WADC TECHNICAL REPORT 55-381

POLY FBA A FLUORINATED ACRYLIC ELASTOMER FOR HIGH TEMPERATURE SERVICE IN THE PRESENCE OF AIRCRAFT FUELS AND LUBRICANTS

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FOREWORD

This report was prepared by the Organic Materials Branch and was initiated under Project No. 7340, "Rubber, Flastic and Composite Materials", Task No. 73405, "Compounding of Elastomers", formerly RDO No. 617-12, "Compounding of Elastomers", and was administered under the direction of the Materials Laboratory, Directorate of Research, Wright Air Development Center, with Major Horace C. Hamlin acting as project engineer.

This report covers the period of work from December 1951 to August 1955.



ABSTRACT

A large number of fluorine containing polymeric systems have been investigated by the Minnesota Mining & Manufacturing Company under Air Force Contract No. 33(038)-515. One of the most successful developments, poly 1.1 dihydroperfluorobutyl acrylate (poly FBA) exhibits good rubbery characteristics, excellent resistance to many fuels, lubricants, solvents, chemicals, and ozone, plus very good stability at elevated temperatures. Compounding and processing studies have been made, and tests conducted in various media at temperatures up to 550°F. Results of laboratory and simulated service tests show poly FBA to be vulcanizable through the action of certain metal oxides or organic amines, reinforcible with carbon blacks and some inorganic fillers, and readily handled on standard rubber processing equipment. Tensile strength ranges from 1000 to 1400 pounds and elongation from 200 to 400%. It is highly resistant to the effects of hydrocarbon fuels, even at elevated temperatures, and has been successfully tested in the presence of synthetic lubricants for up to 300 hours at 400°F, and for shorter periods at higher temperatures.

Poly FBA is also highly resistant to the chemical action of fuming nitric acid, although quite permeable to it. It is completely resistant to czone. Low temperature properties are limited to about O°F unplasticized, although it retains good elasticity almost down to the brittle point.

PUBLICATION REVIEW

This report has been reviewed and is approved.

FOR THE COMMANDER:

M. R. WHITMORE

Technical Director Materials Laboratory

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There



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Developments in military aircraft and aeronautical science occurring during the past decade have resulted in increasingly severe material problems. This is particularly true in the field of elastomeric materials. Rubbery materials capable of functioning over a very wide temperature range and highly resistant to a variety of fuels, lubricants, solvents, chemicals, and ozone are now urgently required.

The purpose of the work herein reported has been threefold namely; to maintain more intimate contact with Minnesota Mining & Manufacturing Company's (3M's) work; to conduct additional developmental compounding studies supplementing like studies by the contractor; and through compounding studies to seek additional desirable properties of interest to the Air Force. For fundamental characteristics WADC Technical Report 52-197, Parts 2, 3, 4, and 5 should be consulted. These reports were prepared by Minnesota Mining & Manufacturing Company (Central Research Department) under contract and are available to qualified organizations from the Armed Services Technical Information Agency, Document Service Center, Knott Building, Dayton, Ohio.

Several samples of poly 1,1 dihydroperfluorobutyl acrylate (poly FBA) made with slightly varying procedures have been received at Wright Air Development Center from 3M. All have been compounded and processed successfully without special difficulty, although samples which were alum coagulated contained a little more ash and were slightly more acidic requiring small adjustments in vulcanizing recipes.

Due to the relatively small quantities of poly FBA available for compounding, special small 2-1/8"x 3-1/8"x .040" sheet molds were made and compounding was done on either a 3"x 3"x 8" mill with adjustable guides or on a 1-1/4"x 1-1/4"x 4" mill also with adjustable guides. Tensile specimens were cut with a miniature die 2" in overall length with 1/2" wide tabs and a 1/2"x 1/8" restricted portion. Specimens were elongated at various rates with a rate of 12" per minute selected finally as a suitable speed at which modulus could also be read fairly accurately. Values obtained at this rate were found to be little, if any, different from those read at a rate of 20" per minute. It is likely that the miniature specimens produced values somewhat exceeding those which might be obtained with conventional samples, but such has not yet been demonstrated.

Foly FBA, uncompounded, is a translucent white snappy rubber which bands readily at processing temperatures around 140°-150°F and requires little breakdown. During compounding it has a strong tendency to adhere to both rolls. On completion of the process, however, cool rolls and proper handling permit relatively easy release and sheeting off. Fatty acids, such as stearic acid (1-2%), tend to reduce sticking and aid processing.



EARLY COMPOUNDING STUDIES

The earliest successful vulcanizing systems for poly FBA were devised at 3M and were based on a combination of sodium metasilicate nona hydrate and calcium hydroxide or on various metal oxides, the best of which were Litharge (PbO) and Magnesia (MgO). Magnesia in particular caked severely on the rolls unless great care was taken. Cool rolls and slow addition seemed to be the best preventative. At WADC other types of vulcanizing agents were tested, and it was found that Lead Feroxide was highly reactive and was accelerated by Folyac, although Folyac had no activity alone or with Litharge or Magnesia. Some recipes and typical properties are shown in Appendix I, Table 1. It was subsequently determined that compounds of equal quality could be produced with greatly reduced levels of both filler and vulcanizing ingredients (See Table 2). It is apparent from both tables, however, that aging stability, elasticity, and set (compression set and set at break) are far too high for a good elastomer.

A series of blends of other polymers was then tried. Three (3) examples are shown in Appendix I. Table 3. Only the Hypalon S-2 blend showed improved strength, and the volume swell of all samples in 70/30 Isooctane/Toluene test fluid was unsatisfactory. It is quite possible that actual blends were not obtained and the process resulted in mere mixtures.

Additional basic compounding studies conducted at 3M revealed that oxide vulcanizing agents for FBA were accelerated by some common rubber accelerators. Outstanding among these were zinc diethyldithic carbamate. Further studies at WADC showed that by the use of small quantities of this accelerator coupled with excellently dispersed Magnesia (added from a masterbatch) would produce a fairly well vulcanized rubber sample, provided the batch was partially set up in an oven (2 hrs at 300°F), then remilled and molded. The initial set-up provided sufficient cross linking to eliminate tendencies toward porosity during the slow curing reaction. A number of compounds of this type were made and partially evaluated. Of particular note is the reduction of swell on immersion in synthetic lubricant at 350°F, as well as the improvement in aged physical properties. It may be assumed that very small amounts of vulcanizing agents are adequate, provided they can be well enough dispersed and made to react adequately. The properties of such compounds are shown in Tables 5 and 6.

It was also found that similar procedures with Litharge were successful, the action of masterbatched fine Litharge being much faster than that of ELC Magnesia. Results are shown in Table 7.

One other vulcanizing agent, namely pulverized sodium hydroxide, was tried. It, somewhat surprisingly, proved very active and easy to employ, though not considered practical. Results of one recipe are shown in Table 8. No further work was done with it.

Although these compounds represent much improved properties over earlier ones, concurrent experimentation at both WADC and 3M resulted in the discovery that poly FBA could be vulcanized quite effectively by certain aliphatic poly amines to give compounds of equal fuel, oil, and solvent resistance, better heat resistance, much superior elasticity, more rapid cure, and lower compression set. Cyclic amines, such as hexamethylene tetramine, and aromatic amines, such as benzidine, showed much lower activity than the aliphatic amines and failed to vulcanize poly FBA.

SECTION II

COMFOUNDING STUDIES WITH AMINE VULCANIZING RECIFES

Triethylene tetramine and Trimene Base (a reaction product of ethyl chloride, formaldehyde, and ammonia) were first found to be very good cross linking agents for poly FBA. Of these, TETA (triethylene tetramine) produced better properties, its compounds being generally more thermally stable and more elastic than those using Trimene Base.

A large number of compounds have been made and tested and some of them are herein reported. These compounds were made from several experimental and pilot batches of poly FBA, which were produced by somewhat varied procedures. It is important to remember that such variations coupled with the very small samples employed may introduce variations in test results not accounted for by changes in recipe or test method. With the exception of variations already noted all tests were in conformity with ASTM Specifications, where applicable.

Compounding and processing techniques for carbon black reinforced, amine vulcanized poly FBA vary somewhat from previous methods. Roll temperatures remain about the same as with oxide loading (about 140°F). Carbon black, however, exhibits much less tendency to cake on the rolls and incorporates more readily. Generally, the polymer bands on both rolls almost immediately with some tendency to adhere to the hotter roll if there is a temperature differential. It also tends to adhere to the faster roll. Carbon black may be added slowly, almost immediately on banding, and should be continued until some band reinforcement has been achieved. Then add processing aids (stearic acid or paraffin) and sulfur, and the remainder of the filler, anti oxidant and other ingredients. After incorporation of all fillers, cool the rolls to a little above room temperature to facilitate blending and refining. Add amine slowly to prevent loss, since most amines are absorbed by poly FBA rather slowly. Final blending and sheeting is accomplished most easily on a cool (below RT) mill.

Among the variables tested in amine vulcanizing studies were processing aids. It was found that the replacement of one or two parts of stearic acid with paraffin wax resulted in a compound which hardened appreciably less on air oven aging. When employed in a later batch of poly FBA, which in itself had better aging characteristics, the paraffin compound actually became softer, a condition easily controlled with a little additional vulcanizing agent. Table 9 shows the effect of paraffin and demonstrates some typical poly FBA characteristics.

It was also noted that, as in other acrylics, sulfur, or a sulfur bearing agent is essential in developing optimum heat stability. Without sulfur, amine vulcanizates harden rapidly. It appears that about one part of sulfur represents an optimum amount since excess quantities are also detrimental. The data from Table 10 compared with Table 9 demonstrate this fact, with respect to original properties and even more noticeably in results of air and oil aging. Compound No. 456-91-1 in Table 11 shows the effect of two parts of Tetrone A replacing sulfur.

A number of amine containing agents have been tested as vulcanizing agents for poly FBA. None have been found which produced better properties than triethylene tetramine. Trimene Base is less active at low temperatures and is consequently less scorchy. Table 12 shows a brief evaluation of four other amines, which readily compound with, and vulcanize poly FBA. Of these, tetraethylene pentamine (TEPA) may be of value as a good vulcanizing agent with somewhat less low temperature reactivity than TETA. Further work with other amines is under way, but not completed.

Due, in part at least, to the alkaline nature of the vulcanizing system, the best reinforcing fillers for poly FBA seem to be furnace blacks ranging from neutral to highly alkaline pH. Fast Extruding Furnace (FEF) and High Abrasion Furnace (HAF) Blacks produce compounds with a good balance of mechanical properties exhibiting good stability under many conditions. Generally lower quantities of filler are required to produce optimum properties than with most other synthetic rubbers. Larger quantities of the coarser type fillers may be used. Little advantage is noted, however, except some reduction in volume cost of the compound and somewhat improved resilience. Table 13 shows poly FBA containing three loading levels of Medium Thermal (MT) Black. Ordinary acidic channel blacks interfere with the amine curing system in poly FBA and require additional amine to effect good cures. The high temperature aging qualities of such compounds tend to be poorer than with furnace blacks. A specially treated alkaline channel black, namely Texas 109 produced very good properties when compounded with poly FBA, and with the possible exception of shore hardness, exhibited high temperature aging properties almost equal to the furnace blacks. Table 14 shows the properties of one recipe containing alkaline channel black and one non black recipe containing Aerosil silica.

Conventional rubber antioxidants and accelerators produce a definite effect on poly FBA. The presence of small quantities of such materials noticeably enhances the retention of tensile strength after hot air and oil aging at the expense of reduced elongation and increased hardness. Table 15 shows the effect of Agerite Powder and Sulfasan R on original properties and after aging in air and synthetic lubricant at 350°F. Considerable additional work in this field has been carried out by the Minnesota Mining & Manufacturing Company and is detailed in their reports (previously referenced).

Although most of the work done on poly FBA has been pointed toward evaluation and improvement of properties in the presence of synthetic lubricants conforming to Specification MIL-L-7808 and other special fluids at very high temperatures, poly FBA has been shown to have excellent resistance to a variety of media over a wide temperature range. Table 16 shows its resistance to two types of hydrocarbon fuels and two lubricants at moderately elevated temperatures.

SECTION III

POLY FBA - COMPOUNDED SHELF LIFE

Manufacturers' literature has shown that other rubbery acrylic polymers undergo a change in properties if stored for an appreciable period subsequent to compounding and before curing. The change in the uncured compound consists mainly of some toughening due, no doubt, to cross linking. After curing the stored compound the changes are more noticeable and are, to some extent, dependent on time and conditions of storage. Increased precure storage results in cured compounds with increased elongation and decreased modulus and hardness. Tensile strength seems to be not greatly affected until after a considerable period, following which it too is reduced. Storage conditions of temperature and humidity appear to greatly affect the acceptable storage time for a given compound. Table 17 shows the effect on three (3) poly FBA compounds of precure storage up to three (3) days. This test was run in the summer under conditions of high temperature (900-1000F) and high relative humidity. The effect of storage is self evident. It is also evident that variation in the amine content may partially overcome the effect of storage time and conditions.

A similar, but more complete, test was subsequently run in the winter under conditions of lower temperatures (65-70°F), and much lower relative humidity. This test was run over a considerably longer period of time with somewhat similar, but much less spectacular results on physical properties. The test was continued to determine the effect of storage on both hot air and hot synthetic lubricant aging. Original physical properties and recipes are shown in Table 18. Results of air

oven aging at 350°F are shown in Table 19 and results of aging in synthetic lubricant at 350°F in Table 20. It is apparent that the conditions encountered in winter or in a properly controlled atmosphere and much less severe than those encountered during the previous test.

To obtain some indication of the effect of humidity alone, a brief additional test was run in which one sample was stored, before curing, in open atmosphere, and one was carefully wrapped in three (3) layers of polyethylene film. The results are shown in Table 21. This test was also run in the winter which probably reduced the difference in the results of open vs wrapped storage. Although the difference in results is slight, it appears that protection from moisture helps prevent changes in the compound during storage.

SECTION IV

FOLY FBA COMPOUNDS FOR HIGH TEMPERATURE SERVICE

Since poly FRA has shown promise for high temperature service under various conditions, some work was devoted to determining the approximate temperature limit and how to extend service life at high temperatures. Tables 22 through 26 show the effect of temperatures ranging from 350 °F to 550°F on poly FBA compounds. From the results it appears that 350°F is a very mild temperature for poly FBA for service in synthetic oil. As temperatures rise it appears that the slight tendency for hardness increase noted at 350°F is gradually reversed. This may be explained by slight additional cross linking which is overcome at higher temperatures, probably by chain scission and gradual decomposition. This effect is definitely retarded by the use of additional amine vulcanizing agent. Employment of such materials as Sulfasan R (previously discussed) should also help. Although properties degraded very rapidly at 500°F and above, some simulated service tests have indicated that poly FBA seals may be serviceable for considerably longer than indicated by the test data.

SECTION V

COMPRESSION SET OF POLY FBA

Compression set is an important characteristic of rubber materials intended for use as seals, diaphragms, etc. In this respect, also, poly FBA compounding is generally similar to that used for other acrylic rubbers. Factors affecting compression set are: type of filler used, period of rest prior to vulcanization, amount of filler used, amount of vulcanizing agent used, use of rubber antioxidant, and period of oven

tempering before testing. Tables 27, 28, and 29 show the effect of some of the above variables. All samples were oven tempered prior to testing and all test discs, except where noted, were molded to one half the thickness and one half the diameter of Standard ASTM compression set test plugs. In these tests it appeared that the small size of the plug and/or the nature of the test jig (unplated) used produced more severe results than similar tests run elsewhere. It should be noted that compression recovery at or above test temperature resulted in quite favorable results. Since this condition may frequently occur in actual applications, it is quite significant. It should also be noted that the test temperature used was 250°F rather than 212°F commonly employed with most other synthetic rubbers. The test results show that reduced filler loading and coarser particle fillers (FEF Black) are favorable for low compression set. Some additional vulcanizing agent is also desirable. TETA vulcanized compounds produce lower compression sets than otherwise similar compounds vulcanized with Trimene Base. The presence of rubber antioxidants may be undesirable.

In order to provide more complete and useful information from the results of compression set tests, these compounds were aged in air and synthetic oil at 350 °F. The results of aging are shown in Tables 30 through 32.

SECTION VI

RESISTANCE OF POLY FBA TO FUMING NITRIC ACID

The addition of fluorine to some other saturated organic compounds has resulted in improved resistance to red and white fuming nitric acid. Due to its saturated chemical composition and the presence of about 50% by weight fluorine, it is reasonable to suppose that poly FBA might withstand the action of the above acids quite well. Three (3) poly FBA compounds were immersed in Red Fuming Nitric Acid conforming to Specification MIL-N-7277 for 70 and 200 hours at room temperature. The results are shown in Table 33. All three compounds swelled at least 25% and suffered quite a reduction in physical properties, retaining, however, fairly reasonable tensile strength. It is of particular interest to note that on aging for 200 hours and then rinsing in water, virtually all of the swell was eliminated and physical properties returned practically 100%. This indicates that the loss of properties on immersion was due largely to the volume increase and there was little or no chemical degradation.

To determine the resistance of poly FBA to nitric acid or oxide penetration, a 15 mil film of compounded and cured poly FBA was tested in a standard H cell and the permeation measured by reduction in pH on the water side of the cell. The results are shown in Figure 1. Although chemically quite resistant to fuming nitric acid, poly FBA is quite transparent to it and forms only a poor barrier.



LOW TEMPERATURE PROPERTIES OF POLY FBA

For military applications low temperature properties of materials are of extreme importance. The low temperature properties of two (2) poly FBA compounds (Peroxide cured and amine cured) were measured by the Temperature-Retraction Test and by the ASTM D746-52T Brittle Point method. The Temperature-Retraction Test was run essentially in conformity to the ASIM D599-4C procedure except that the entire retraction curve was recorded and methanol was substituted for acetone as the immersion medium. The results are shown in Figure 2. Although it is apparent that poly FBA is not a good low temperature material, two features can be noted. From the steepness of the curves it can be observed that poly FBA retains most of its original elasticity right down to the brittle point, and from the relation of the left hand portion of the curves to the brittle point it seems that, for some applications at least, poly FBA may have sufficient flexibility to be used a little below the measured brittle point. Poly FBA has been successfully plasticized with fluorine containing plasticizers to flexibility as low as -65°F. All plasticizers tested, however. have been readily extracted by common fuels and lubricants.

SECTION VIII

POLY FBA - MOLDING AND EXTRUSION

Generally, neither molding nor extruding poly FBA presents any particular problems. Early compounds vulcanized with oxides, peroxides, etc. were slow to exhibit incipient vulcanization. This, coupled with the extra polymer breakdown resulting from dispersing high oxide loadings, caused the heated polymer to become a very soft runny mass when placed in a hot mold. There was little or no back pressure exerted and consequently many moldings were badly blown. Oven preheating to incipient cross linkage before molding alleviated this difficulty and permitted relatively easy production of good molded sheets.

Amine vulcanized compounds do not exhibit the above difficulty and can be press cured or autoclave cured very readily. For all moldings, fairly generous application of a mold release such as Dow Corning DC 35B is essential, although it need not be repeated after each pressing. Unchromed molds will be discolored by the action of triethylene tetramine and some other vulcanizing agents. The discoloration has at no time been shown to damage the mold; although, the deposit may build up to appreciable thickness after some time. It can very readily be removed with wet steel wool and pumice. There is considerable experimental evidence that the amine stain acts as a corrosion preventative for the steel.



Molding temperatures may run from 280°F to 340°F or above; although, the higher temperatures may cause difficulty with large moldings. Relatively low molding pressures may be used without difficulty.

Experimental extrusion of poly FBA was initially done with a small plunger operated extruder which was preheated to about 165°F. A 40-50 gm slug of compounded stock was placed in the plunger well, the plunger partially inserted, and the assembly placed in a power driven heated molding press. As the press was closed at uniform speed, the extruder plunger was forced downward and approximately a two foot length of extrusion produced. This device was resorted to for reasons of economy of material. The extruding device assembled and disassembled is pictured in Figures 3 and 4.

later extrusion tests with larger quantities of material were run on a No. 1/2 Royle rubber extruder. These extrusions were made very easily and many of the amine cured recipes herein reported have been successfully extruded. No extrusions were attempted with oxide cured stocks. All extrusions on the Royle extruder were run with a water cooled screw and a die heated to about 150-165°F. The extruder was otherwise uncontrolled. For long time continuous extrusions, the above figures will not necessarily hold true and extrusion conditions must be determined by experience. The extruder was operated from 1/2 speed to full speed (Linear extrusion rate not measured). Extrusion swell appeared to be minimized by faster extrusion rates.

Whereas in the plunger type extruder, only fresh stock could be smoothly extruded, in the Royle this problem was not particularly noticeable. The problem is believed to be due to the room temperature cross linking or scorching which occurs with TETA. It probably was reduced in the Royle extruder by the working which the stock received from the action of the screw.

Freshly compounded (or freshly extruded) poly FBA stocks are rather tacky and care should be exercised on handling fresh extrusions until scapstone can be applied. Extrusions can be readily cured in an autoclave at temperatures from 310°F to 340°F for 30' to 60'. Autoclave pans with a 1/2" or more layer of Talc or scapstone will prevent flattening one side of the extrusion.

A fifty (50) foot length of approximately .430" 0.D. poly FBA tubing was run off at WADC and covered with two (2) layers of stainless steel wire braid and vulcanized through the courtesy of the B. F. Goodrich Company. Sections of the hose thus produced were coupled by the Aeroquip Corporation. A section of the tubing and the hose are shown in Figure 5.

Sections of this hose were subsequently tested on the Materials Laboratory, WADC, hose tester. The test was conducted by vibrating the hose mounting 1800 times per minute at 1/2" amplitude while circulating MIL-L-7808 synthetic lubricant through the hose at 350°F and 75 pounds

pressure. Chamber temperature was maintained at 300°F. The test was carried on for over 300 hours and then discontinued without failure. Further hose development and testing is now being carried out by the Flexotube Division of the Meridan Corporation and the United States Rubber Company under Air Force Contract.

SECTION IX

POLY FBA LATEX

Although no work has been done at WADC with poly FBA Latex some studies carried out by 3M have shown poly FBA Latex capable of being compounded and processed without particular difficulty employing procedures and equipment generally used in handling latices.

SECTION X

SUMMARY

A new synthetic elastomer has been developed having a unique combination of fuel, oil, and solvent resistance and high temperature stability. The results of many tests suggest application in a variety of types of mechanical goods including seals, diaphragms, hose, coated fabrics, and other items.

For anyone interested in evaluating poly FBA or doing additional compound development, it is suggested that either Compound No. 257-86-1, described in Table 11 or Compound No. 257-18-3, described in Table 27 (if a soft stock is desired) exhibits a good balance of properties and makes a good starting point for further development.

Poly FBA is available in pilot quantities from the Minnesota Mining and Manufacturing Company, Fluorochemicals Department, St. Paul, Minnesota.

Poly FBA has been developed to the commercial stage and is finding increasing application in both military and non-military applications. Research and development work on improved materials is continuing with emphasis on promising polymers for equal chemical and high temperature stability and much improved low temperature flexibility.



TABLES

Table 1

Poly FBA - Properties of Peroxide-Polyac Cure

Compound No.	<u>769-5-1</u>	<u>769-6-1</u>	<u>769-6-2</u>	<u>769-6-3</u>	<u>772-78-1</u>
Poly FBA	100	100	100	100	100
Lead Peroxide	20	20	20	20	20
Polya c	3	3	3	3	3
EPC Black	10	•	-	_	SAF 20
FEF Black		10	30	30	
Cure	30'at 310°F	45'at 310°F	30'at 310°F plus 30'oven	35'at 310°F	60'at 310°F
100% Modulus Lbs/In ²	.	_	-	-	_
Tensile Streng	•				
Lbs/In ²	1320	1265	1310	1210	1256
Elongation, %	360	450	138	193	183
Set at Break					
% Hardness	55	32	12	14	32
Shore A					82
Vol. Increase 70 hrs in 70 Isooctane/To	luene				02
at RT	18.8%				14.4

Compression Set 70 hrs at 212°F 30% Compression 85.5%

Table 2

Poly FBA - Oxide Curing Recipes

-			
Compound No.	000-18-1	000-19-8	000-19-9
Poly FBA	100	100	100
Magnesia	15	5(added from Masterbato	
Lead Peroxide	3		2
Stearic Acid	3 1	3 1	-
Polyac	3	HAF Black 10	10
Cure	60' at 310°F	60' at 310°F 6	0' at 310°F
	Original F	roperties	
100% Modulus, Lbs/In ²	432	-	183
Tensile Strength,	3100	1015	1062
Lbs/In ²	1420 373	1345 415	697
Elongation, % Set at Break, %			56
Hardness, Shore A	44 48	43 55	48
Compression Set			
70 hrs at 212°F 30% Compression	52%	88.2	90•3
30% Compression	J2/8) -
Aged 70 Hours a	t asoor in Air		
	عد د درر ۰		
Tensile Strength, Lbs/In ²	1701	1139	687
100% Modulus, Lbs/In2	1442	- 25	456 226
Elongation, % Hardness, Shore A	146	7 5 81	60
Set at Break, %	13	04	14
Volume Increase	-		
in MIL-L-7808 Oil		60.0%	
70 hrs at 350°F		00.0%	

B.	T-64	trails
077 80	lymer	

Table 3

Elastomers
other
with
Blends
FBA
Pol

astomers	772-55-3	Poly FBA - 100 Paracril 18-80 - 100 Magnesia - 25 Zinc Oxide - 5 Lead Oxide - 5 Stearic Acid - 1 Stearic Acid - 1 Sulfur - 2 Altex - 10.5 HAF Black - 40 Blend these two (2) compounds in ratios of 60% FBA and 40% Paracril 18-80 by polymer weight.		A.C.	1010 455 15	4.1
Foly FBA Blends with other Elastomers	772-53-1	Foly FBA - 60 Hypalon S-2 - 40 Staybellite - Resin - 10 Magnesia - 25 Lead Oxide - 5 HAF Black - 10 Cure - 45'at 310°F		Properties	1804 148 5	65
	Compound No. 772-46-1	Poly FBA - 60 Neoprene GN - 40 Litharge - 25 Stearic Acid - 0.5 Cure - 30' at 310°F			Tensile Strength, Lbs/In ² Elongation, % 435 Set at Break, % 15 Volume Increase, %, 70 hrs in 70/30	Isocrane/Toluene at RT 40
55•	-381		13			

Controls

Polv FBA - Oxide Cured - Volume Swell in MIL-L-7808 Lubricant at 350°F*

Compound No. 000-	<u>112-1</u>	<u>12-5</u>
Foly FBA Stearic Acid Magnesia (Masterbatched) Ethyl Zimate HAF Black Cure - 90' at 310°F after 2	100 1 3 1 15 hr Oven Precure and Remill	100 1 1.5 1 15
Volume Increase	in MIL-L-7808 0il at 350°F	
150 Hours 300 Hours 800 Hours	4.2% 3.2 27.3	- 10.8% 28.0

^{*} See Table 5 for physical properties of similar compounds. This table included merely to show effect of advancing type cures on volume swell at elevated temperatures.

12	4.	- eA .
	Wil	ails

	62-1	100 100 25 25 -	858 1314 44 78	1 1 1 1 1
4	58-2	100 1 1.5 0.5	584 1167 408 50 58	- 424 57 22 23
thes for Polv	3-97	100 1 3 1 10 -	352 1323 570 74 54	at 350% - 693 115 76
Table 5	E-917	100 1 1.5 1 25 age.	Original Properties 569 1281 478 69 65	Air Oven Aged 70 Hrs at 674 680 100 90 06 13 69 83
Table 5 Masterbatched Low Oxide Curing Recines for Polv FBA	1-917	100 1 1.5 1 1.5 1.5 - 1.5 Ent cross linkage. F. Oven tempered	299 1255 667 82 52	Air Oven - 674 100 06 69
	Compound No. 000-	Poly FBA Stearic Acid Magnesia (Masterbatched) Ethyl Zimate HAF Black Sulfur Cure - Oven precured to incipient Fress cured 90' at 310°F.	100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A
WADC TR 55	- 381		15	

Table 6

Properties of Compounds in Table No. 5 After Aging in MIL-L-7808 Lubricant at 350°F

Compound No. 000-	<u>46-1</u>	<u>46-3</u>	<u> 46-5</u>	<u>58-2</u>	<u>62-1</u>
		168 Hours			
100% Modulus, Lbs/In ²	-	-	-	-	-
Tensile Strength. Lbs/In ²	1077	1176	946	-	-
Elongation, %	152	147	85	-	-
Set at Break, %	07	- -	-	-	-
Hardness, Shore A	50	58	60	-	-
		340 Hours			
100% Modulus, Lbs/In ²	-	-	-	-	-
Tensile Strength, Lbs/In ²	1170	1238	_	1120	947
Elongation, %	160	165	_	203	146
Set at Break, %	09	09	-	11	05
Hardness, Shore A	-	70	-	-	64
		570 Hours			
					_
100% Modulus, Lbs/In ²	-	-	-	-	_
Tensile Strength, Lbs/In ²	865	870	_	· •	_
Elongation, %	135	132	-	-	-
Set at Break, %	06	06	-	-	-
Hardness, Shore A	63	70	-	-	-



Table 7

Physical Properties of Accelerated Litharge Compounds - Masterbatched

Foly FBA	100	100	100						
Stearic Acid	1	1	1						
MB PbO (Fine)*	4	3	. 3						
Ethyl Zimate	2	ĺ	í						
HAF Black	35	-	35						
SAF Black	-	35	- -						
Cure - All precured 5'	at 300°F in c	wen and remilled.							
Press Cure -	15' at 310°		60' at 310°						
Oven Bake -	60' at 300°	4 hrs at 300°	4 hrs at 300°						
Physical Properties									
100% Modulus, Lbs/In ²	860	736	739						
Tensile Strength, Lbs/In ²	1416	1710	1429						

250 438
27 77
- 72

Due to poor Permanent Set - Not further tested

Elongation, % Set at Break, % Hardness, Shore A

^{*} Extra fine sublimed Litharge



Sodium Hydroxide Vulcanized Poly FBA

Recipe

Compound No.	456-41-1
Poly FBA	100
HAF Black Finely Pulverized	40
Sodium Hydroxide Cure - 60' at 310°F	1

Physical Properties

100% Modulus, Lbs/In ²	757
Tensile Strength, Lbs/In ²	1108
Elongation, %	254
Set at Break, %	25
Hardness, Shore A	7 2

This compound was made solely to gain additional information on possible cross linking methods for poly FRA. No further work carried out.



Effect of Processing Aid on Poly FBA

Recipes

Compound No. 456-	<u>87-1</u>	<u>87-2</u>	257-87-1A*
Foly FBA Stearic Acid Faraffin HAF Black TETA Cure	100 1 - 35 1 60' at	100 - 1 35 1 310°F	100 - 1 35 1 30' at 320°F
	Original Fro	perties	
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	347 1240 366 25 63	335 13 15 349 17 60	304 1407 395 17 57
Air Oven Aged	70 Hrs at 35	OOF in Open Tes	st Tubes
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	541 32 7 51 83	364 758 328 34 74	312 765 405 35 52

^{*} Compound 257-87-1 was made from later batch of poly FBA than the others. Slight differences in method of producing polymer produced noticeable changes in aging characteristics.

Con	for a	66 A
E CH	NEW	nee

		7-72	35.1.25	'on	Đ	820 1307 181 14 75		667 105 17 88		
	Sulfur Content in Foly FEA	74-3	100 1 35 1			407 1222 312 19 62	된	417 516 214 33 82		
Table 10	and	<u> 74-2</u>	100 1 35 1 1.55	Physical Froperties	Original	909 1325 179 11 74	Aged in Air Oven 70 Hours at 350°F	747 105 17 185		
Tak of Variation of Amine	of Variation of Amin	Effect of Variation of Amine	t of Variation of Amin	74-7	100 1 35 .75 .75 .75	Physic		335 1151 366 25 56	Aged in Air (Yield Foint 369 276 45 80(Flastic)
	Effect	Compound No. 456-	Foly FBA Stearic Acid HAF Black Sulfur TETA Cured - 60' at 3100F after 24 hour Rest			100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A		100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A		

Con	én a	:44
L'OH	N. C.	res

		7-77			- 1188 90 08 77		- 1106 100 10		1 1 1 1 1
	t on Poly FBA	<u>74=3</u>	ben Test Tubes		591 967 205 10 66		551 973 203 16 68		461 646 197 31 78
Table 10 (Cont'd)	mine and Sulfur Content on Poly FBA	74-2	Aged at 350°F in MIL-L-7808 Synthetic Oil in Open Test Tubes	100 Hours	1232 97 08 75	250 Hours	971 62 62 05 87	500 Hours	(Fails 180° Bend) 844 59 10 91
Table	Effect of Variation of Amine	74-1	l at 350°F in MIL-L-78		473 1025 240 25 25		427 867 263 18 53		390 580 252 49
	Effe	Compound No. 456-	Aged		100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A		100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A		100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A
WADC TR	55 - 3	381				21			

Two Poly FBA Compounds - (1) Effect of Tetrone A (2) A General Purpose Good Hardness Compound

Recipes

Compound No.	<u>456-91-1</u>	<u> 257-86-1</u>
Foly FBA Paraffin Sulfur Tetrone A HAF Black TETA Cure	100 1 - 2 35 1 60' at 310°F	100 1 1 - 40 1.25 30' at 320°F
	Original Properties	
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	253 1175 470 30	417 1133 249 15 67
Air C	wen Aged 70 Hours at 350°F	
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	544 897 234 26 78	- 785 253 25 66

Aged at 350°F in MIL-L-7808 Synthetic Oil

	250 Hours	100 Hours
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	897 1021 138 13 78	1070 158 08 68
	500 Hours	500 Hours
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	- 871 90 16 82	718 157 22 70
WADC TR 55-381	22	

			7-1	100	-	7	. X	1	1	1	a			320	1035	362	36	53
	Table 12 Various Aliphatic Amines as Curing Agents for Foly FBA Recipes		7-7	100	H	1		1	•	1.25	•			247	1127	707	بر	55
			1-3	100	ч	~	,	1	1.50	•	•		<u> </u>	332	1166	5/1 /	60	53
Table 12	Amines as Curi	Recipes	1-2	100	Н	-	35	ı	1,00	•	t		Physical Froperties	225	1082	390	71	84
	irious Aliphatic		1-1	100	-1	1	35	1,25	•	•	8.			343	1024	212	9 (28
	<u>V</u>		Compound No. 2707-	Foly FBA	Faraffin £ - 5	Sulfur	HAF Black	DETA	TEFA	Foly Amine H	Foly Amine T			100% Modulus, Lbs/In2	Tensile Strength, Lbs/In2	Liongation, %	Set at Break, %	Hardness, Shore A
DC TR	55 - 5	81										23						



Table 13

Highly Loaded FBA Compounds

		Recipes	
Compound No. 257-	43-1	<u>43-2</u>	43-3
Poly FBA Paraffin Sulfur TETA Thermax (MT) Black Cure - 30' at 320°F	100 1 1 1.25 60	100 1 1 1.25 80	100 1 1 1.25 100
Phys	ical Properties		
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	431 650 195 03 56	515 698 239 05 63	638 696 168 04 73



Effect of Two (2) Non Furnace Black Fillers on Poly FBA

Recipes

Compound No.	<u>456-86-1</u>	<u> 257-15-2</u>
Poly FBA Paraffin Stearic Acid Sulfur TETA Aerosil Alkaline Channel Black Cure - 60' at 310°F	100 - 1 1 1 - 35	100 1 - 1 1 35
	Original Properties	
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	377 1343 377 33 63	639 1178 196 10 73
Air C	ven Aged 70 Hours at 350°F	
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	525 562 198 3 7 93	995 95 15 85

Aged in MIL-L-7808 Synthetic Oil at 350°F

	250 Hours	100 Hours
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	760 1095 181 17 75	544 674 140 02 53
	500 Hours	250 Hours
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	666 861 149 22 84	578 85 03 52
WADC TR 55-381	25	



		_		
Effect of	Antioxidants	on Prop	erties of	Poly FBA

Ellect of Antion	cidants on Properti	es of Poly FDA	25
	Recipes		49. 101
Compound No. 257-	<u>61-1</u>	86-2	100 1 2-29 Compound 49.
Foly FBA	100	100	100 g h
Paraffin	1	1	1 ပွဲ မို
Sulfur	1	1	1 9 6
FEF Black	45	45	
TETA	1.25	1.25	n'n' (refer Table
Agerite Fowder	2	-	- Pre
Sulfasan R	*	1	2 🖰
Cure - 30' at 320°F			
<u>O1</u>	riginal Properties		
100% Modulus, Lbs/In ²	746*	438	469
Tensile Strength, Lbs/In ²	i092	1050	1102
Elongation, %	166	245	157
Set at Break. %	08	12	04
Hardness, Shore A	73	68	65
Air Over	Aged 70 Hours at	350°F	
100% Modulus, Lbs/In ²	461**	531	_
Tensile Strength, Lbs/In ²	580	855	965
Elongation, %	176	162	90
Set at Break, %	20	14	06
Hardness, Shore A	80	73	84
Aged at 350°F in MIL-	L-7808 Synthetic Oi	l in Open Test	Tubes
	100 Hours		
100% Modulus, Lbs/In ²	•	_	•
Tensile Strength, Lbs/In ²	993	1000	1283
Elongation, %	100	100	84
Set at Break, %	08	04	02
Hardness, Shore A	78	74	82
•	250 Hours		
100% Modulus, Lbs/In ²	-	-	_
Tensile Strength, Lbs/In ²	951	-	1137
Elongation, %	75	-	64
Set at Break, %	06	-	07
Hardness, Shore A	87	-	82
	500 Hours		
100% Modulus, Lbs/In ²	<u> </u>	-	-
Tensile Strength, Lbs/In ²	800	849	994
Elongation, %	74	92	. 60
Set at Break, %	iī	ó <u>9</u>	04
Hardness, Shore A	85	7 <u>5</u>	88
* Compound 61-1 somewha ** Compound 61-1 Aged 140	t overcured.	· .	
·	· ••• • •		

Effect of Various Fluids on Typical Foly FBA Compounds (Compound 456-30-1 - See Table 22)

1		JP-4 Fuel 70 Hrs at 2120F	Hydrocarbon Reference Fuel 70/30 Isooctane/Toluene 70 Hrs at 2120F
	Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A Volume Increase, %	1070 160 - 65 10	560 115 - 47 28(14 at RT)
27		Hydrocarbon 011 MIL-L-6081A 300 Hrs at 2500F	Synthetic Lubricant Poly Glycol base 300 Hrs at 350°F
	Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A Volume Increase, %	1114 233 16 74 3	655 106 5 75 14

* Tests run and recorded by W. W. Jackson, WADC, WCRTR-3.

Table 17

Effect of Precure Storage on Physical Properties of Poly FBA - Summer*

1-3	32 7 7		×			
			72 Hr. Delay	1-3	9617	1190 217 15
			72 H	1-2	349	309 309 315
1-2	100 1 35 1.25			1	240	1174 407 19
				៕	577	55 12 12
		8	Ħ	7	ίΛ	1162 195 12
1-1	100 1 35 1	operti	40 Hr Delay	1-2	465	1243 262 13
⊢ lj		Physical Froperties	9	1-1	285	1114 357 22
	320	Phys				
	60° at		ure	1-3	1	1069 83 10
	e Cured		Immediate Cure	1-2	906	1083 128 09
	utoclav		Imme	1-1	69†	1208 225 14
Compound No. 456-	Foly FBA Stearic Acid Sulfur HAF Black TETA Extruded and Autoclave Cured 60' at 320°F			Compound No.	100% Modulus, Lbs/In ²	Tensile Strength, Lbs/In ² Elongation, % Set at Break, %
-381			28			

* Test run in summer time in uncontrolled (non air conditioned) atmosphere.

Table 18

				eree	E LEV		
			0		s t	87-3	648 1598 175 06 60
		87-3	100 1 35 1.70		200 Hrs Rest Before Cure	87-2	414 1463 282 99 55
	rter				200	87-1	300 1379 442 19 48
	FBA - Winter	<u>87-2</u>	100 1 35 1•35		est Te	87-3	730 1409 158 04 65
	of Poly	ΦI			100 Hrs Rest Before Cure	87-2	389 1518 298 11 58
	Storage on Physical Properties of Foly FBA			Original Physical Properties	100 Be:	87-1	240 1431 417 18 50
oT eTds.T.	Physical	87-1	100 1 35	Physical	φ τ	87-3	750 1424 142 06 67
	orage on			<u>Original</u>	24 Hrs Rest Before Cure	87-2	468 1440 240 10 60
			És.		24 Be	87-1	303 1407 395 17 57
	Effect of Frecure	Compound No. 257-	Foly FBA Faraffin Sulfur HAF Black TETA Press Cured 30' at 320°F		Conditions	Compound No.	100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A
		0-			00		

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Table 19

Effect of Precure Storage on Air Oven Aging - (for Recipes see Table 18)

			[O I)	(lor hecipes see table to)	erger ee	(01				
Test Conditions	ទួប	Aged	in Forced	in Forced Convection Oven 70 Hours at 350°F (Open Test Tubes)	Tubes)	Hours at	350°F			
Compound	Compound No. 257-	87-1	87-2	87-3	87-1	87-2	87-3	87-1	87-2	87-3
Parts TETA	STA	п	1.35	1.70	ч	1.35	1.70	4	1.35	1.70
Precure	Precure Storage		24 Hours		∄	100 Hours		ଧ	200 Hours	
100% Mo	100% Modulus, Lbs/In 2	315	487	792	247	627	630	260	354	919
Tensile Lbs/I	Tensile Strength, Lbs/In ²	765	1027	1045	693	286	1181	681	935	1167
Elongation, %	ion, %	405	251	100	†8 [†]	301	168	495	343	202
Set at	Set at Break, %	35	77	20	41	16	90	9†7	16	90
Hardnes	Hardness, Shore A	52	57	29	50	53	58	24	52	55

Zou trails

Table 20

Effect of Frecure Storage on Synthetic Oil Aging (for Recipes see Table 18)

38					•	•						
31	Test Conditions		Ϋ́	Aged in (in Open Test at 350°F	Tubes in for 100.	Synthetic 250, and	oil (MII 500 Hours	Synthetic Oil (MIL-L-7808B) 250, and 500 Hours			
	o	257-		87-1	87-2	87-3	87-1	87-2	87-3	87-1	87-2	87-3
	Parts TETA			1.0	1.35	1.7	1.0	1.35	1.7	1.0	1.35	1.7
	Precure Storage				24 Hours	82		100 Hours	Ø	,	200 Hours	8
31	100% Modulus, Lbs/In	4 Q H Z	100 250 500	425 409 331	653 719 628	180°Bend OK	348 316 296	614 606 542	- 180ºBend OK	364 370 266	617 573 542	_ 180°Ber 0K
	Tensile Strength, Lbs/In ²	០ ក្ គ	100 250 500	11 7 7 984 741	1138 1002 879	962 832 814	1267 1007 660	1107 1135 900	961 996 874	1222 1017 628	1206 1143 933	1102 1090 968
	Elongation, %	жнор	100 250 500	243 236 246	152 131	97 76 70	294 281 305	170 166 176	93 82	310 302 377	159 174 167	91 88 96
	Set at Break, %	. HOH	100 250 500	08 15 18	90 90 90 90	\$ 5 6	08 08 21	06 10 07	848	88 28	90 - 90	†0 †0
	Hardness, Shore A	o tek tû	100 250 500	51 53 53	57 61 58	68 57 57	24 94 74	5,5%	62 67 68	333	55.52	3.22 3.22 3.22



Effect of Atmosphere during Precure Storage on Cured Properties

Recipe

Compound No.	<u> 257-90-1</u>
Poly FBA	100
Paraffin	1
Sulfur	1
TETA	1.25
HAF Black	35
Cured 30' at 320°F	20

Physical Properties

Frecure Storage	24 Hours in Air	180 Hours in Air	180 Hours Wrapped in Polyethylene
100% Modulus, Lbs/In ²	428	338	354
Tensile Strength, Lbs/In ²	1473	1484	1561
Elongation, %	301	344	3 15
Set at Break, %	16	15	15
Hardness, Shore A	53	40	43

Configuration Table 22

Aging Poly FBA in MIL-L-7808 Synthetic Oil at 350 °F

ARTIN TOLY FOR IN FILE-1-7000 Syllinetic OII at 370 F				
Compound No. 456-	<u> 30-1</u>	<u> 30-2</u>	<u> 30-3</u>	
Poly FBA Stearic Acid Sulfur HAF Black SAF Black TETA Cured - 60' at 310°F	100 1 1 35 -	100 1 1 - 35 1	100 1 1 - 35 1.25	
Original Phys	sical Properties			
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	315 1256 301 09 52	245 1154 450 31 53	352 1296 389 32 63	
Aged at 350°F in MIL-L-	7808 Synthetic O	il in Open Tes	Tubes	
	250 Hours			
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	587 1060 186 02 63	402 1017 302 05 61	629 1081 221 13 74	
	500 Hours			
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	526 929 167 06 63	357 935 375 18 63	539 1018 251 18 75	
	1000 Hours		,	
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	440 935 221 04 63	295 696 494 59 57	466 869 303 42 70	

Effect of Aging Poly FBA in MIL-L-7808 Synthetic Lubricant at 400°F

Recipes	
30.3	10.0

Compound No. 456-	<u>13-1</u>	<u>13-2</u>	<u> 30-3</u>
Poly FBA Stearic Acid Sulfur HAF Black SAF Black TETA Cure - 60' at 320°F	100 1 1 35 - 1 (Autoclave)	100 1 1 - 35 1	100 1 1 - 35 1.25
<u>Or</u>	iginal Properties		
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	286 134 5 2 97 08 56	348 1312 395 22 63	352 12 96 395 32 63
	Aged 100 Hours		
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	299 511 198 19 60	346 564 355 62 60	633 805 161 18 68
	Aged 175 Hours		
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	232 441 217 19 56	277 355 366 98 58	602 779 172 20 73
	Aged 265 Hours		
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	232 348 194 20 52	252 315 345 96 60	529 620 142 18 70

Table 24

Effect of Aging Poly FBA in Silicate Ester Fluid (MLO-5277) at 400°F*

Recipes

Compound No. 456-	<u> 30-1</u>	<u> 30-2</u>	<u>45-3</u>
Poly FBA			100
Stearic Acid			1
Sulfur	See Ta	ble 22	1
SAF Black			35
TETA			1.5
Cure			60' at 310°F

Poly FBA Stearic Acid Sulfur SAF Black TETA Cure	See T a	able 22	100 1 1 35 1.5 60' at 310°
Origin	nal Physical Prope	erties	
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	See Ta	able 22	771 1293 208 33 81
	Aged 50 Hours		
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	240 683 230 13 57	501 721 211 18 69	- - - -
	Aged 100 Hours		
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	23 8 583 244 19 52	511 148 19 73	805 1022 190 32 85
	Aged 175 Hours		
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	190 307 214 13 52	337 397 185 35 71	- - - -

^{*} Samples taken from molded and extruded stocks.

Effect of Aging Poly FBA at 500°F in MIL-L-7808 Synthetic Oil and in Silicate Ester (MLO-5277) (Samples Taken From Extruded Stock)

Compound No. 456-45-3 (For Recipe and Properties - See Table 24)

Aged 8 Hours

	MIL-L-7808 0il	ML0-5277 Fluid
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	715 771 145 18 75	569 780 181 32 75
Age	1 16 Hours	
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	498 543 153 38 75	296 360 191 30 64
Ageo	1 24 Hours	
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	448 495 141 38 72	225 246 157 15 57



Effect of Aging Poly FBA at 550°F in Silicate Ester MLO-8200

Compound No. 456-45-3
(For Recipe and Froperties - See Table 24)
(Samples taken from Molded Stock)

Aged 3 Hours*

100% Modulus, Lbs/In ²	549
Tensile Strength, Lbs/In ²	731
Elongation, %	218
Set at Break, %	54
Hardness, Shore A	83

Aged 5 Hours*

100% Modulus, Lbs/In ²	-
Tensile Strength, Lbs/In ²	320
Elongation, %	3 79
Set at Break, %	114
Hardness, Shore A	70

* Aged in lightly stoppered test tubes to reduce evaporation and fire hazard.

(Farenthetical figures represent additional 30' recovery at 300°F)

	(See Table 31	Table 28 (See Table 31 for further information)	ormation)		
BIL	Effect of Compounding Variations on Compression Set (70 Hours at 2500F - 30% Compression)	Variations on Compress	Compression Se ression)	PI IC	
Compound No. 257-	19-2	23-1	23-2	25-1	1-67
Foly FBA Paraffin Sulfur TETA Trimene Base HAF Black FEF Black Cured	100 1 1 60' 8	100 1 0.5 1 - 35 60' at 310°F after 1	100 1 0.5 1.25 25 er 1 Day Rest	100 1 1.25 25	100 1 1 2.5 - 35 30' at 3200]
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A Compression Set, % (70 hrs at 250°F 30% Compression after 24 hours Temper at 300°F 1/2 size discs) Hardness, Shore A	235 1224 366 10 44 27.9 (14.8)	373 1248 312 12 62 29.9 (20.7)	485 1342 223 223 3 65 (13.9)	692 1136 174 3 63 20.7	418 1188 287 8 62 35.3 (22.5)
	\tat\	2)	(1)	90	(1)

(Parenthetical figures represent additional 30' recovery at 3000F)

	information)
Table 29	for further
	32.1
	Table
	(See

		COUNT	ULL	
	55-3	100 1 1,25 - 25	238 1145 567 22 42 25.9	except using standard 1/2" thick discs.
	55-2	100 1 1 1 45 - 50' at 3100F -	304 1023 348 10 50 36.8	using standard 57
30% Compression)	55-1	100	319 813 388 17 55 49•3	others except
2500F - 30% Con	6-67	100 1 1 1-25 - 0' at 3200F	243 1125 303 303 56 34.7 (26.1)	same as
(70 Hours at	7-67	100 1 1.25 - 45 - - - - - - - - - - - - - - - - -	426 1016 256 5 68 32.7 (25.4)	20.0% (Tested
	Compound No. 257-	Poly FBA Faraffin Sulfur TETA Trimene Base FEF Black SAF Black Agerite Powder Cure	100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A Compression Set, % (70 hrs at 250°F	30% Compression after 24 hours Temper at 300°F 1/2 size discs) Hardness, Shore A
0.2		1.0		

(Farenthetical figures represent an additional 30' recovery at 300°F)

Pour	fe A	ELA
Con	N EVE	OK)

		C	wir	aik	OK)	OK)
19-1	506 872 177 06	.	883	95 58	(180° Bend - 916 95 95 08 65	(180° Bend - 838 63 05 68
18-3	321 321 882 267 12	Test Tubes	586 910 164	54.	710 1010 146 06 55	586 894 153 07 58
18-2	at 3500F in Open Test Tubes 253 828 828 325 450 15	in Open	281 1150 260	የያ የታ	361 1158 252 06 45	363 912 220 04 47
18-1	Oven Aged 70 Hours at 3500F i 253 828 325 15	in MIL-L-780	378 1051 190	05 53 250 Hours	605 1059 168 07 54 500 Hours	470 912 193 05 56
Compound No. 257.	/In ² Lbs/In ²	Aged at 3500F	100% Modulus, Lbs/In 2 Tensile Strength, Lbs/In 2 Elongation, %	Set at Break, % Hardness, Shore A	100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Flongation, % Set at Break, % Hardness, Shore A

Aging Characteristics of Compression Set Studies

Table 30

Contrac	Bend)
413	0

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	(
31	,
Table	ı
	•
	į

rs at 3500F in Open Test Tubes 251	;	Aging Characteristics	of C	Set	}	
Oven Aged 70 Hours at 350°F in Open Test Tubes bs,In² 158 251 582 628 h, Lbs,In² 425 402 169 143 Aged at 350°F in MIL-L-7808 Synthetic Oil in Open Test Tubes 100 Hours - - bs/In² 285 502 - - h, Lbs/In² 1145 1123 97 88 bs/In² 497 623 (Fails 180° Bend) - h, Lbs/In² 1093 1088 927 85 bs/In² 497 623 (Fails 180° Bend) - bs/In² 1093 1088 927 85 bs/In² 520 Hours 63 61 63 66 bs/In² 497 623 (Fails 180° Bend) - - bs/In² 1093 1088 927 85 bs/In² 520 Hours 63 6 65 66 bh, Lbs/In² 942 86 71 76 bh, Lbs/In² 942 86 71 77 bh, Lbs/In²	und No. 257-	<u>19-2</u>	23-1	23-2	<u> </u>	7-67
bs/In ² h, Lbs/In ² bs/In ² bs/In ² Aged at 350°F in MIL-L-7808 Synthetic oll in Open Test Tubes bs/In ²		70 Hours	350°F in			
1, 1, 1, 1, 1, 1, 1, 1, 1, 1, 1, 1, 1,	Modulus, Lbs/In ²	158	251	582	628 881	481 888
A Aged at 350°F in MIL-L-7808 Synthetic 0il in Open Test Tubes bs/In² h, Lbs/In² bs/In² con Hours con Ho	ation. %	125	710	169	143	219
A heed at 3500F in MIL-L-7808 Synthetic Oil in Open Test Tubes bs/In² b	t Break, %	133	917) S	70	, ZI
bs/In ² bs/	ess, Shore A	74	65	56	53	55
bs/In ² bs	Aged	൱	8 Synthetic 0	먑	89	
bs/In ² bs/		~!	00 Hours			
h, Lbs/In ² h, Lbs/In ² bs/In ² csi csi csi csi csi csi csi csi csi cs	Modulus, Lbs/In2	285	505	1	1	(180° Bend -
246 211 97 88 02 02 06 06 06 03 02 02 02 06 06 06 03 02 02 02 02 02 02 02 02 02 02 02 02 02	le Strength, Lbs/In2	1145	1123	983	803	1067
bs/In ² bs/	ation, %	546	211	26	88	< 100
bs/In ² bs/	t Break, %	90	90	င္ဝ	8	1
bs/In ² h, Lbs/In ² bs/In ²	ess, Shore A	L 17	19	63	99	82
bs/In ² h, Lbs/In ² log3 log8 228 log9 log8 228 log9 log9 log9 log9 log9 log9 log9 log9		⊘]	50 Hours			
h, Lbs/In ² 228 190 65 08 05 08 05 08 05 08 05 h, Lbs/In ² 228 190 05 08 05 05 08 05 05 05 05 05 05 05 05 05 05 05 05 05	Modulus, Lbs/In ²	267	623	180°	1	(Fails 180° Ber
228 190 < 100 05 08 05 05 08 05 71	le Strength, Lbs/In ²	1093	1088	126	980	1050
A 500 Hours bs/In ² bs/In ² bs/In ² by 2 531 (180° Bend - 0K) b, Lbs/In ² 327 531 (180° Bend - 0K) c251 205 4100 c251 205 4100 c251 205 4100 c251 205 70 71	ation, %	228	190	~ 100	ጺ	<i>ኢ</i> 3.
48 68 71 500 Hours 327 531 (180° Bend - OK) 942 845 865 251 205 < 100 12 18 05 51 70 71	t Break, %	05	08	સ્	₽.	90
500 Hours 327 531 (180° Bend - OK) 942 845 865 251 205 <100 12 18 05 51 70 71	ess, Shore A	8 [†] 7	89	17	92	33
327 531 (180° Bend - OK) 942 845 865 251 205 < 100 12 18 05 51 70 71		ч	00 Hours			
251 205 4. 251 205 6. 100 12 18 05 51 70 71	Modulus, Lbs/In2	327	531	Bend -	- 664	817
% 12 18 05 re A 51 70 71	ation, %	251	205 205	~ 100	22	~ 50
A 51 70 71	t Break, %	12	18	S	ည	1
		51	20	17	22	26

Con	-	A.
COM	wa	KES

		55-3		245	612	52	ı			333	1108	346	<u>,</u>	847		367	869 869	30.	1 22	ζ.	367	260	348	77	ဂိ
		55-2		382 742	343	19	1	Tubes		554	1062	197	9,	62		516	843	وا د	e %	;	797	759	217	07	70
	Compression Set Studies	E	Tean Inpea	560 260	425	62	1	- 1		1	850	35	01	82		1	772	† 0°	8 8 8	3	1	787	~ 100	7 %	8
Table 32	of Compression	19-3	3200 II Open	2 6 0 890	332	90_	35	Synthetic 011	Hours	545	870	165	70	50	250 Hours	599	975	747		500 Hours	667	802	155	5 (53
Ta	Aging Characteristics	2-67	ed (U Hours at	360 804	261	90	50	at 350°F in MIL-L-7808 Synthetic Oil in Open Test	100	299	626	241	ይ	62	250	771	766	145	90¢		750	862	120	Š	70
	Aging Ch	Compound No. 257-	Oven Aged	100% Modulus, ${ m Lbs/In}^2$ Tensile Strength, ${ m Lbs/In}^2$	Elongation, %	Set at Break, %	Hardness, Shore A	Aged at 350°F		100% Modulus, Lbs/In ²	Tensile Strength, Lbs/In2	Elongation, %	Set at Break, %	Hardness, Shore A		100% Modulus, Lbs/In2	Tensile Strength, Lbs/ ${ m In}^2$	Elongation, %	Set at Break, %	naromess, sucre a	100% Modulus, Lbs/In2	Tensile Strength, Lbs/In2	Elongation, %	Set at Break, %	Hardness, Shore A

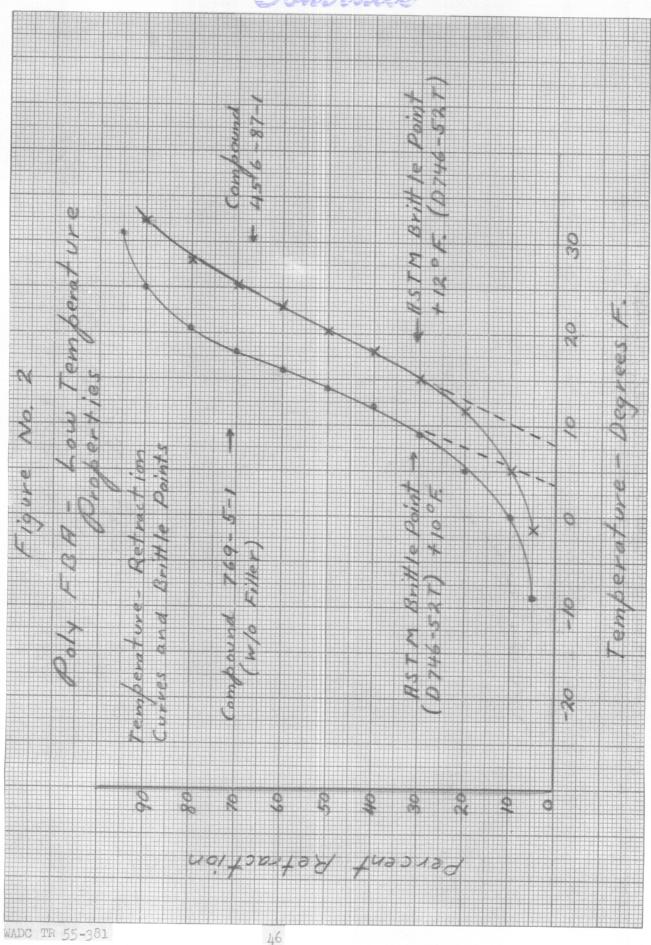


Table 33

Poly FBA Resistance to Red Fuming Nitric Acid

Compound No. 456-	<u> 30-1</u>	<u>45-3</u>	74-4
Poly FBA Stearic Acid Sulfur HAF Black SAF Black TETA Cured - 60' at 310°F	100 1 1 35 - 1	100 1 1 - 35 1.5	100 1 2 35 35 1•5
Origin	al Properties	Ŀ	
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	315 1256 301 09 52	771 1293 208 33 81	820 1307 181 14 75
Aged 70 Hours	in RFNA (Roc	om Temp.)	
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	149 579 514 35	130 592 542 50	81 492 510 37 -
Aged 200 Hours then rinsed 24			
100% Modulus, Lbs/In ² Tensile Strength, Lbs/In ² Elongation, % Set at Break, % Hardness, Shore A	346 1278 329 20 50	256 1007 448 55 -	323 1163 345 27 -

15 mil Film Minutes



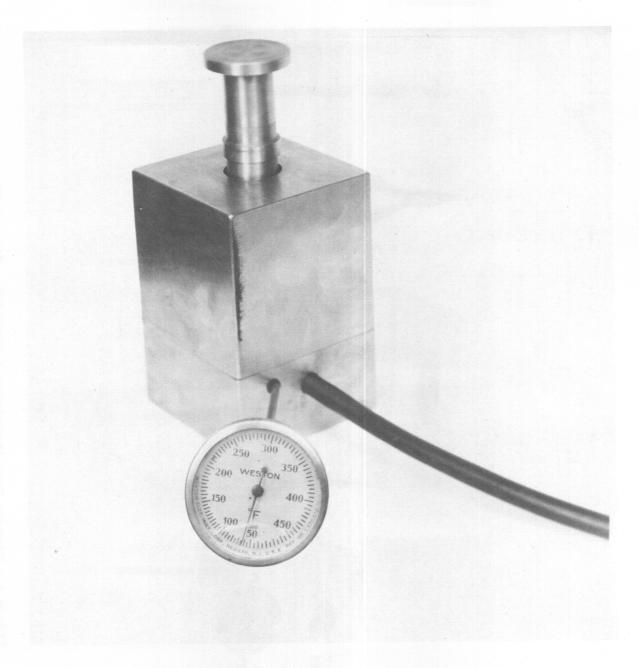
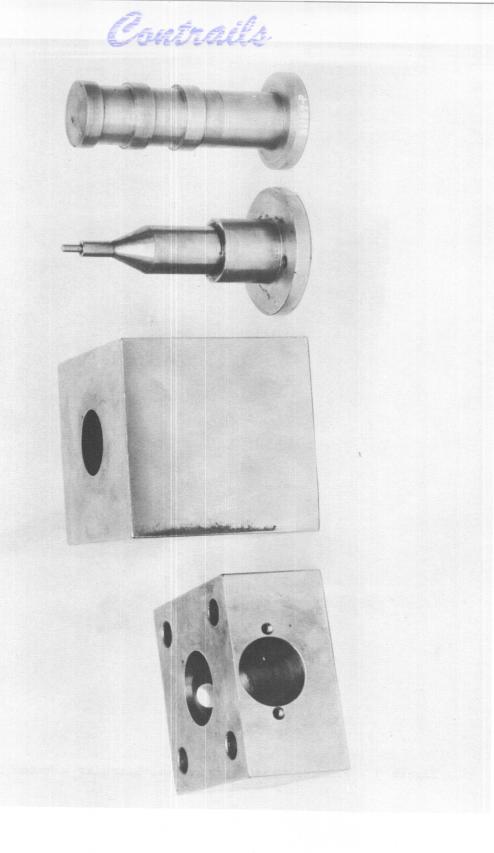


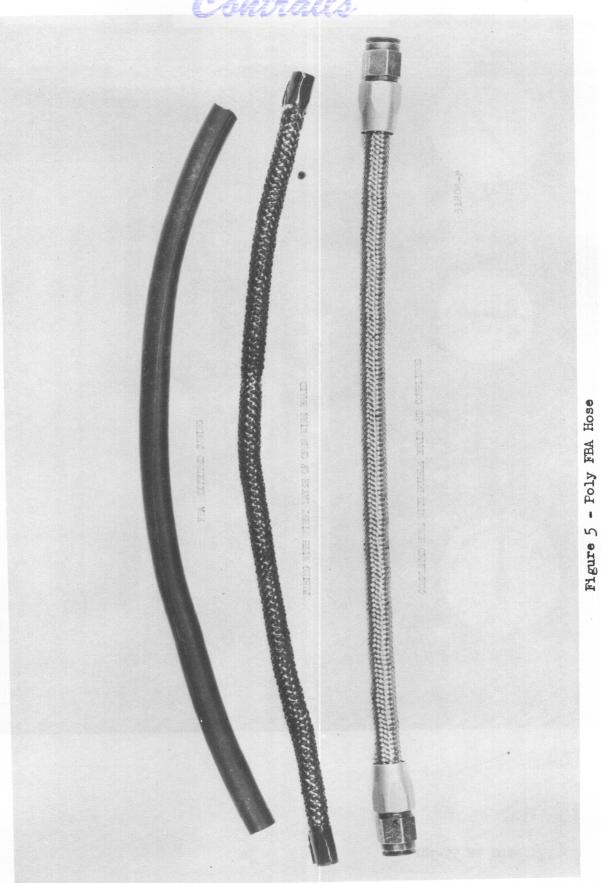
Figure 3 - Plunger Type Experimental Extruder - Assembled





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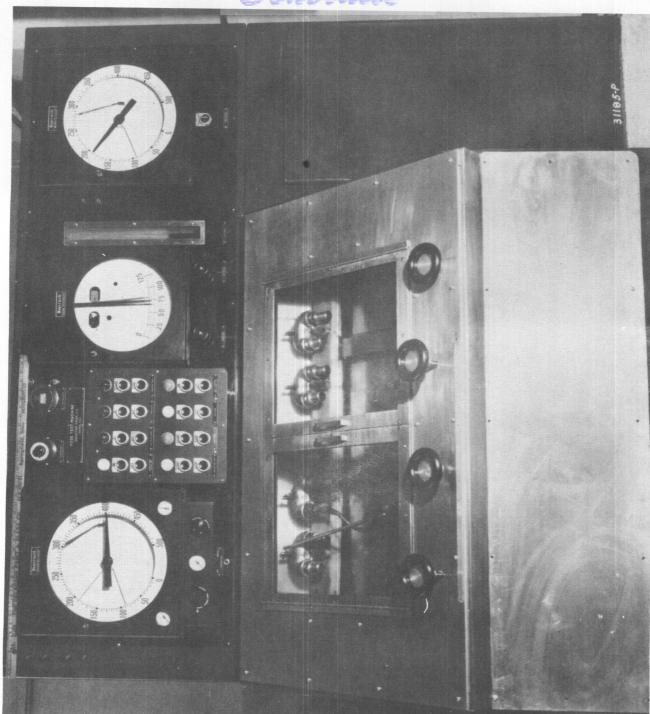
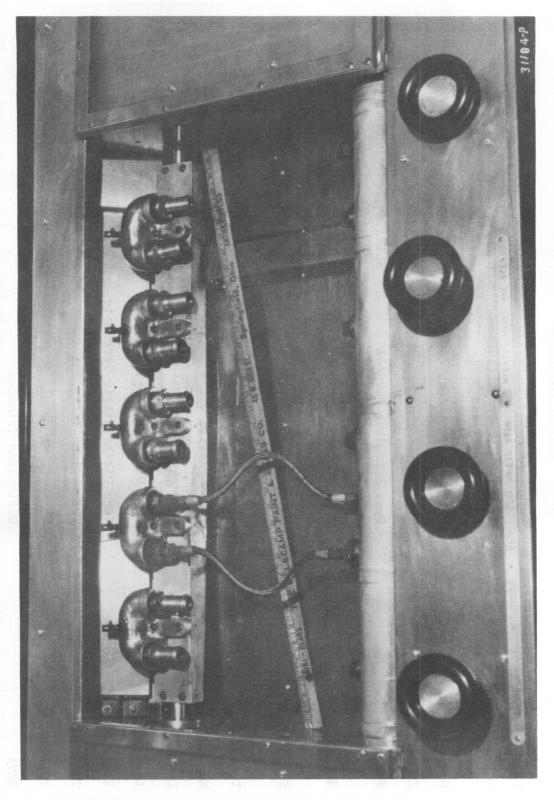


Figure 6 - Materials Laboratory Hose Tester



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Figure 7 - Close Up View of Hose Test Chamber

APPENDIX III

List of Materials

	STOT TO AST	
Common Name or Symbol	Technical Name	Source
Foly FBA	Poly 1,1 dihydroperfluoro- butyl Acrylate	Minnesota Mining & Mfg. Company
Folyac	Foly Para Dinitroso Benzene	E. I. duFont de Nemours & Co., Inc
EFC Black	Easy Processing Channel Black	•
FEF Black	Fast Extruding Furnace Black (Philblack A)	Phillips Petroleum Company
HAF Black	High Abrasion Furnace Black (Fhilblack 0)	Phillips Petroleum Company
SAF Black	Super Abrasion Furnace Black (Philblack E)	Phillips Petroleum Company
Texas 109 Black	Alkaline Channel Black	Sid Richardson Carbon Company
Magnesia	ELC Magnesium Oxide	1
Litharge	Lead Oxide (PbO)	1
Lead Peroxide	Lead Dioxide (FbO ₂)	1
Ethyl Zimate	Zinc Diethyldithio Carbamate	R. T. Vanderbilt Company
TETA	Triethylene Tetramine	i
Trimene Base	ı	Naugatuck Chemical Company
DETA	Diethylene Triamine	1

AFFENDIX III

List of Materials (Cont'd)

Technical Name	Source
Tetraethylene Pentamine	
ì	Carbide & Carbon Chemical Company
ŧ	Carbide & Carbon Chemical Company
Phenyl Beta Naphthylamine	R. T. Vanderbilt Company
Morpholine Disulfide	Monsanto Chemical Company
Very Fine Silica	Godfrey L. Cabot Company
<u> </u>	tractnylene rentamine - enyl Beta Naphthylamine rpholine Disulfide ry Fine Silica

The synthetic lubricant conforming to Specification MIL-L-7808 used in these tests was Esso Turbo Oil No. 15, a product of The Standard Oil Company of New Jersey. Note:

The two "Silicate Esters" (MLO-5277 and MLO-8200) used in these tests were organo-silicon compounds developed under Air Force Contract by the California Research Corporation.