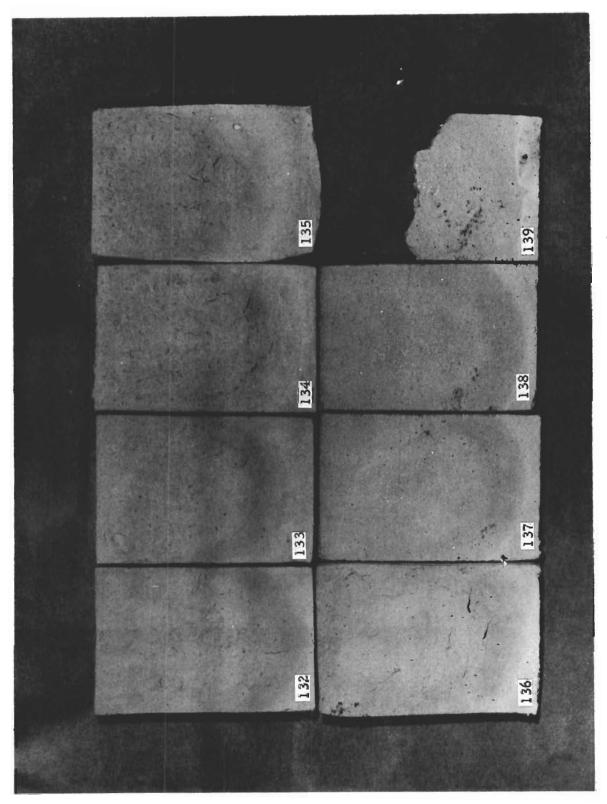


FIGURE 4. TherMarg ZSF-132 through -139 after 4000°F Exposure

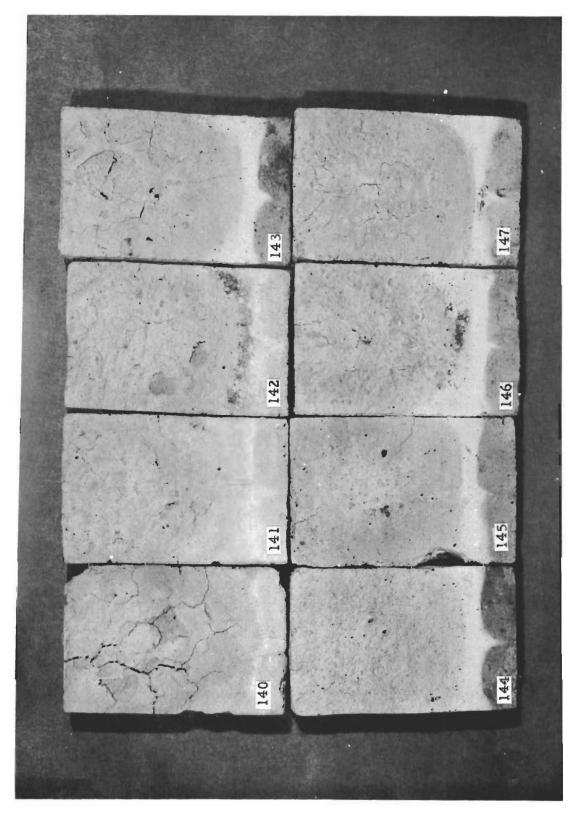


FIGURE 5. TherMarg ZSF-140 through -147 after 4000°F Exposure

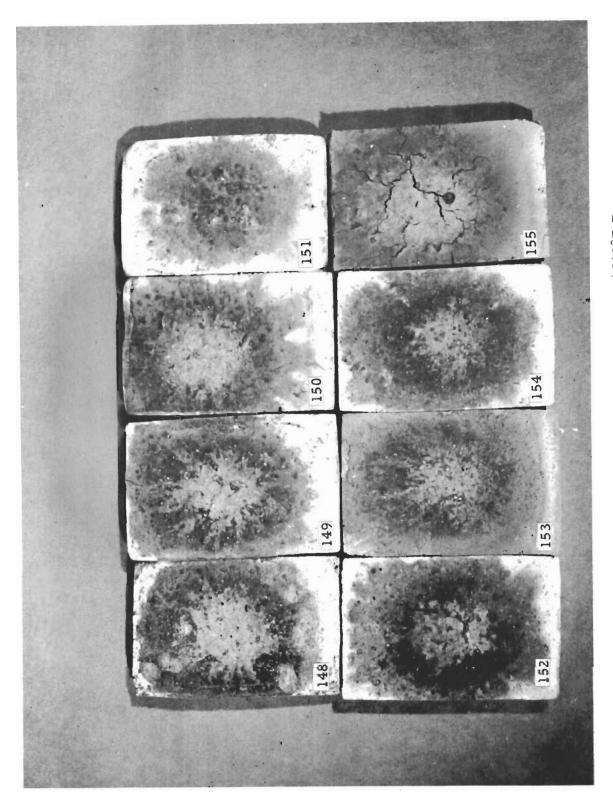


FIGURE 6. TherMarq ZSF-148 through -155 after 4000°F Exposure

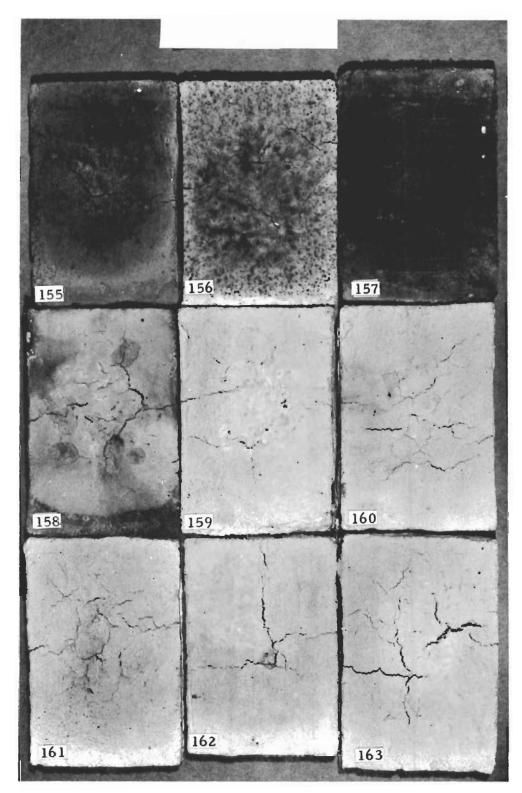


FIGURE 7. TherMarq ZSF-155 through -163 after 4000°F Exposure

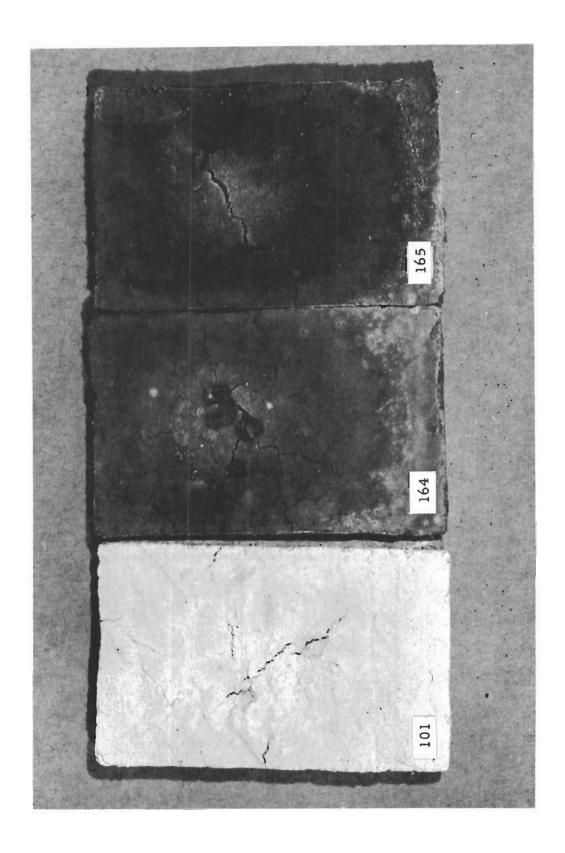


FIGURE 8. TherMarg ZSF-101, -164, and -165 after 4000°F Exposure

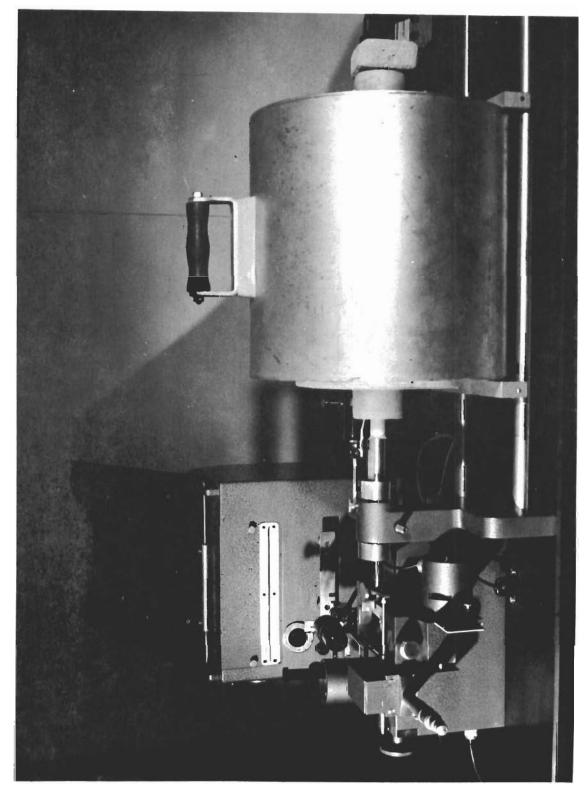


FIGURE 9. Thermal Expansion Apparatus

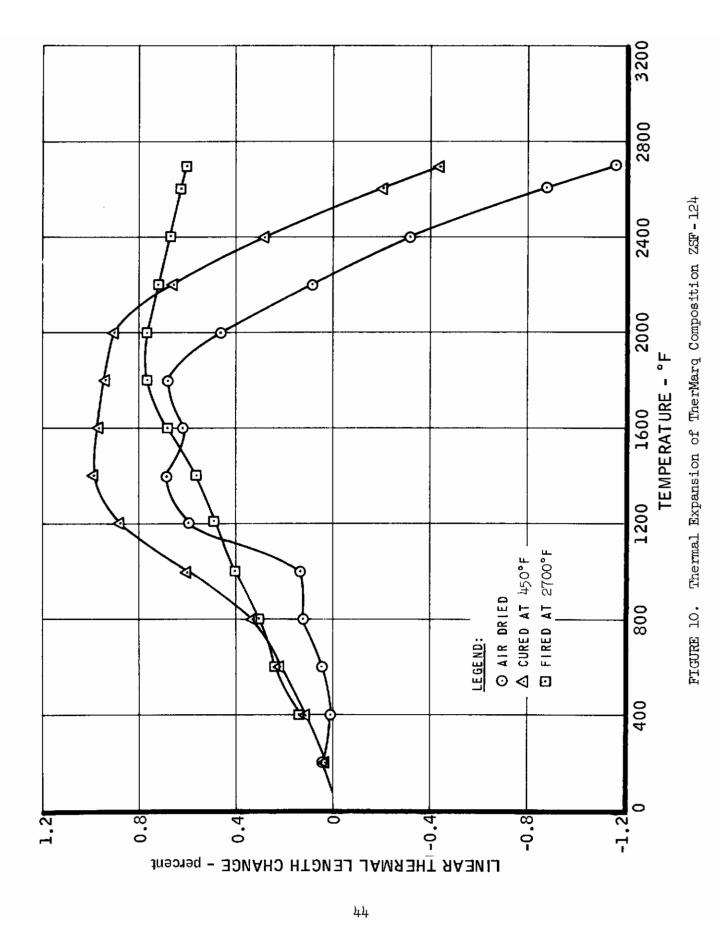
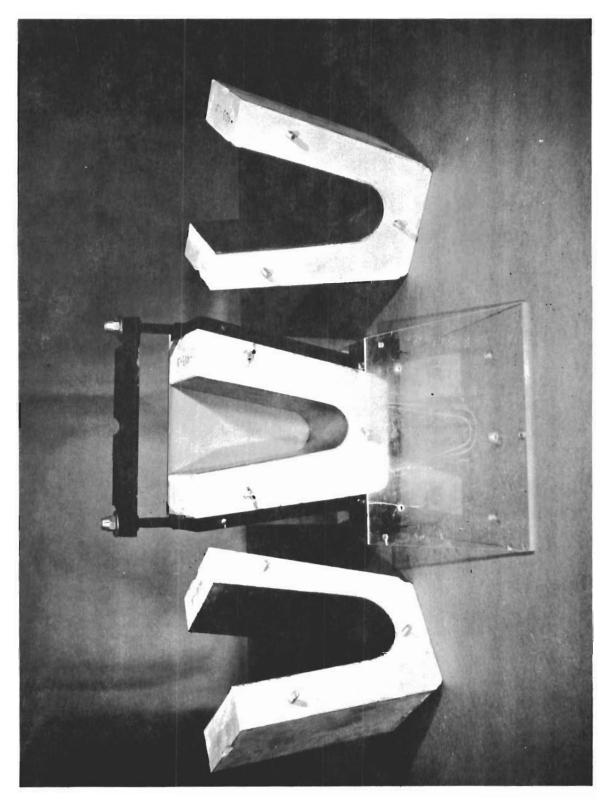




FIGURE 11. Differential Thermal Analysis Apparatus



Small Scale Leading Edge Mold with Interchangeable Inserts FIGURE 12.

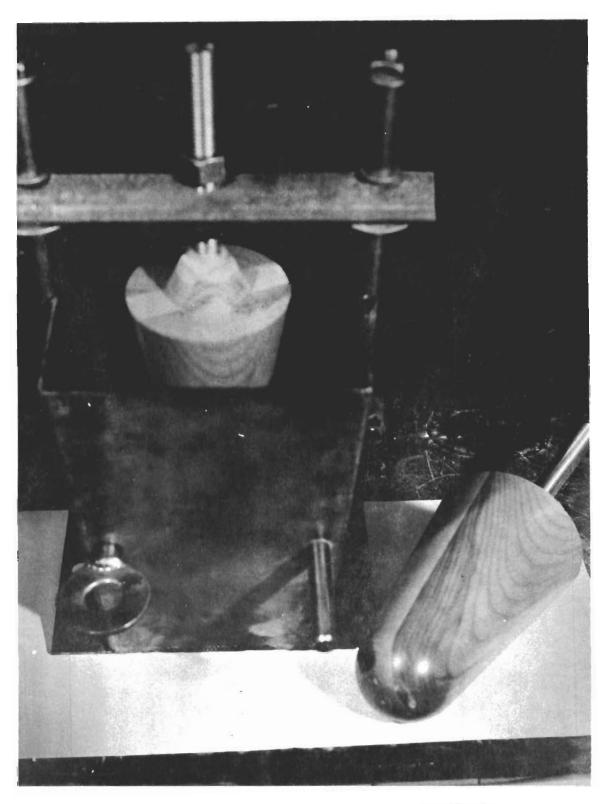


FIGURE 13. Nose Cone Mold before Plaster Casting

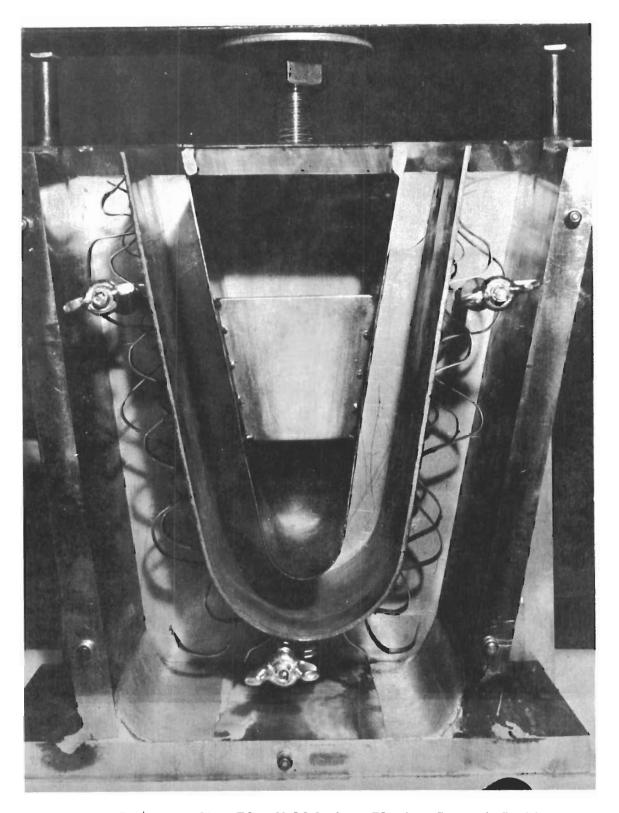


FIGURE 14. Leading Edge Mold before Plaster Support Casting

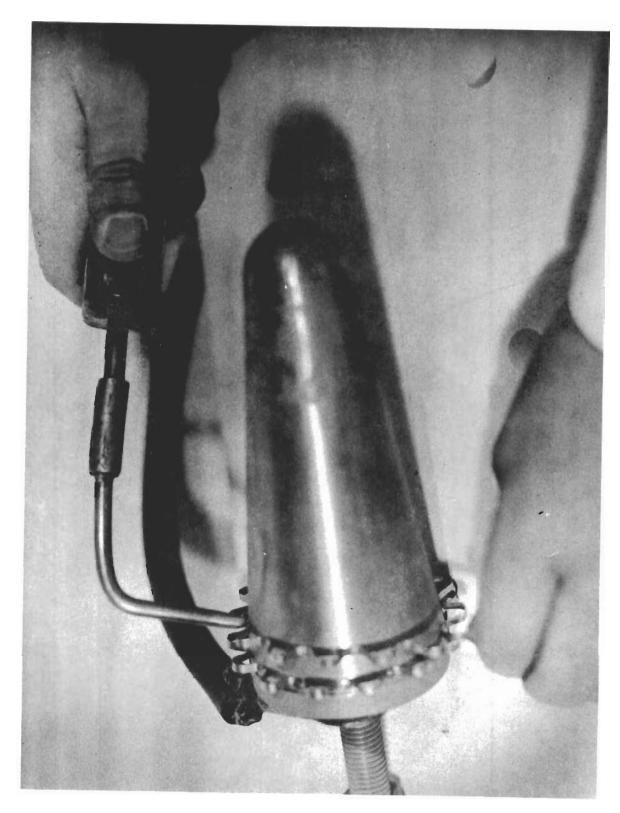


FIGURE 15. Welding Reinforcement Strip on Model Nose Cone

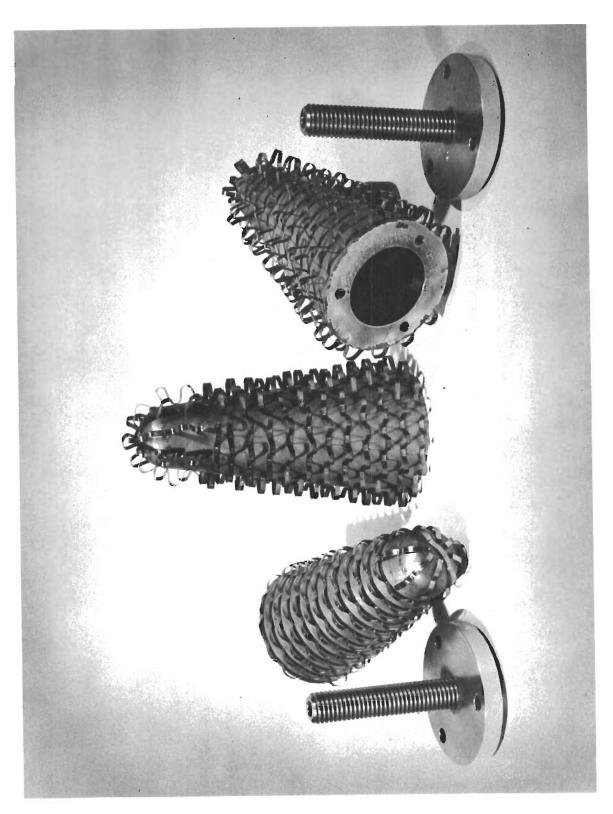


FIGURE 16. Small Scale Model Nose Cones with Reinforcement

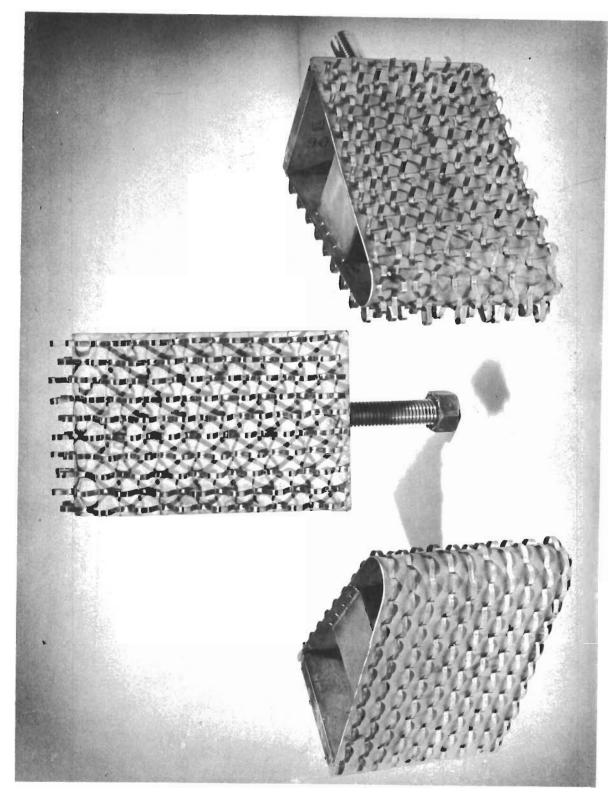


FIGURE 17. Small Scale Model Leading Edges with Reinforcement

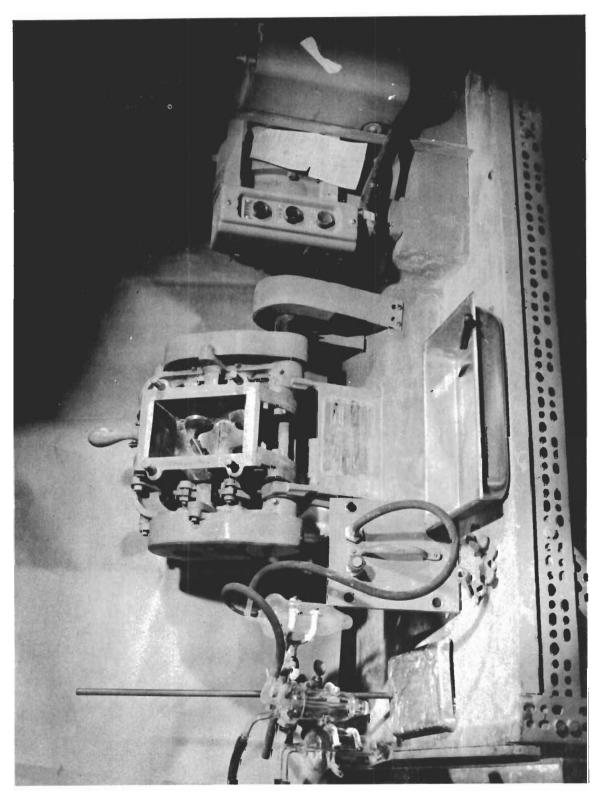


FIGURE 16. Signa Bladed Vacuum Dough Mixer

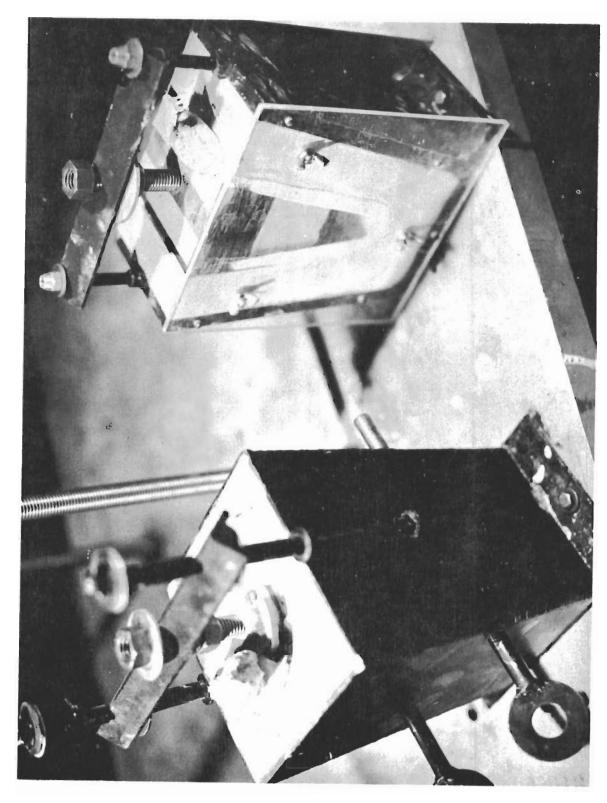


FIGURE 19. Casting TherMarq ZSF-124 in Model Nose Cone and Leading Edge Molds

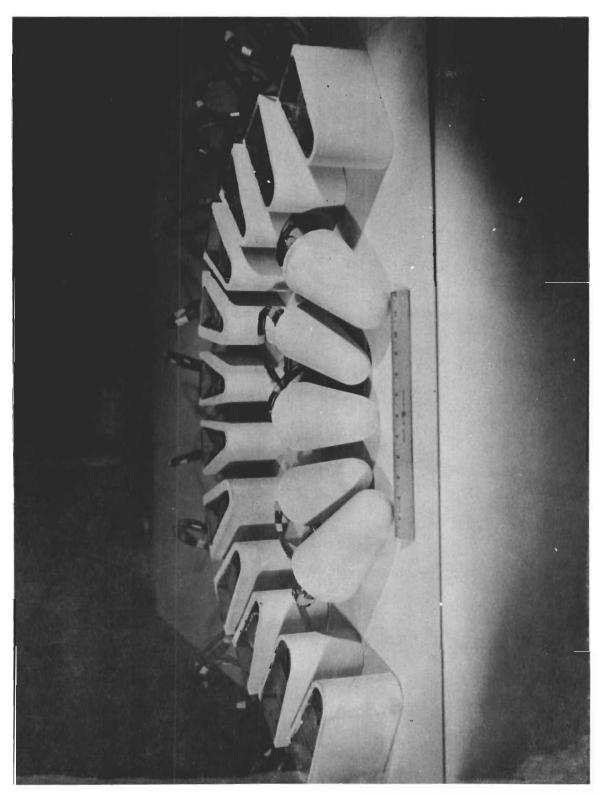


FIGURE 20. Small Scale Model Nose Cones and Leading Edges

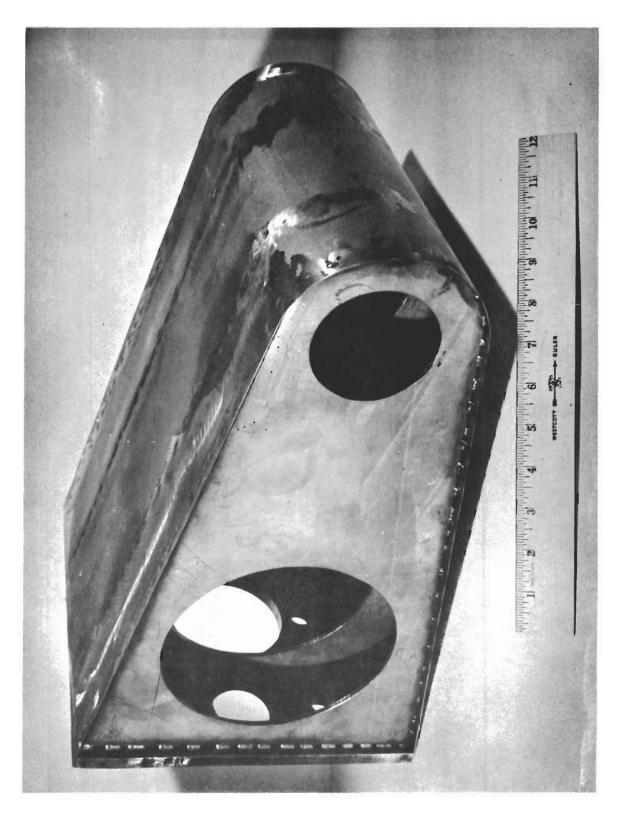


FIGURE 21. 500 square inch Model Leading Edge Substrate



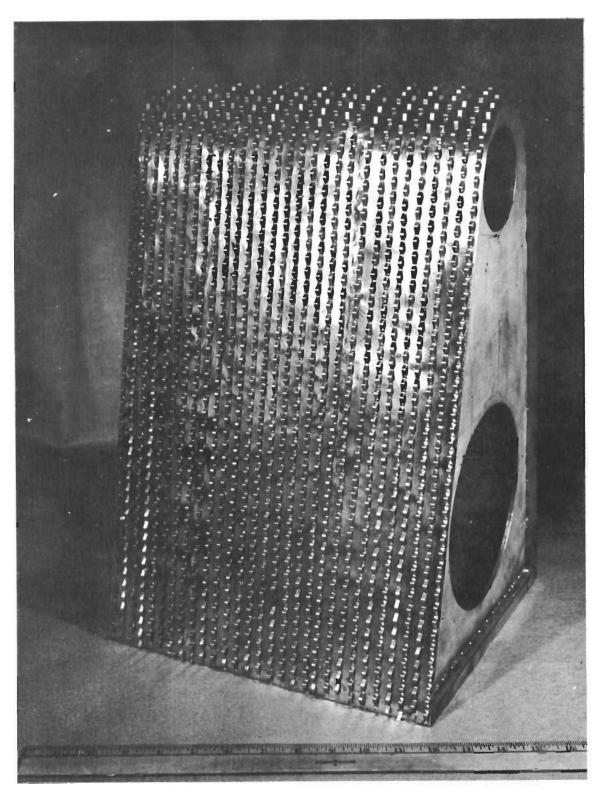


FIGURE 22. 500 square inch Model Leading Edge with Reinforcement

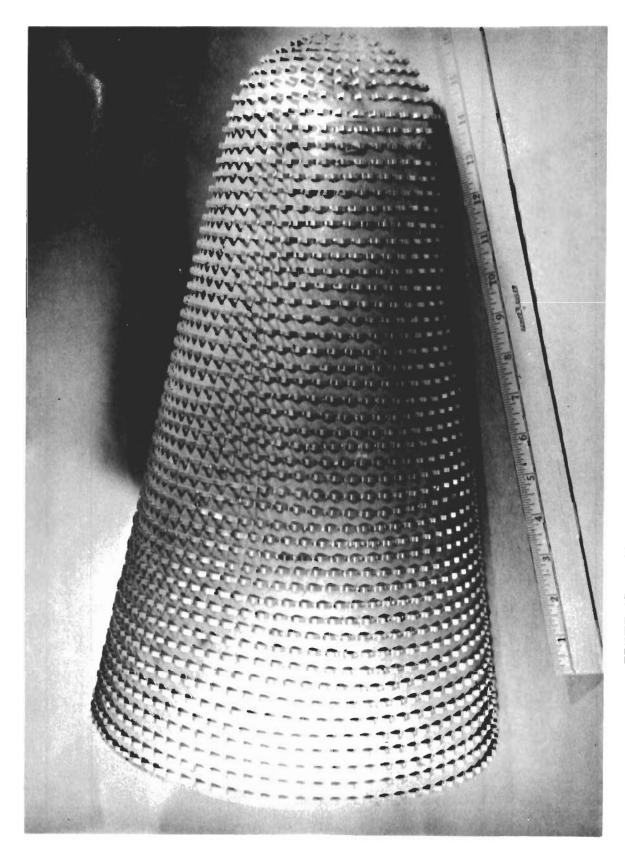


FIGURE 23. 500 square inch Model Nose Cone with Reinforcement

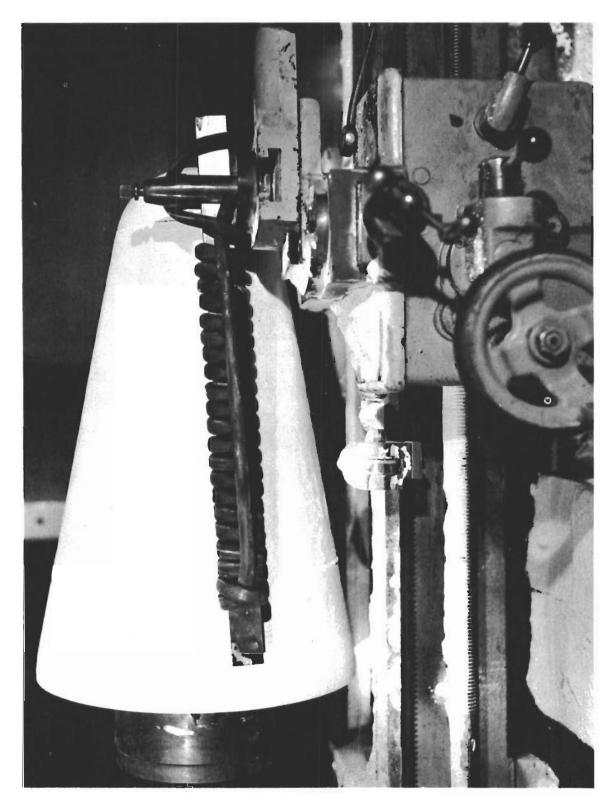


FIGURE 24. Machining 500 square inch Model Nose Cone Mandrel



FIGURE 25. Plaster Mandrel for 500 square inch Nose Cone

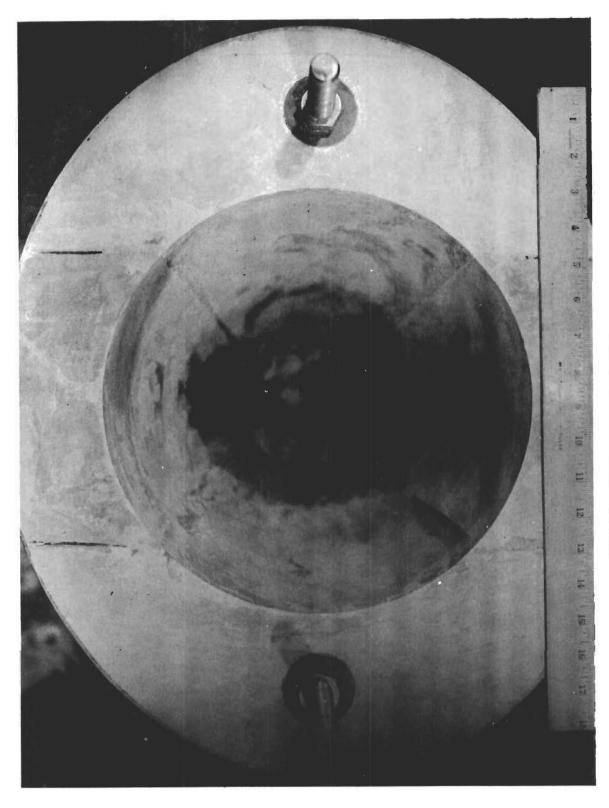


FIGURE 26. 500 square inch Model Nose Cone Mold

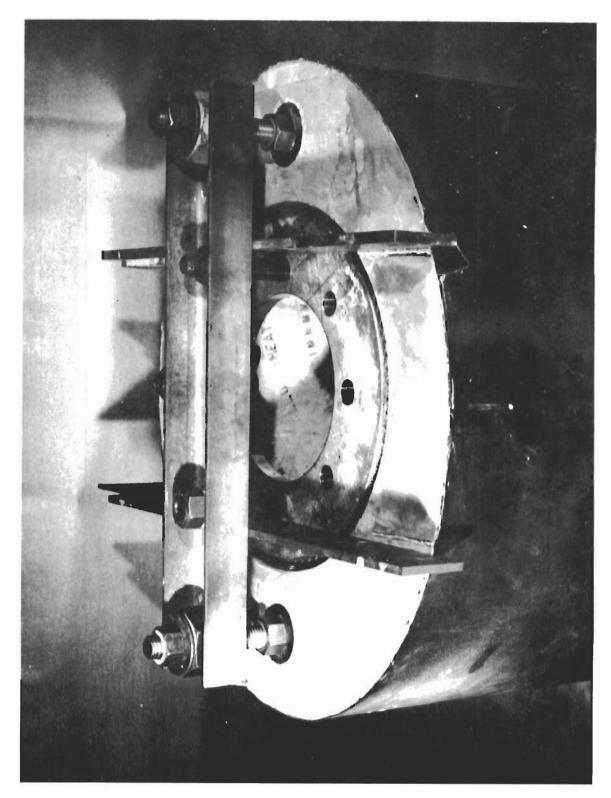


FIGURE 27. 500 square inch Model Nose Cone in Mold--Ready for Refractory Casting

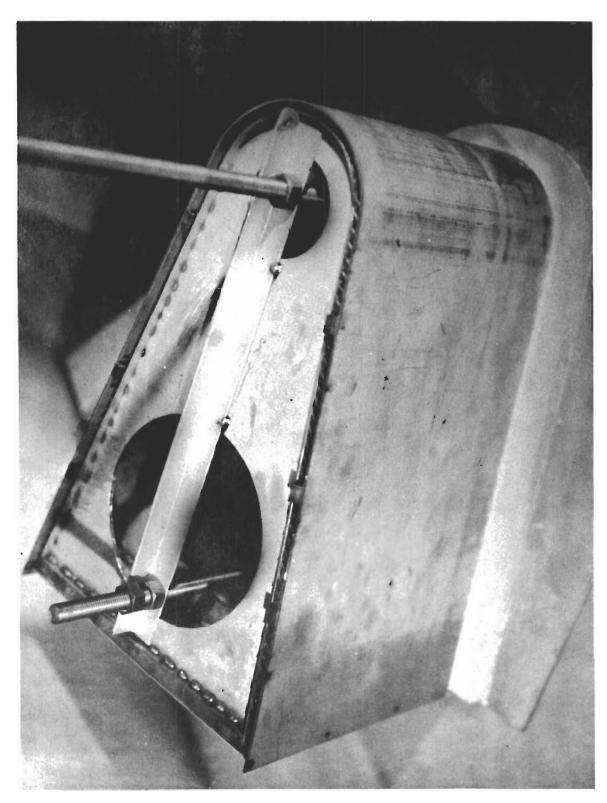


FIGURE 28. 500 square inch Leading Edge Ready for Refractory Casting

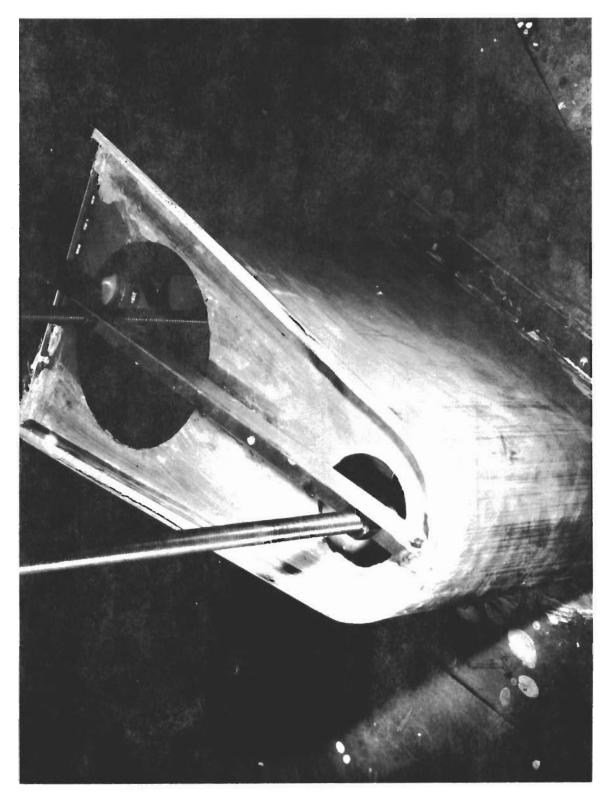


FIGURE 29. 500 square inch Model Leading Edge in Mold after Casting

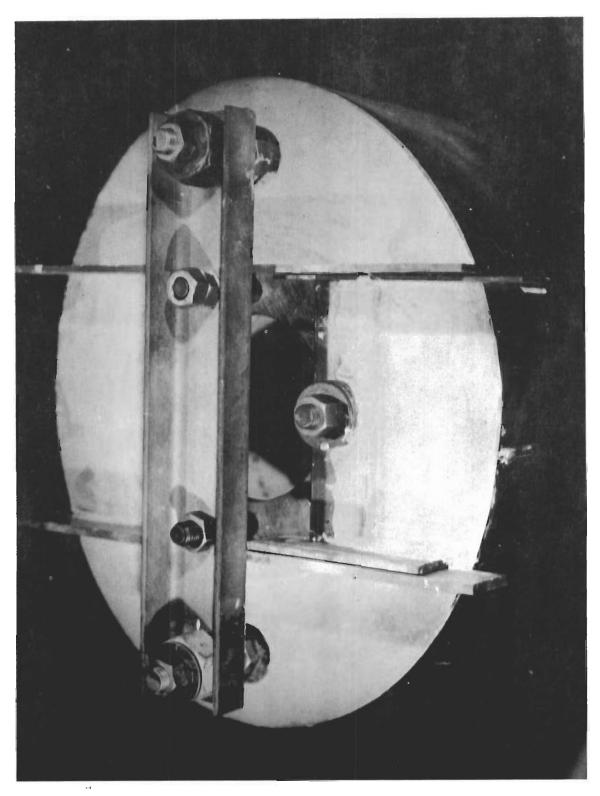


FIGURE 30. 500 square inch Model Nose Cone After Casting

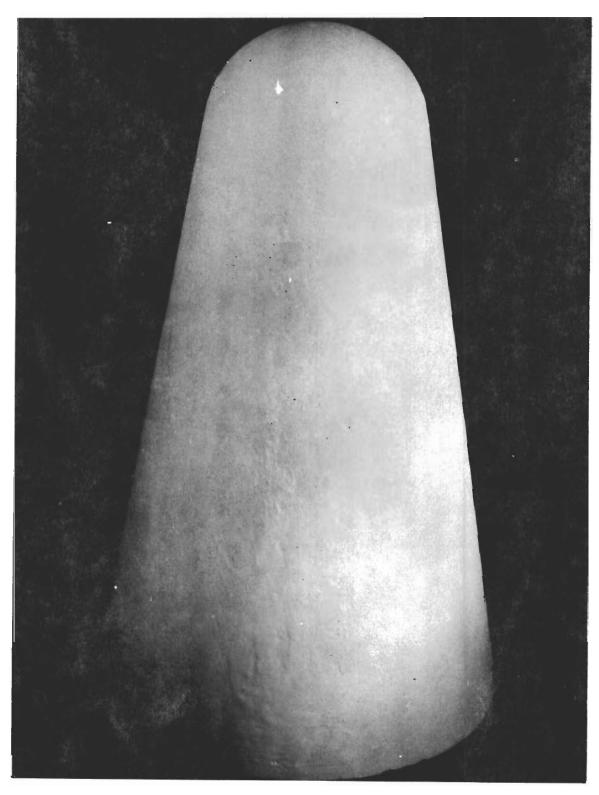


FIGURE 31. 500 square inch Model Nose Cone

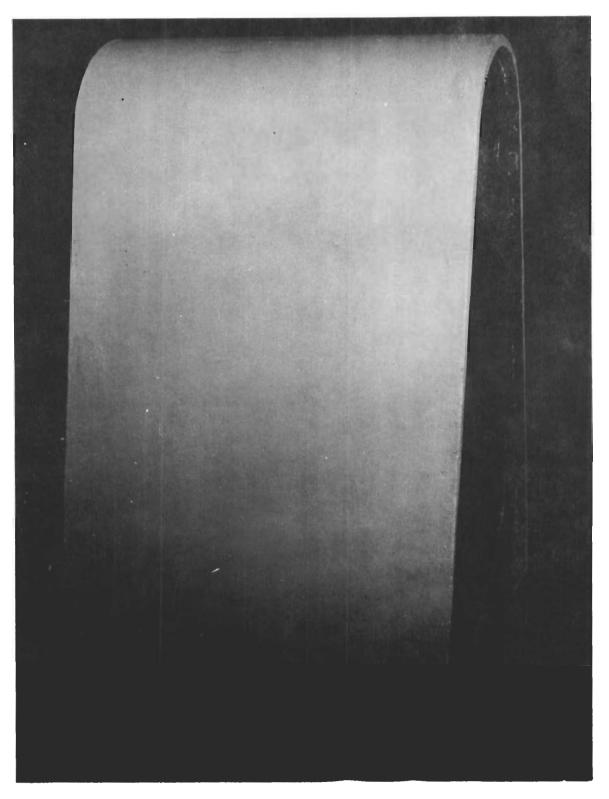


FIGURE 32. 500 square inch Model Leading Edge

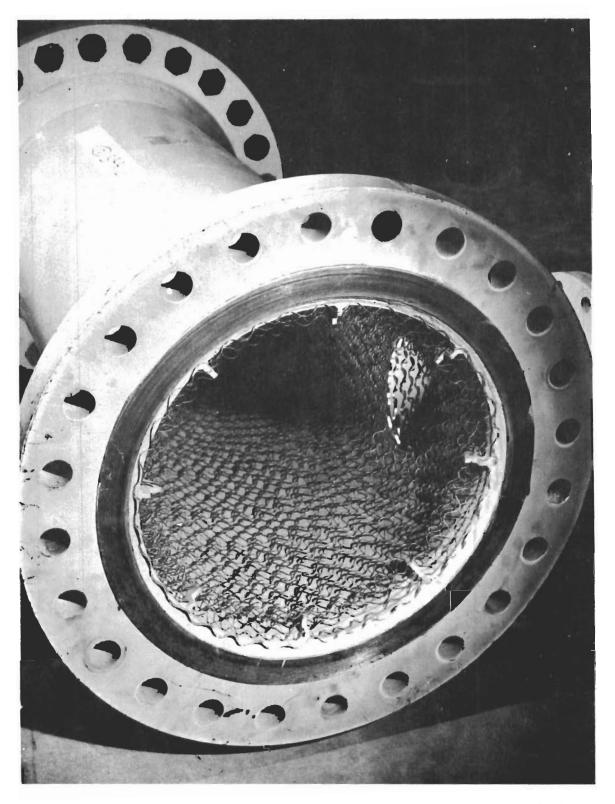


FIGURE 33. Bottom Center Section Showing Reinforcement before Coating

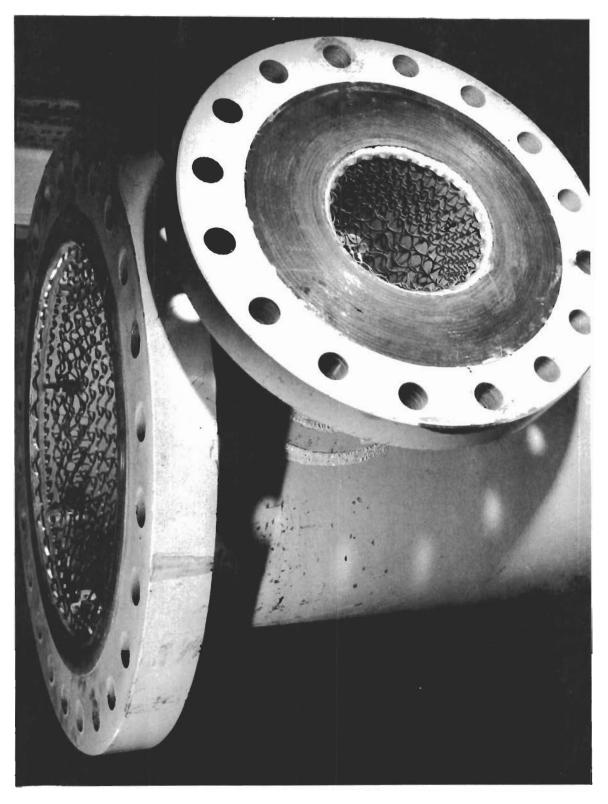


FIGURE 34. Upper Center Section Showing Reinforcement before Coating

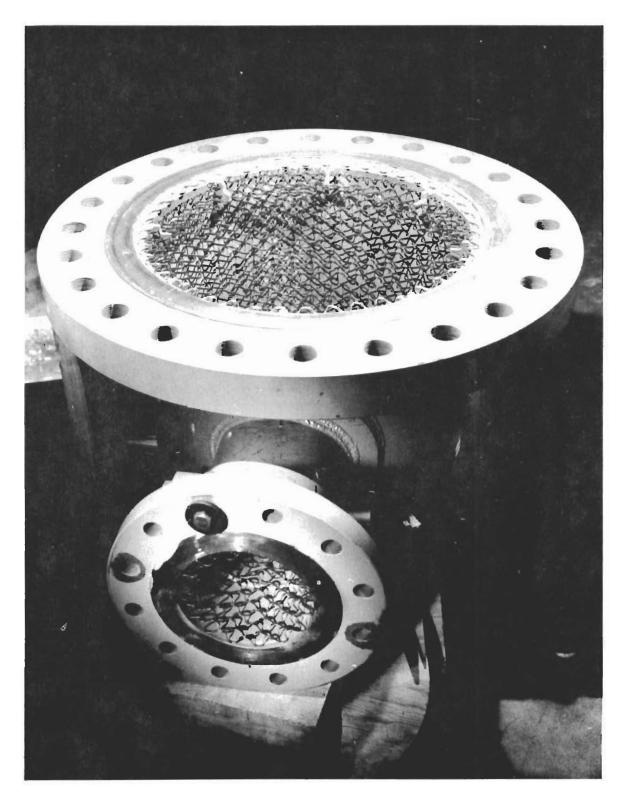


FIGURE 35. Bottom Section with Reinforcement before Coating

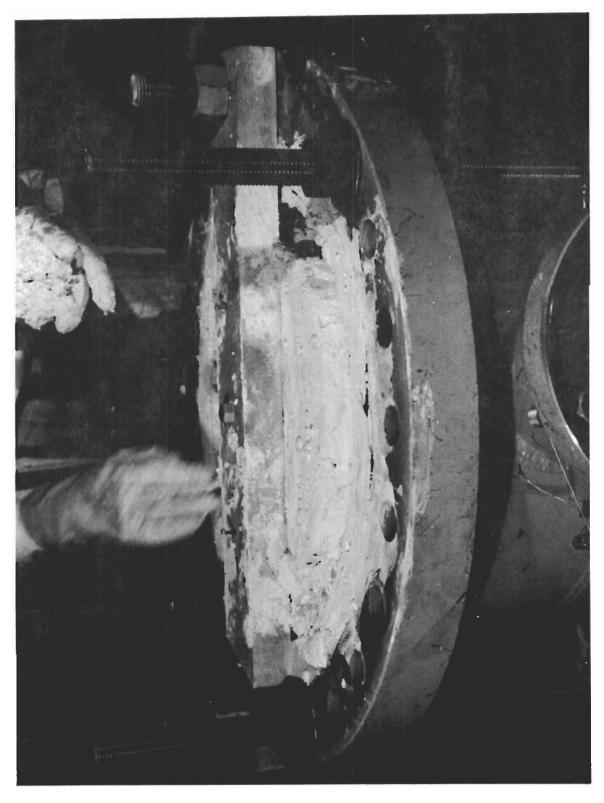


FIGURE 36. Top Dome During Coating Operation

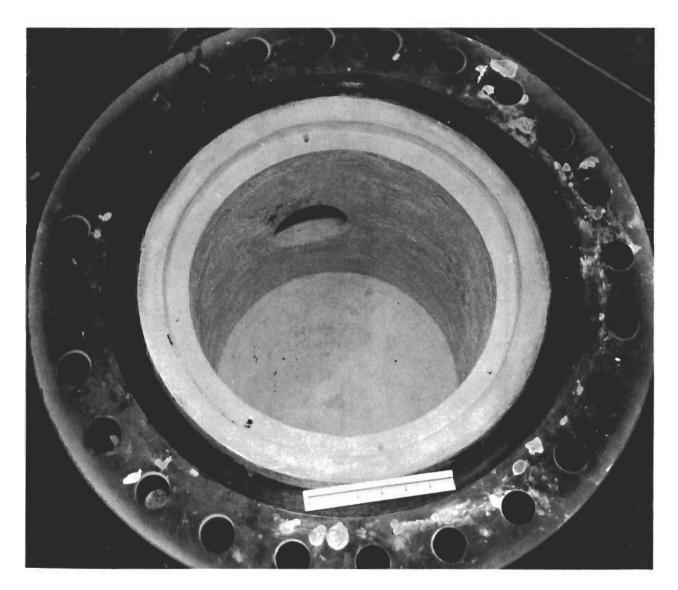


FIGURE 37. Top Dome after Coating

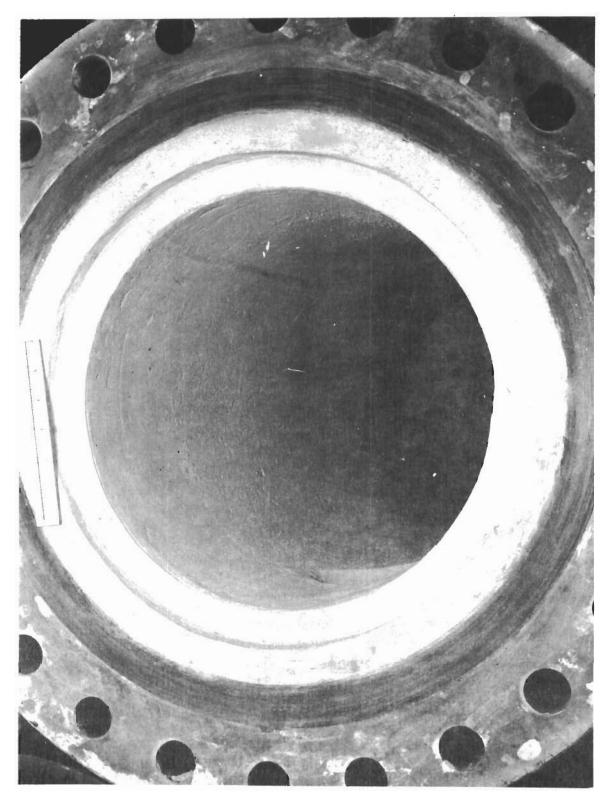


FIGURE 38. Upper Center Section after Coating

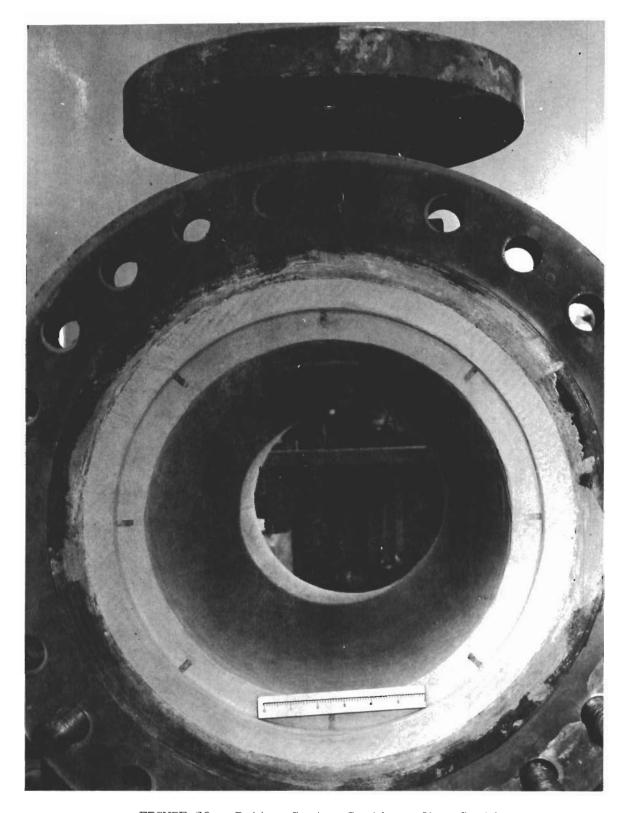


FIGURE 39. Bottom Center Section after Coating

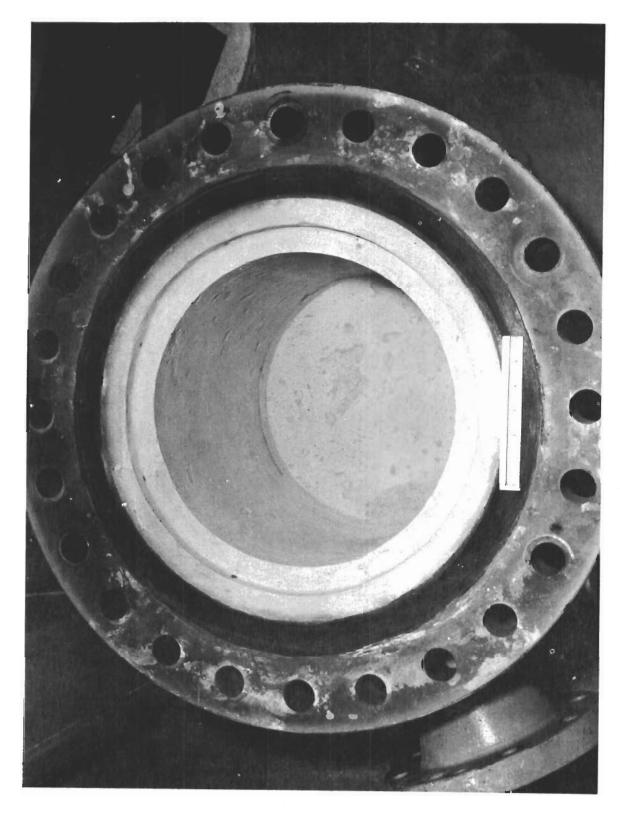


FIGURE 40. Coating in Bottom Section

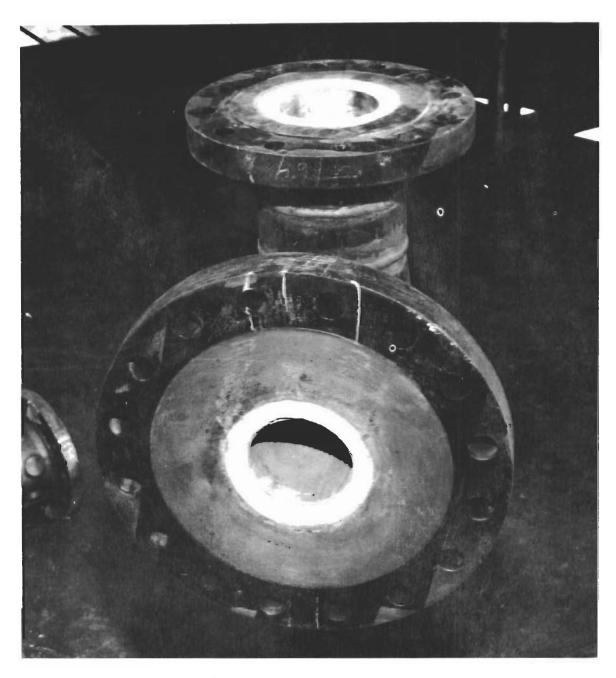


FIGURE 41. Accessory Pipe Ready for Service

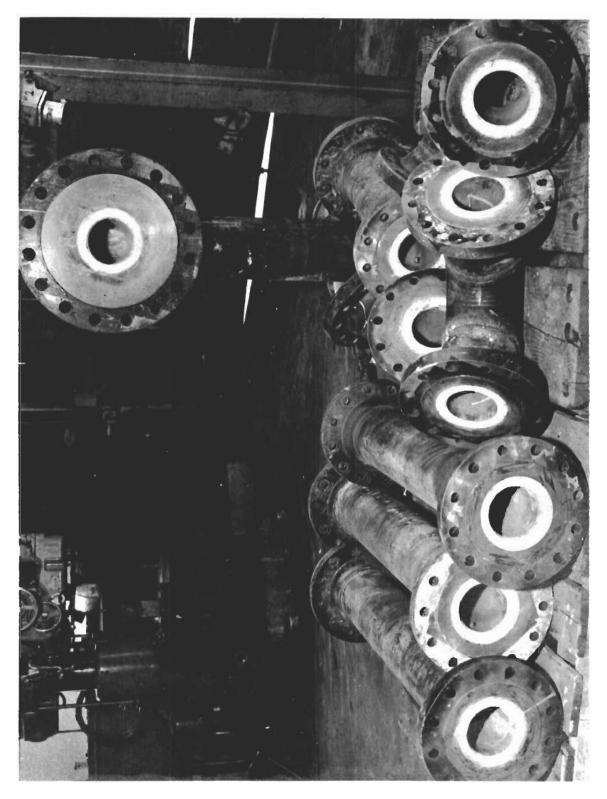


FIGURE 42. Accessory Pipes after Coating

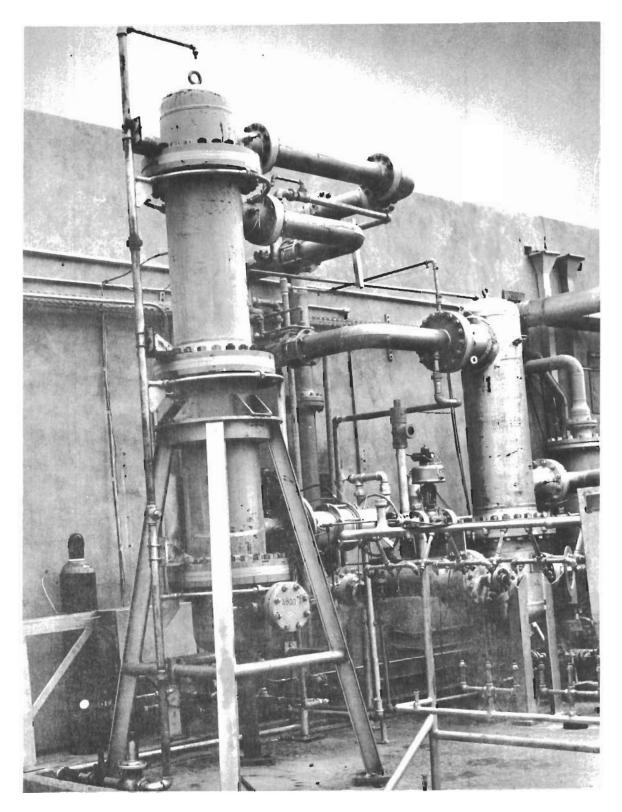


FIGURE 43. Heat Exchanger Assembly



APPENDIX I

SOURCE DATA FOR MATERIALS DISCUSSED IN THIS REPORT



APPENDIX I SOURCE DATA FOR MATERIALS DISCUSSED IN THIS REPORT

No.	Material	Vendor
1.	Tantalum ribbon (0.010 inch thick by 0.125 inch wide)	Kawecki Chemical Co.
2.	0.5% Ti-Mo ribbon (0.010 inch thick by 0.125 inch wide)	Fansteel Metallurgical Corporation
3.	321 stainless steel ribbon (0.010 inch thick by 0.125 inch wide)	H. K. Porter Co., Inc.
4.	N-155 alloy sheet	Haynes Stellite (Div. of Union Carbide Corp.)
5.	Hydrofluosilicic Acid (H ₂ SiF6)(Baker & Adamson)	Allied Chemical & Dye Corp.
6.	Hydrofluosilicic Acid (H ₂ SiF6)	J.T. Baker Chemical Co.
7.	Hydrofluosilicic Acid (H ₂ SiF6)	Davison Chemical Co.
8.	Hydrofluosilicic Acid (H ₂ SiF6)	The Harshaw Chemical
9.	Fluosulfonic Acid (HSO3F)	Allied Chemical & Dye Corp.
10.	Monofluorophosphoric Acid (H2PO3F)	Ozark-Mahoning Co.
11.	Fused zirconia - Norton H, 30F	Norton Company
12.	Fused zirconia - Norton H, 36F	Norton Company
13.	Fused zirconia - Norton I, 325F	Norton Company
14.	Fused zirconia - Zircoa B, 325 mesh	Zirconium Corporation of America
15.	Ammonium dihydrogen phosphate (NH4H2PO4)	(J.T. Baker Chemical Co. (Braun Chemical Co.
16.	Fiberfrax bulk fiber	The Carborundum Co.
17.	Chrome oxide (Cr ₂ O ₃)	Varlacoid Chemical Co.
18.	Cerium zirconate (CeZrO4)	B.F. Wagner & Co.



No.	Material	Vendor
19.	Strontium zirconate (SrZrO3)	B.F. Wagner & Co.
20.	Sodium silicate, "S-35" Brand	Philadelphia Quartz
21.	Potassium silicate-"Kasil"	Philadelphia Quartz
22.	Pyro-Chrome	Preferred Utilities Corp.
23.	CP-44 (Ceramic paint developed by the University of Illinois)	The Marquardt Corpora- tion
24.	Alkaphos C	Monsanto Chemical Co.
25.	Aluminum powder -325 mesh	Kaiser Aluminum Co.
26.	Brolite	Andrew Brown Co.
27.	Chromium powder -325 mesh	Kawecki Chemical Corp.
28.	Nicrobraz cement	Wall Colmonoy Corp.
29.	Columbium powder -325 mesh	Wah Chang Co.
30.	Vanadium powder -325 mesh	Vanadium Corporation of America
31.	Nickel powder -325 mesh	Metals Disintegrating Co.
32.	Nichrome powder -325 mesh	Metallizing Engineering
33.	Silver paint	Co., Inc. Sherwin-Williams
34.	Aluminum paint	Westinghouse Electric
35.	Alumina paint (Al ₂ O ₃ + H ₂ PO ₃ F)	The Marquardt Corpora- tion
36.	Phosphatherm	Alpha Molykote Corp.
37.	Zirconia paint (ZrO ₂ + H ₂ PO ₃ F)	The Marquardt Corporation
38.	Zirconium sulfate (Zr(SO4)2 · 2 H2O)	The Carborundum Co.
39.	Zirconium tetrafluoride (ZrF4)	Allied Chemical, Baton Rouge Dev. Lab.



No.	Material	Vendor
40.	Hydrofluoric acid (HF)	Mallinckrodt Chemical Works
41.	Ammonium fluoride (NH _{l4} F)	Allied Chemical & Dye Corp.
42.	Ammonium chloride (NH ₁₄ Cl)	J.T. Baker Chemical Co.
43.	Ammonium vanadate (NH ₄ VO ₃)	J.T. Baker Chemical Co. and Fisher Scientific Co.
44.	Chromium fluoride (CrF ₃ · 3 1/2 H ₂ 0	Allied Chemical & Dye Corp.
45.	Zirconium powder - 200 mesh	Zirconium Corporation of America
46.	Lithium zirconate (LiZrO3)	Lithium Corporation of America, Inc.
47.	Ammonium silicofluoride (NH ₄) ₂ SiF ₆	American Agricultural Chemical Company
48.	L-1155 wetting agent	Minnesota Mining & Manufacturing Co.
49.	L-1159 wetting agent	Minnesota Mining & Manufacturing Co.
50.	Manganese dioxide (MnO ₂)	Mallinckrodt Chemical Works & Allied Chemical and Dye Corp.
51.	Cobaltic oxide (Co ₂ O ₃)	Harshaw Chemical Co. and L.H. Butcher Co.
52.	Iron (Ferric) oxide (Fe ₂ 0 ₃)	Harshaw Chemical Co.
53•	Holmium oxide (Ho ₂ 0 ₃)	Michigan Chemical Corp.
54.	Gadolinium oxide (Gd ₂ 0 ₃)	Michigan Chemical Corp.
55.	Erbium oxide (Er ₂ 0 ₃)	Michigan Chemical Corp.
56.	Polyvinyl chloride resin	Escambia Chemical Corp.
57 •	Lanthanum oxide (La ₂ 0 ₃)	Vitro Chemical Co.
58.	Tungstic anhydride (WO3)	Fisher Scientific Co.



No.	Material	Vendor
59.	Zirconium oxide - E-220 (ZrO ₂)	Harshaw Chemical Co.
60.	Colloidal zirconia - #180	E.I. DuPont de Nemours
61.	Tin flake	Metals Disintegrating Co., Inc.
62.	Aluminum flake	Metals Disintegrating Co., Inc.
63.	KEL-F soft wax, Grade 40	E.I. DuPont de Nemours



APPENDIX II

BIBLIOGRAPHY OF REINFORCED REFRACTORY CERAMIC COATINGS AND RELATED ART

APPENDIX II

BIBLIOGRAPHY OF REINFORCED REFRACTORY CERAMIC COATINGS AND RELATED ART

B-I. REINFORCED REFRACTORY CERAMIC COATINGS

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No. 26 - 1947

- 10. Battelle Memorial Institute Report No. 14, "Gypsum or Modifications of Gypsum as a Combustion Chamber Liner for Special Short-Term Rocket Motors", Slyh, Schofield, Austin, 1947.
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12. Battelle Memorial Institute Reports, "Refractory Oxides as Liners for Combustion Chambers of Rocket Motors", Slyh, Schofield, Austin:

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APPENDIX III

TABULATION OF COMPOSITIONS



FORMULATION OF	TherMaro KFS-	COMPOSITIONS
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	ORMULA'	LION OF	There							
Constituent	Parts by Weight									
Competitutent	101	102	103	104	105	106				
ZrO ₂ (30m) Norton H	40	40	40	40	40	40				
ZrO ₂ (325m) Norton I	30	30	30	30	30	25.9	ļ			
NH4H2PO4	1	1		1	1		-			
(NH4)2S1F6			:							
H ₂ F0 ₃ F					2.5					
H ₂ SiF6										
H ₂ O	5	5	2.5	14	6	5.1				
HSO3F	5	6		5	2.5					
HF			2.5							
H2SO4			6.1							
Zr(SO4)2 · 4H2O						11.4				
ZrFų						2.1				
Zr (200m)										
(NH ₄) ₂ CO ₃										
NHцF										
nh ₄ c 2										
Cr ₂ O ₃										
NH ₄ VO ₃										
CrF3 · 3.5 H2O										
Sr2r03										
Cr (325m)		·								
ZrO ₂ (60 x 150)									ļ	
Lithium Zirconate										
Fiberfrax										
CeZrO ₄										
Co203										
La ₂ 03										
Mn02										
W03										
ZrO ₂ (36m)										
ZrO ₂ (325m, Zircoa)										
L 1155										
L 1159										
Fe ₂ 0 ₃										
Ho ₂ O ₃										
Gd ₂ O ₃						•				
Er ₂ 0 ₃										



F	ORMULAT	TION OF	TherMa	arq ZPF	- COMPO	SITIONS	3			
					irts by					
Constituent	100	101	102	103	104	105	106	107	108	
ZrO ₂ (30m) Norton H	40				40	40	40	50	50	
2r02 (325m) Norton I	30	20	17.5	17.5	27.5	25	20	17.5	20	
NH4H2F04	1	9	9	9	1	1	1	1	1	
(NH4) ₂ 81F6		- -								
H ₂ F0 ₃ F	5	5	5	5	5	5	5	5	5]
H2SiF6										
H2O	2	3	3	3	2	2	2	2	2	
HSO3F										
HF					- -					
H ₂ SO ₄										
Zr(SO ₄) ₂ · 4H ₂ O										
ZrF4										
Zr (200m)										
(NH4)2 CO3									- -	
инце										
NH ₄ C ℓ										
Cr ₂ O ₃			'							
ин4403										
CrF3 · 3.5 H20						- -				
SrZr03										
Cr (325m)			2.5		2.5	5	10	2.5		
z_{ro_2} (60 x 150)		80	80	80						
Lithium Zirconate				2.5						
Fiberfrax										
CeZrO ₄										
co ₂ 0 ₃										
La ₂ 03										
MnO ₂										
WO3										!
2r0 ₂ (36m)										
ZrO ₂ (325m, Zircoa)										
L 1155										
L 1159										
Fe ₂ 0 ₃										
Ho ₂ 0 ₃										
Gd ₂ O ₃										
Er ₂ 0 ₃								<u> </u>		L



FORMULATION	OF	TherMarq	ZSF-	COMPOSITIONS
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FORMULATION OF TherMarq ZSF- COMPOSITIONS Parts by Weight												
Constituent			T						· ·	T		
	101	102	103	104	105	106	107	108	109	110		
ZrO ₂ (30m) Norton H	40	40	40	48	40	40	40	40	40	40		
ZrO ₂ (325m) Norton I	30	30	30	33	30	30	30	30	30	30		
NH4H2PO4	1				- <i>-</i> -							
(NH4)2S1F6												
H ₂ P0 ₃ F												
H ₂ S1F6	7	7	8	9	7	7	7	7	7	7		
H ₂ O												
HSO3F												
HF												
H ₂ SO ₄												
Zr(SO ₄) ₂ · 4H ₂ O												
ZrF4								-				
Zr (200m)				10								
(NH4) ₂ CO ₃					1.4							
инць						1		2				
NH ₄ C 2	- -						1	-	2			
Cr ₂ O ₃										1		
NH ₄ VO ₃												
CrF3 · 3.5 H20												
SrZrO3												
Cr (325m)												
Zro ₂ (60 x 150)												
Lithium Zirconate												
Fiberfrax												
CeZrO ₄	- -											
Co203												
La ₂ 03												
MnO ₂												
W03												
ZrO ₂ (36m)												
ZrO ₂ (325m, Zircoa)												
L 1155												
L 1159												
Fe ₂ 0 ₃												
Ho ₂ O ₃												
Gd ₂ O ₃ Er_O-												
Er ₂ 0 ₃		<u> </u>		·		<u> </u>	L	<u> </u>	<u> </u>	<u> </u>		



FORMULATION	OF	TherMarq	ZSF-	COMPOSITIONS

	ORMULA									
Constituent	111	112	113	11 ¹ 4	arts by	116	117	118	119	120
ZrO ₂ (30m) Norton H	40	40	40	40	40	80	¥O	40	40	40
ZrO ₂ (325m) Norton I	30	30	30	30	22	5	26.5	26.5	26.5	26.5
NHLH2POL		- -		1	ı	9				
(NH ₄) ₂ S1 F 6										
H ₂ PO ₃ F				7						
H ₂ SiF6	7	7	7		6.2	6.2	6.2	6.2	6.2	6.2
H ₂ O			3							
HS03F										
HF										
H ₂ SO ₁ ,										
Zr(SO4)2 · 4H2O										
ZrF4			- <i>-</i>	- -						
Zr (200m)										
(NH4)2 CO3										
nhip										
nh _{li} c ?										
Cr ₂ O ₃					8	15				
NH ₄ VO ₃	1								1.75	
CrF ₃ · 3.5 H ₂ O		1			;					
SrZrO3			1					1.75		
Cr (325m)								~ ~	- -	
2r0 ₂ (60 x 150)										
Lithium Zirconate										
Fiberfrax			1/16" layer							
CeZrO _l							1.75			
Co203										3.5
Le ₂ 03						·				
MnO ₂										
W03										
ZrO ₂ (36m)										
ZrO ₂ (325m, Zircoa)										
L 1155]							
L 1159										
Fe ₂ 0 ₃										
Ho ₂ O ₃										
Gd ₂ O ₃										
Er ₂ ⁰ 3	<u> </u>	<u> </u>								



Ţ	ORMULAI	ION OF	TherMa	ara ZSF-	COMIPO	SITIONS	5			
					arts by					
Constituent	121	122	123	124	125	126	127	128	129	
ZrO ₂ (30m) Norton H	40)†O	40	40	40	49	40	40	40	
%r0 (325m) Norton I	26.5	26.5	26.5	30	30	30	30	30	30	
NHth510t				1	1	1	1	l	1	
(NH4) ₂ S1F6										
H ₂ F0 ₃ F										
H2S1F6	6.2	6.2	6.2	7	13	5	5	14	3	
H ₂ O							1.7	2.3	3.3	
н503г										
HF										
H ₂ SO ₄										'
Zr(SO ₄) ₂ • 4H ₂ O										
ZrFi										
Zr (200m)		. 								
(NH4) ₂ CO ₃										
ин _ц г										
nh _h c l										
Cr ₂ 0 ₃										
NH ₄ vo ₃										
CrF3 · 3.5 H2O		~-								
Sr2r03										
Cr (325m)										
ZrO ₂ (60 x 150)										
Lithium Zirconate										
Fiberfrax										
CeZrOl										
Co ₂ O ₃	. :									
La ₂ 0 ₃	1.75	 1 P5								
MnO ₂		1.75	1				- -			
W03			1.75							
ZrO₂ (36m) ZrO ₂ (325m, Zircoa)										
_									<u></u>	
L 1155]				<u>'</u>					
L 1159										
Fe ₂ 0 ₃	<u></u>								<u> </u>	
Но ₂ 03 Gd ₂ 03										
Er ₂ 0 ₃										
2 3	<u> </u>	<u> </u>	<u> </u>		<u> </u>	<u> </u>	L	L	1	



TOTAL TO A TOTAL ON	\triangle	ThorMora	7779_	COMPOSITIONS
FORMULATION	OF.	Thermara	Z#5-	COMPOSTITONS

	Parts by Weight									
Constituent	170	171	170		134	135	136	137		
70 /70) 77	130 40	131	132 40	133	40	40	40	40		
ZrO ₂ (30m) Norton H ZrO ₂ (325m) Norton I	30		30	30	30	30	30	30		
NH4H2PO4	1	1	1	1	1	1	2	3		
(NH4)2S1F6										
H ₂ P0 ₃ F										
H ₂ SiF6	2	7	5	6	7	8	6	6		
Н20	4		1							
HSO3F										
нь										
H ₂ SO ₄										
Zr(SO ₄) ₂ · 4H ₂ O							 - -			
ZrF4										
Zr (200m)									ļ	
(NH4)2 CO3										
NHLF										
nh _l c 2			- -							
Cr ₂ 0 ₃										
NH ₄ VO ₃										
CrF3 · 3.5 H20										
Sr2r03										
Cr (325m)										
ZrO ₂ (60 x 150)										
Lithium Zirconate										
Fiberfrax										
CeZrO ₄										
ია ₂ 0 ₃ ლაეკ										
La ₂ 05									[
MnO ₂										
W03										
ZrO ₂ (36m)		40								
ZrO ₂ (325m, Zircoa)		30								
L 1155										
L 1159										
Fe ₂ 0 ₃										
Ho ₂ 0 ₃										
Gd ₂ 0 ₃										
Er ₂ 0 ₃										



FORMULATION	OF	TherMarq	ZSF-	COMPOSITIONS
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Constituent ZrO ₂ (30m) Norton H ZrO ₂ (325m) Norton I NH4H2FO4 (NH4)2SiF6 H2FO3F H2SiF6 H2O	138 40 30 4.0 6.0 3.0	139 40 30 5.0 6.0 3.0	140 40 30 0.5	141 40 30 1.0	142 40 30 	143 40 30 1.0	144 40 30 1.0	145 40 30 1.0	146 40 30 1.0	147 40 30
ZrO ₂ (325m) Norton I NH4H2PO4 (NH4) ₂ SiF6 H ₂ PO ₃ F H ₂ SiF6 H ₂ O	40 30 4.0 6.0 3.0	40 30 5.0 6.0	40 30 0.5	40 30 1.0	40 30	40 30	40 30	40 30	40 30	40 30
ZrO ₂ (325m) Norton I NH4H2PO4 (NH4) ₂ SiF6 H ₂ PO ₃ F H ₂ SiF6 H ₂ O	30 4.0 6.0 3.0	30 5.0 6.0	30 0.5	30 1.0	30 	30	30	30	30	30
NH ₄ H ₂ PO ₄ (NH ₄) ₂ SiF ₆ H ₂ PO ₃ F H ₂ SiF ₆ H ₂ O	4.0 6.0 3.0	5.0 6.0	0.5	1.0			· -	-		1
(NH4) ₂ 81F6 H ₂ F0 ₃ F H ₂ 81F6 H ₂ 0	6.0 3.0	 6.0	0.5	1.0		1.0	±•∪	1.0		1.0
H ₂ F0 ₃ F H ₂ S1F6 H ₂ O	6.0 3.0	 6 . 0								
н ₂ sif6 н ₂ o	6.0 3.0	6,0	7.0		3.5	3.5	3.0	1.0 3.0	1.0 3.0	3.5
H ₂ 0	3.0 		7.0							
		3.0	' ' '	7.0	3.5	3.5	3.0	3.0	3.5	3.5
l I									1.4	1.0
HSO3F										
нг										
H ₂ SO ₄										
Zr(SO ₄) ₂ · 4H ₂ O										
ZrF4										
Zr (200m)										
(NH4) ₂ CO ₃										
NHцF										
NH ₄ C ℓ										
Cr ₂ O ₃										
NH ₄ VO ₃										
Crf ₃ · 3.5 H ₂ O										
SrZrO3										
Cr (325m)										
2r0 ₂ (60 x 150)										- -
Lithium Zirconate										
Fiberfrax										
CeZrO ₄										
^{Co} 2 ^O 3										
La ₂ 03										
MnO ₂										
W03										
ZrO ₂ (36m)										
ZrO ₂ (325m, Zircoa)										
L 1155										
L 1159										
Fe ₂ 0 ₃										
Но203										
Gd ₂ 03										
Er ₂ 0 ₃								. 		



F	FORMULATION OF TherMarq ZSF- COMPOSITIONS								
	Parts by Weight								· · · · · · · · · · · · · · · · · · ·
Constituent	148	149	150	151	152	153	154	155	156
ZrO ₂ (30m) Norton H	40	40	40	40	40	40	40	40	40
Zro (325m) Norton I	30	28	28	28	30	30	30	29.5	29.5
NH4H2PO4	1.0	1.0	1.5	1.0	1.0	1.0	1.0	1.0	1.0
(NH4)2S1F6	1.0	1.0	1.0	1.0	1.0	1.0	1		
H ₂ PO ₃ F	3.0	3.0	3.0	3.0	3.0	3.0	3.0		
H2S1 F 6	3.0	3.0	3.0	3.0	3.0	3.0	3.0	7.0	7.0
H ₂ O	0.7	1.0	0.5	1.0	3.0	3.0	2.5		
hso3 f									
HF									
H ₂ SO ₄						:			
Zr(SO4)2 · 4H2O									
ZrFų									
Zr (200m)									
(NH4)2 CO3]
nнµf									
NH ₄ C2									
Cr ₂ 03	0.5	1.0	1.0	0.5	0.5	0.5	0.5	0.5	
NH4V03									
CrF3 · 3.5 H2O									
SrZr03	i								
Cr (325m)									
Zro ₂ (60 x 150)									}
Lithium Zirconate									
Fiberfrax									
CeZrO ₄									
co ₂ 0 ₃									
Ia ₂ 0 ₃									
MnO ₂									0.5
WO3									
ZrO ₂ (36m)									
ZrO ₂ (325m, Zircoa)									
L 1155]		0.03				
L 1159						0.03			
Fe ₂ 0 ₃									
Ho ₂ 0 ₃									
Gd ₂ 0 ₃									
Er ₂ 0 ₃								L <u></u>	



FORMULATION	OF	TherMarq	ZSF-	COMPOSITIONS

FORMULATION OF TherMarq ZSF- COMPOSITIONS										
Constituent		Parts by Weight								
	157	158	159	160	161	162	163	164	165	
ZrO ₂ (30m) Norton H	40	40	40	40	40	40	40	40	40	
ZrO ₂ (325m) Norton I	28.5	28.5	28.5	28.5	28.5	29 .	29	29	28	
NH ₄ H ₂ PO ₄	1.0	1.0	1.0	1.0	1.0	1.0	1.0	1.0	1.0	
(NH4)2S1F6										
H ₂ P0 ₃ F										
H2SiF6	7.0	7.0	7.0	7.0	7.0	7.0	7.0	7.0	7.0	
H ₂ O						1.0				
HSO3F										
HF				:						
H ₂ SO ₄										
Zr(SO4)2 · 4H2O										
ZrF4										
Zr (200m)										
(NH ₄) ₂ CO ₃										
NH ₄ F										
NH ₁₄ C ℓ										
Cr ₂ 0 ₃										
NH4VO3										
CrF ₃ · 3.5 H ₂ O										
SrZrO3						1.0				
Cr (325m)										
ZrO ₂ (60 x 150)										
Lithium Zirconate										
Fiberfrax							1.0			
CeZrO ₁	0.5							1.0	2.0	
್ದಾರಿ ₃ La203			l							
MnO ₂										
W03 Zr02 (36m)										
ZrO ₂ (325m, Zircoa)										
L 1155										
L 1159										
Fe ₂ 0 ₃	0.5									
Ho203		0.5	-							
Gd ₂ O ₃			0.5							
Er ₂ 0 ₃				0.5			•			